

Investigation of Co Decorated N, S Doped Reduced Graphene Oxide (Co/NS-rGO) Catalysts For Oxygen Reduction Reaction (ORR) and NH₃ Electrosynthesis through Nitrate Reduction Reaction (NO₃RR)

by

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ABSTRACT

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One of humanity's most challenging problems in our century is enabling sustainable and carbon-zero energy, chemical, and fuel production and minimizing the use of fossil fuel sources to reduce CO₂ emissions and decelerate climate change. Two crucial routes to decarbonize energy consumption are the widespread commercialization of proton exchange membrane fuel cells (PEMFC) for transportation and the environment-friendly NH₃ electrosynthesis. It is essential to understand reaction kinetics, adsorption tendencies, and current performance of feasible catalysts to actualize the use of affordable and high-performance catalysts for oxygen reduction reaction (ORR) at PEMFCs' limiting cathode compartment and to provide a pathway for zero-emission NH₃ electrosynthesis that compensate for NH₃ production of carbon-intensive Haber-Bosch process. Comparably cheap and abundant Co and high-surface-area graphene-based Co decorated N, S Doped Reduced Graphene Oxide (Co/NS-rGO) catalysts with different metal compositions and heat treatments are investigated for NH₃ electrosynthesis through nitrate reduction reaction (NO₃RR) and ORR for their activity and selectivity.

In the first part of the thesis, it is unraveled that the optimum composition of Co enables the formation of ORR-active CoS species and homogenous distribution of Co Single Atoms (SAs) steer the selectivity towards 4e⁻ pathway that is desired for PEMFC and provide the lowest overpotential with the highest current output. With the help of rotating disk electrode (RDE) and rotating ring-disk electrode (RRDE) techniques that eliminate the effect of mass-transfer limitations the number of electrons transferred (n_e⁻) is measured for Co/NS-rGO as 4e⁻ throughout the studied potential range as of industrial PEMFC catalyst of Pt/C while the 2Co/NS-rGO catalyst that has higher Co composition with CoO species, favored more HO₂⁻ production. Furthermore, the origin of the catalysts' kinetic activity and the materials' electronic properties are investigated using

electrochemical impedance (EIS) techniques. Co/NS-rGO catalyst that has the highest performance obtained a higher density of states ($D(E_F)$) around the Fermi level for the electroreduction, higher charge carrier density concentration (N_D) and the flatband potential. Also, electrochemical surface area (ECSA) normalized RDE and RRDE figures indicate the maximum utilization of electrochemically active sites for the Co/NS-rGO catalyst, which has the highest charge storage properties detected by EIS capacitive measurement.

In the second part of the thesis, the electrochemical NO_3RR -to- NH_3 activity of NS-rGO and its Co incorporated and/or 900 °C pyrolyzed counterparts are investigated. It is observed that NS-rGO is inclined to generate high amounts of NO_2 while unable to convert NO_3 to NH_3 with high efficiency. On the other hand, pyrolysis facilitates hydrogen evolution reaction (HER) for both metal and non-metal catalysts. Co incorporation is claimed to promote NH_3 yield rate and Faradaic Efficiency (F.E.) significantly at lower overpotentials with higher current outputs. A mechanistic approach using X-ray Absorption Spectroscopy (XAS), EIS, EIS-driven Distribution of Relaxed Species (DRT), and Tafel analysis enabled us to find that the Co/NS-rGO follows a mass transfer limited mechanism. At the same time, Co/NS-rGO 900 is driven by charge transfer limitations and protonation properties. Furthermore, mechanistic studies show that NO_2 desorption plays a significant role in NO_3RR . To modulate and increase NH_3 yield and F.E., a pulsed electrolysis strategy is inherited at both mass-transfer and charge-transfer limited potential regimes. Pulsed electrolysis results show that Co/NS-rGO catalyst with a very high NH_3 yield rate and F.E. for NO_2RR can increase its NH_3 F.E. by two-fold and yield rate by 3-fold through pulsed electrolysis. In contrast, the pulsed strategy facilitates HER and suppresses NH_3 production efficiency for the Co/NS-rGO 900 catalyst.

ÖZETÇE

Kobalt Gömülü Azot ve Kükürt Katkılanmış İndirgenmiş Grafen Oksit (Co/NS-rGO) Katalizörlerin Oksijen İndirgenmesi ve Nitrat İndirgenmesi Reaksiyonları İçin İncelenmesi

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Çağımızda insanlığın en zorlayıcı problemlerinden birisi sürdürülebilir ve karbon-sıfır enerji, kimyasal ve yakıt üretimini sağlamak ve fosil yakıt kaynaklarının kullanımını azaltarak karbon dioksit salınımını düşürmek ve iklim krizini yavaşlatmaktır. Enerji tüketimini dekarbonize edebilmek için iki önemli yol proton değişim membranı yakıt pillerinin ve çevre dostu amonyak elektrosentezinin yaygın ticarileşmesini sağlamaktır. Katalizörlerin reaksiyon kinetiklerini, bağlanma isteklerini ve akım performanslarını anlamak proton değişim membranı limitleyici kompartmanında olan oksijen indirgenmesi reaksiyonunda kullanılmak için satın alınabilir ve yüksek performanslı katalizörlerin kullanımını sağlamak ve sıfır emisyonlu amonyak elektrosentezini Haber-Bosch sürecini tazmin edebilecek kadar ilerletebilmek adına çok önemlidir. Ucuz ve yaygın kobalt yüksek yüzey alanlı grafen bazlı Co/NS-rGO katalizörlerinin farklı metal kompozisyonlarında ve farklı ısı işlemlere tabi tutulmuş halleri nitrat indirgenmesiyle amonyak elektrosentezi ve oksijen indirgenmesindeki seçicilikleri ve aktiviteleri için test edilmiştir.

Tezin ilk bölümünde, kobaltın optimum kompozisyonunun oksijen indirgenmesi aktif CoS türlerinin ve kobalt tekli atomlarının homojen dağılmasının katalizörün proton değişim membranı yakıt pilleri için de istenen 4 elektron yönünde seçici olduğunun ve en az aşırıgerilim ve en yüksek akım çıkışına sahip olmasını sağladığı görülmüştür. Katı-transferi limitlerini yok eden dönen disk elektrot ve dönen halka-disk elektrot tekniklerinin transfer edilen elektron sayısının Co/NS-rGO için çalışılan potansiyel aralığında endüstriyel katalizörler gibi 4 elektrob olduğu bulunmuştur. Aynı zamanda daha çok kobalt içeren ve kobalt oksit içeriği olan 2Co/NS-rGO katalizörlerinin peroksit formasyonuna yol açtığı bulunmuştur. Sonrasında, katalizörlerin kinetik aktivitelerinin

orijini bulmak için materyallerin elektronik özellikleri elektrokimyasal impedans spektroskopisi teknikleri kullanılarak incelenmiştir. En yüksek performansa sahip Co/NS-rGO katalizörünün Fermi seviyesi etrafında en yüksek durum yoğunluğuna, en fazla yük taşıyıcı yoğunluğuna ve en yüksek düz bant potansiyeline sahip olduğu bulundu. Ayrıca, elektrokimyasal yüzey alanıyla normalize edilen dönen disk elektrot ve dönen halka-disk elektrot deneyleri sonuçlarının normalize edilmeyen halleri gibi yine elektrokimyasal aktif alanlarının maksimum kullanılabilirliğinde olduğu gözlenmiştir. Ayrıca, elektrokimyasal impedans teknikleri Co/NS-rGO katalizörünün kapasitif özelliklerinin en yüksek olduğunu bulmuştur.

Tezin ikinci bölümünde, elektrokimyasal nitrattan amonyağa aktivitesi kobaltlı veya kobaltsız; ısıyla eritilmiş veya eritilmemiş kükürt ve azot katkılı indirgenmiş grafen oksit katalizörleri için incelenmiştir. Bulgular kobaltsız katalizörün nitrit üretimini desteklediği ve nitrata amonyağa dönüştürmekte çok etkin olmadığını göstermiştir. Aynı zamanda, ısıtma eritiminin hidrojen oluşumu reaksiyonunu desteklediği ve kobalt gömülmesinin de amonyak seçiciliğini ve verimini arttırdığı gözlenmiştir. Mekanizma yaklaşımı XAS, EIS, DRT modeli ve Tafel analizinin Co/NS-rGO için katı-transferi limitli bir mekanizmaya sahip olduğunu gösterir. Aynı zamanda, ısıyla eritilmiş kobaltlı katalizörün ise yük transferi limitlemelerinin olduğu ve protonlama özelliklerinin düşük olduğu gözlenmiştir. Üstelik, mekanizma çalışmaları nitrit desorpsiyonunun nitrat indirgenmesinde önemli bir rolü olduğunu gösteriyor. Nitriti modüle etmek ve amonyak verimi ve seçiciliğini arttırmak için, kısa aralıklarla elektroliz stratejisi seçilmiştir katı-transferi limitli ve yük transferi limitli rejimlerde. Kısa aralıklı elektroliz sonuçları çok yüksek amonyak verim hızı ve seçiciliğiyle nitrit indirgenmesini gerçekleştiren Co/NS-rGO katalizörünün nitrat indirgenmesi için amonyak seçiciliğini 2 kat ve verimini 3 kat arttırdığı gözlenmiştir. Fakat, ısıyla eritilmiş kobaltlı katalizör hidrojen oluşumu reaksiyonunu takip etmiş ve amonyak üretimi bastırılmıştır.

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Chapter 1:

INTRODUCTION**1.1 Catalyst Design for the Electrochemical Energy Applications: Co decorated N, S Doped Reduced Graphene Oxide (Co/NS-rGO)**

After the Industrial Revolution, people's need for energy increased exponentially, and to meet this demand, high amounts of fossil resources needed to be utilized [1]. The high consumption of fossil fuels has caused a significant increase in CO₂ emissions, which has caused climate change and is threatening the habitats of many living species, including humans. The need for energy increases every day and causes the use of more fossil fuels, which can cause the earth to be uninhabitable in the foreseeable future [2]. To prevent these disasters from happening, sustainable energy production needs to be put to use in a rapid, affordable, and effective way in the shortest time. Much industrial and academic research is carried out to balance and limit fossil-fuel-driven systems' use [3], [4]. Recently, electric cars powered by lithium-ion batteries, hydrogen-fueled automobiles with proton exchange membrane fuel cells (PEMFC), and large-scale CO₂ electrolysis to produce C_xH_yO_z and NH₃ electrosynthesis through N₂ (NRR) and NO₃⁻ reduction (NO₃RR) are some of the crucial steps to decarbonize energy production [5], [6], [7], [7], [8], [9].

The cost and operability of sustainable energy systems heavily depend on the catalyst design. Pt/C catalysts are used in the performance-limiting compartment of PEMFC systems to supply enough energy for oxygen reduction reaction (ORR) to carry out and ensure system durability [10], [11], [12]. On the other hand, to this day, there is no definitive catalyst for NH₃ electrosynthesis, while the energy-intensive Haber-Bosch process primarily depends on industrial Fe catalysts [13]. It is essential to synthesize durable, cost-effective, high-performance, and selective catalysts to actualize widespread use of these sustainable energy technologies to replace fossil-driven energy production [14], [15].

By definition, catalysts are materials in the surface reaction that happens on the surface without the catalyst remaining unchanged. To enable efficient electrocatalysis with a high yield of the target product, the catalyst must have a high surface area and moieties

that are selective for the desired catalysis process[8], [12]. The catalyst's activity can be evaluated by its number of active sites and the intrinsic activity or turnover frequency of these active sites for the catalysis[16], [17]. A tremendous amount of effort is put into synthesizing high-surface area electrocatalysts that have a significant amount of active sites and active sites that are active for the selective catalysis process such as N sites such as pyridinic N[18], atomically dispersed metal atoms of Pt, Fe, Co in high-surface-area Metal-N-C catalysts[19], [20], catalytic pairs of Metal-S, Metal-P in chalcogenides[21], [22] and tandem electrolysis on core-shell structures of noble and non-noble metals[23], [24].

Graphene is selected as catalyst support to provide a high surface area and enhance the number of active sites[25]. It holds superior properties owing to its unique electronic[26], [27], [28], thermal[29], and optical properties[30]. The high surface area of graphene allows the dispersion of a high amount of nanoparticles or single atoms on its surface[31], [32], [33] and makes an efficient platform for heterogeneous catalysts. It is impossible to actualize large-scale graphene production using methods such as epitaxial growth[34] or mechanical exfoliation of graphite (Scotch tape)[35], which provides single or few-layer graphene[36]. However, the chemical route to synthesize graphene oxide (GO) [36], [37] follows the exfoliation of graphite through strong oxidation and can enable bulk manufacturing of graphene-like materials with defects and oxygen functionalities[38], [39] through reduction. The production of contamination-free GO, i.e., using phosphoric acid to eliminate Mn content⁶¹, through chemical reduction, enables graphene production on a bulk scale, and oxygen functionalities in GO enable manipulation and engineering through doping and heat treatment for different applications[40]. Even though the aforementioned superior properties of graphene exist, zero band gap energy or defect-free structure of the material makes it insufficient for energy applications[26], [41]. Also, metal incorporation to pristine graphene is not favored due to weak bonding[42]. When graphene is functionalized upon doping, its open structure, high surface area, conductivity, and stability make it an excellent catalyst support[43], [44]. Properties that can be attained by doping of graphene and other carbon derivatives, i.e., nanotubes and amorphous materials, are investigated primarily with N-doping[45], [46], [47], [48] and multi-heteroatom doping[49] with different elements, such as S[42], [50], [51], P[52], [53], [54] and B[55].

N-doped graphene[45], [56] is thoroughly investigated for the ORR and found to have high activity, durability, and low overpotential during catalysis. N-doping changes the atomic charge distribution on the material[57] and introduces multiple active sites, higher surface area, disorder, and defects, which become anchoring sites for metals[58], [59]. N changes the surface electronic distribution of solely sp^2 hybridized graphene through partial charge difference. During the process, C and N induce partially negative and positive charges ($C^{\delta+} N^{\delta-}$) by C-N bonding[47]. Doping can provide different N active sites, such as pyrrolic, pyridinic, and graphitic N[45]. There is no clear conclusion about the active center for oxygen electroreduction. In many studies, it is proposed that high ORR activity emerges from existing pyridinic N sites[60], [61]. In contrast, Yang et al. claim that electron-donating quaternary N sites are responsible for high ORR activity[62]. Also, carbon atoms adjacent to pyridinic N or sp^2 carbons next to oxide regions are proposed to be active sites for ORR. Since pyridinic N can impact the electronic structure of surrounding atoms, it is claimed to generate more active sites for ORR compared to its counterparts (pyridinic, graphitic N)[18], [63], [64], [65]. According to another report, limiting current density is determined by graphitic N, while the reduction in overpotential and $4e^-$ selectivity is dependent on the presence of pyridinic N.[61]. On the other hand, Li. et al. investigated Cu_xO/N -doped Graphydiyne material enriched with pyridinic, pyrrolic, and graphitic N sites. They found that the synergistic effects between pyridinic N and Cu_xO achieve 85% F.E. with a high NH_3 production rate [66]. Similarly, Zhu et al. introduced that the interactions between pyridinic N and adjacent carbon atoms steer the selectivity towards NH_3 by 99.24 % and enhance the production yield [67].

Even though not as frequently investigated as N-doping, heteroatom doping of graphene materials boosts the kinetics of electrochemical reactions and improves the adsorption abilities of the material[25], [49]. Heteroatom doping involves different S, F, or B atoms studied in different electrochemical processes[50], [52]. The high oxophilic character of S[68] and its ability to form catalytically active metal sulfide species lower the energy barrier for adsorption [69], [70] and subsequent steps of various electrochemical reduction reactions[71], [72]. Surprisingly, while phosphorus is used intensively for high-activity NO_3RR catalysts[21], [73], [74], S is not used as a doping element, or its role for the catalysis is studied for NO_3RR . Although it is not investigated for NO_3RR , N, S doped

carbon-based catalyst and catalyst supports are thoroughly examined for ORR[53], [75]. Doping of graphene with S can introduce additional defects due to its larger atomic size and more stable structure[76], [77]. Opposed to N, the electronegativity difference between S and C is lower, and it is claimed to have a positive effect on the ORR by facilitating oxygen chemisorption[78]. The presence of sulfur is postulated to facilitate the protonation transfer step in ORR. It alleviates the reduction of adsorbed oxygen ($O_{2, aq}$) to superoxide ($O_2^{\cdot-}$), which is claimed to be the rate-determining step of ORR[76], [79]. Also, it is proposed that co-doping of graphene with N and S can present defects, more active sites, asymmetrical charge, and spin density distribution on graphene, which results in higher electrical conductivity[50], [80], [81], [82]. Since the electronegativities of C and S are very close, instead of adjacent C becoming positively charged for N-doped C, S is positively charged through a mismatch in the outermost orbital of S and becomes an active site for the ORR[50], [78]. High ORR activity emerging from S and N co-doped graphene-based catalysts relates to synergistic effects between dopants[50], [76], [83], [84]. It is found that heteroatom doping can also enhance metal/support interactions [85], [86].

Co is a cheap and abundant catalyst and is used in different electrochemical reactions such as oxygen reduction reaction (ORR)[87], [88], oxygen evolution reaction (OER)[89], and carbon dioxide reduction reaction (CO_2RR)[90] thanks to its partially filled d orbital and the improvability of many catalytic sites[91]. Co incorporation to heteroatom-doped graphene matrix is further investigated to increase ORR catalysts' activity and durability[44], [59], [92], [93], [94], [95]. Liang et al. synthesized a Co_3O_4 /N-rGO hybrid, where he explained the high ORR activity with the synergistic coupling between Co_3O_4 and N-doped reduced graphene[59]. Yang et al. claimed that both Co-N-G and Co-N, S-G, and Co-N_x sites act as the active sites for the ORR. Higher activity of Co-N and S-G is associated with better mass transport and electron-rich characteristics of N due to the presence of S[92]. Also, it is proposed that metal incorporation to carbon ends up with non-metal ORR active sites. In another study, a metal composite catalyst subjected to heat treatment with impregnation of Co-Fe-N chelate complex is synthesized, showing low overpotential and no degradation in 480 h of continuous fuel cell conditions. It is found that the decomposition of a metal chelate complex during heat treatment accelerates the formation of ORR active N species at lower temperatures[96], [97]. Co

catalysts are extensively used in NO_3^- and NO_2^- reduction reactions as atomically dispersed Co metal-nitrogen-carbon[20] or Co-P catalytic pair[21], Co_3O_4 nanoarray[98], core-shell CoP@Co [74] and $\text{Cu/CuO}_x - \text{Co/CoO}$ phases[99], and Co metalloprotein[98]. Murphy et al. found that despite the high NH_3 F.E. of the Co-N-C catalyst for NO_2RR , the Co-N-C catalyst suffers from low selectivity towards ammonia due to the inability of NO_3^- activation for NO_3RR . Furthermore, they signified the different effects of bulk NO_2^- and locally channeled NO_2 by examining NO_2^- F.E. after NO_3RR and NH_3 yield after NO_2RR , eventually telling that rather than a direct $8e^-$ mechanism for NO_3RR , a complex mechanism might have been driven by the NO_2 desorption[20]. In another article, Ni et al. show that the Co sites in Co-P catalytic pairs favor the activation of NO_3^- and P sites, allowing water disassociation to take place to supply H^+ needed for proton-coupled electron transfer for NO_3RR [21].

Single-atom catalysts (SACs) offer maximum atomic utilization, high activity, selectivity, and stability as an emerging frontier in heterogeneous catalysis. Instead of metallic bonding, SACs form through bonds between metal and heteroatoms on support surfaces[100], [101]. SACs are found to have low overpotential and high activity towards ORR[102], [103], [104]. Yin et al. found out that Co SAs/N-C(900), which has dominant reactive sites of Co-N₂ followed near the four-electron pathway and onset potential of 0.881 V[104].

It has been shown recently in operando studies[105], [106] that as claimed active sites, i.e., Co-N₄ moieties[105] and Co SACs[106], tend to show higher F.E. for H_2O_2 production via $2e^-$ transfer pathway while Co-N₂ moieties favor $4e^-$ mechanism[105]. It can be postulated that Co-N₂ moieties and cobalt sulfide species in a catalyst as tandem ORR active sites can accelerate the $4e^-$ mechanism. A theoretical study uses quantum computations that cobalt sulfide surfaces show ORR activity similar to platinum. S^{2-} ions in cobalt sulfide surfaces can be an adsorption site for oxygen, facilitating O-O bond cleavage and initiates $4e^-$ mechanism[107]. Furthermore, for NO_3RR , Ren et al. used a bioinspired approach where they studied Cd single-atom anchored $\text{Mo}_2\text{TiC}_2\text{T}_x$ catalyst to achieve >95% Faradaic efficiency (F.E.) and $48.5 \text{ mg h}^{-1} \text{ cm}^{-2}$ NH_3 yield rate by activating NO_3^- by Mo active center as of nitrate reductase and suppressing HER simultaneously[108]. Murphy et al. utilize a FeMo-N-C atomically dispersed catalyst that follows a cascade NO_3RR pathway through associative and disassociative adsorption of

NO_3^- . They could effectively adsorb and reduce NO_3^- at Mo-N₄ sites to desorbable NO_2^- and $^*\text{NOOH}$ to NH_3 on Fe-N-C. The catalyst achieved 94% F.E. and a high stability of 90% F.E. for 60 h[109].

Cobalt sulfides, i.e., Co_9S_8 , CoS , CoS_2 , show robust electrochemical activity for different applications owing to their accessible high surface area, high capacitive behavior, and intrinsic activity[110]. CoS is used in different areas, such as lithium-ion batteries[111], [112], [113], oxygen evolution reaction (OER)[89], [93], [114], hydrogen evolution reaction (HER)[115], [116], and ORR[93], [94], [95]. $\text{CoS}_2(400)/\text{N, S-GO}$ catalyst was synthesized using cobalt, thiourea, and GO with solid-state thermolysis at 400 °C. High ORR performance of the catalyst where cobalt sulfide nanoparticles dispersed on N, S doped graphene oxide is correlated with strong coupling between CoS_2 and N, S-GO. XPS could detect the formation of CoS_2 [93]. Ma et al. also associated high ORR activity with synergistic effects between Co_3O_4 and N, S dual-doped graphene foam[44] whereas Liu et al. related superior ORR activity to synergistic effects between NiCo_2S_4 nanoparticles and N/S-rGO[94]. To our knowledge, NO_3RR studies do not comprehensively examine the activity and selectivity of cobalt sulfide or metal sulfide catalysts. One study only included metal sulfides as a synthesis intermediate to produce their primary catalyst[99].

1.2 Investigation and Elucidation of Electrochemical Reactions

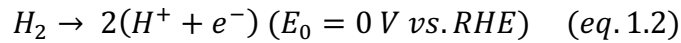
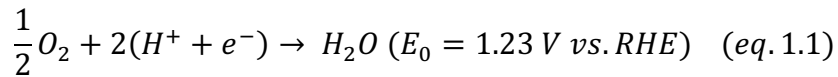
Understanding and enhancing the aforementioned electrochemical reactions is essential due to the design and operation of catalysts and systems that can make their respective applications economically viable worldwide[91]. The performance of a catalyst can be categorized by its tendency to follow (1) favorable adsorbable and desorbable intermediates by the catalyst along the associative or disassociative chemisorption processes[117], (2) the slowest reaction step is for the catalyst to drive the reaction: rate-determining step (RDS)[118], [119], (3) the potential loss to perform the reaction concerning reaction's thermodynamic onset potential: overpotential[117], [120], (4) how fast the catalyst carries out the electron transfer step: Tafel analysis[121], [122], (5) number of electrons transferred (n_{e^-}) for the reaction in which potential range[123], [124], [125] and (6) the selectivity of the catalyst towards the desired electrochemical reaction to facilitate the specific energy application[126], [127], [128]. On the other

hand, a catalyst must be durable and preserve its desired performance during the projected operation lifetime.

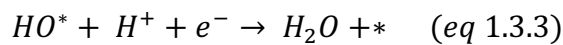
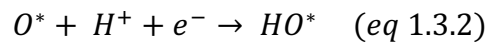
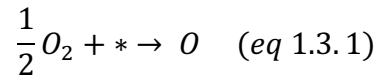
Some of the most studied and crucial electrochemical reactions can be listed as hydrogen evolution reaction (HER), carbon dioxide reduction reaction (CO₂RR), NRR, oxygen evolution reaction (OER), ORR and NO₃RR -to-NH₃. The scope of this thesis focuses on the investigation of ORR and NO₃RR using Co/NS-rGO catalysts.

1.2.1 Oxygen Reduction Reaction (ORR)

Hydrogen fuel cells are expected to acquire widespread commercial use soon due to their advantages such as high efficiency, no environmental pollution, and unlimited source of reactants. One of the main challenges for fuel cells in replacing thermal engines is the sluggish kinetics of oxygen reduction reaction (ORR) at the cathode. The cathode requires $\sim 0.4 \text{ mg cm}^{-2}$ of Pt loading for the commercial Pt/C catalyst to achieve the desired performance[129]. Considering scarcity and high cost, intensive research is being done to develop a non-noble metal catalyst that can replace Pt/C catalysts[72]. Performance-limiting cathode reaction ORR (eq.1) and hydrogen oxidation reaction (HOR) (eq. 2) equations at the anode are shown below:

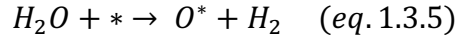
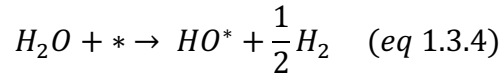


According to Nørskov et al., The mechanism of ORR can be categorized into two different mechanisms: dissociative and associative mechanisms. The steps of the associative mechanism can be listed as follows[117]:

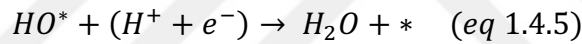
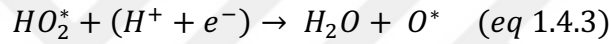
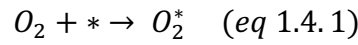


* stands for sites on the surface.

Further decompositions of H_2O for associative mechanism can be equated as:



The disassociative mechanism does not consider the O_2 disassociation and leads to either peroxy intermediates or the end-product of water. The steps of the disassociative mechanism can be listed as follows[117]:



ORR can acquire separate n_e^- along different reaction pathways in alkaline environments, the $4e^-$ to HO^- , $2e^-$ HO_2^- to, and $2e^- + 2e^-$ HO^- pathways[11]. The former produces HO^- , which is more desirable for fuel cell applications, while the latter produces HO_2^- , which can lead to fuel cell poisoning and activity loss but can be utilized to produce industrial chemical hydrogen peroxide[125]. Different ORR electron transfer pathways in alkaline media are shown in the figure below[130]:

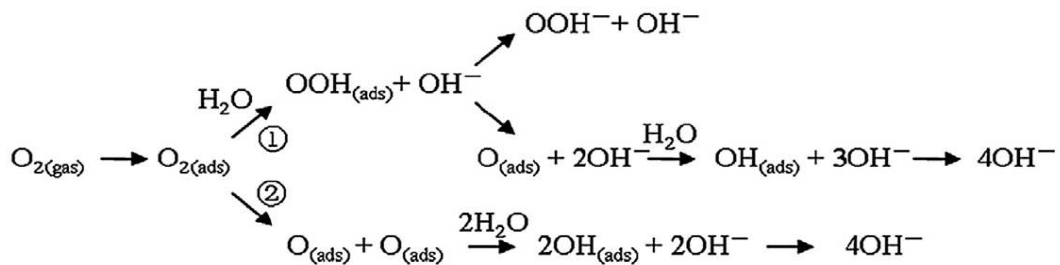


Figure 1-1: ORR Electron Transfer Pathways in Alkaline Media[130]

The origin of the ORR activity of materials can be investigated by examining their electronic properties[131]. Gerischer proposed that determining parameters for electron-

transfer kinetics of semiconductors is the position of electrons or holes and their density of states (DOS) near the Fermi level[124]. N-doping not only enhances the ORR activity of the catalyst but allows zero-band gap semiconductor graphene to change its band and electronic structure and obtain n-type semiconductor properties[26], [63], [131], [132]. Also, it is found that N-doping not only induces electronic states around the Fermi level in its bandgap but is the primary reason for the increase in DOS on the carbon surface near the Fermi level. Since the electrocatalytic activity of the material is claimed to depend on DOS near the Fermi level, it increases with an increase in DOS[131]. For the ORR mechanism, it is postulated that O_2 reduction occurs by filling empty or half-filled anti-bonding orbitals of O_2 by donating electrons from the catalyst surface. Electron transfer ends up with O-O bond cleavage and H_2O_2 and/or H_2O forms. The presence of electronic states around the Fermi level introduced by doping enables the filling of anti-bonding orbitals of O_2 that are needed for reduction and lower the energy barrier for ORR[131].

1.2.2 Nitrate Reduction Reaction (NO_3RR)

Electrochemical nitrate reduction (NO_3RR) to NH_3 is a complex $8 e^-$ and $9 H^+$ transfer process with many intermediates with different adsorption and desorption characteristics[7], [133]. The reaction begins with the adsorption of NO_3^- and, consequently, $2 e^-$ reduction to the NO_2^- , commonly recognized as the rate-determining step (RDS)[20]. Afterward, suppose NO_2^- is not desorbed to the solution[99], [134], [135], [136]; in that case, it is reduced to NO and further to its possible products of NO_2 , NO, N_2O , N_2 , NH_2OH , N_2H_4 , and NH_3 (Fig 1.2), which is classified as a selectivity-determining step[137], [138]. The binding strength of the reaction intermediates differ from each other, i.e., the binding affinity of NO_3^- to transition metal surfaces is limited owing to its favorable adsorption to H_2O and its symmetrical (D_{3h}) resonant structure[139]. Furthermore, it is possible to modulate the adsorption of the intermediates by changing the electronic structure[140], [141], whereas tailoring the binding affinity of NO_3^- can become counterproductive for the adsorption of $*H$, and high negative potentials are needed to provide multiple $*H$ atoms in multiple steps of NO_3RR [119], [142]. An insufficient amount of $*H$ species for the deoxygenation step will cause a high NO_2^- concentration in the electrolyte. Also, it is essential to note that

H₂O provides the H⁺ ions necessary for NO₃RR, and H⁺ concentration is very low ($\leq 10^{-7}$ M) in alkaline media[143], [144]. At the same time, the concentration of NO₃⁻ ions is considered low due to the negative bias on the electrode[145]. The NO₃RR and HER are competing reactions and NO₃RR has many possible by-products[146], so the material should disassociate water to get protons and activate the NO₃⁻. Different reactions of the nitrate reduction with their n_e^- and thermodynamic potentials are shown below equations, along with different side-products of the reactions (Figure 1.2) [137]:

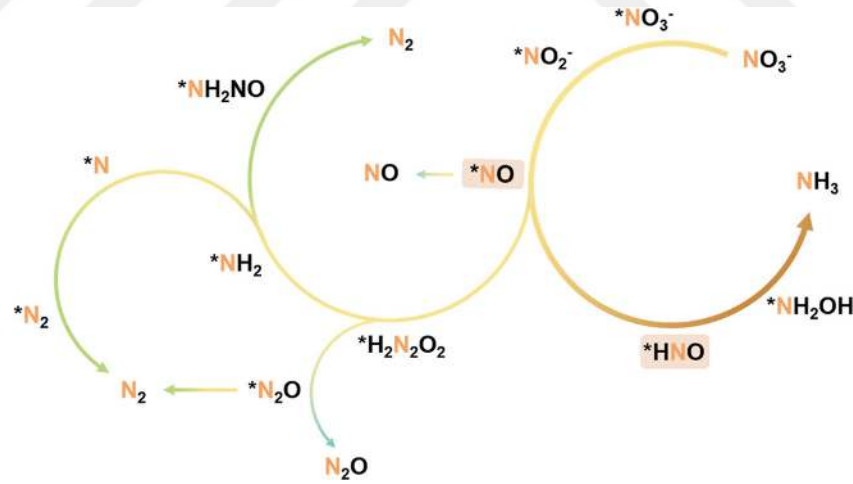
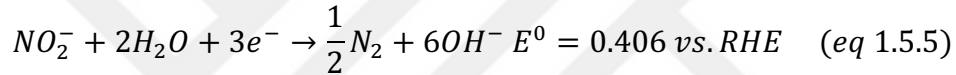
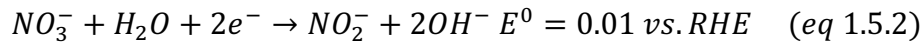
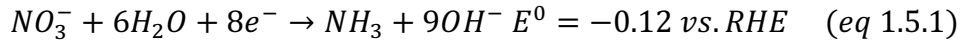


Figure 1-2: Electrochemical Nitrate Reduction (NO₃RR) Pathways

As seen in the above figure and equations, electrochemical NO₃RR is a reaction that becomes a reaction that depends heavily on selectivity if one considers more thermodynamically feasible HER at 0 V vs. RHE, too. Therefore, for NH₃ electrosynthesis, a catalyst should not be a suitable HER catalyst like MoS₂[146] while not favoring N₂ production like Pd[147]. The catalyst must be selective towards the NH₃ production pathway with intrinsic properties that favor the pathway rather than being only a catalyst that exhibits high current loads.

1.3 *Aim of the Study*

This study aims first to characterize and understand the structural, morphological, and elemental properties of Co/NS-rGO catalysts with different metal loadings and heat treatments. After identifying the material, its activity and selectivity towards ORR and NO₃RR are thoroughly investigated. Rather than only unraveling the superior activity of Co/NS-rGO catalysts towards electrochemical reactions, its electron transfer mechanism and adsorption tendencies are also examined. Electrochemical methods such as EIS, RRDE, DRT model, and Tafel analysis are studied to understand the reaction pathways. For NO₃RR, activity and the selectivity of the catalyst are investigated using both NO₃ and NO₂ species in a wide potential range. This potential range is adapted to identify the catalysts' mass-transfer limited and kinetic behaviors in different potentials. After these identifications, through XAS, EIS, DRT, and Tafel analysis, the NO₃RR mechanism of the catalyst is elucidated. For ORR, after studying the activity of the catalyst using cyclic voltammetry (CV) and RDE techniques, we analyzed the electron transfer mechanism of the catalyst using RRDE techniques. Mechanistic studies were followed by experiments to understand the origin of the ORR activity of the catalysts through electronic property analyses. The density of states, charge carrier concentration, and flatband potential of the catalyst are studied to understand the number of electronic states available for ORR around the fermi level, to understand the density of the carriers to drive the reaction and its semiconductor properties' allegiance with ORR performance respectively.

Chapter 2:

Synthesis and Experimental / Characterization Techniques**2.1 Synthesis of the GO, NS-rGO, NS-rGO 900, Co/NS-rGO, 2Co/NS-rGO, and Co/NS-rGO 900 materials****2.1.1 Synthesis of Graphene Oxide (GO)**

GO is synthesized using the improved Hummer's method[37]. The synthesis starts with the preparation of the acid mixture by stirring 180 mL of H₂SO₄ (Sigma-Aldrich, cat #339741, ACS reagent, ≥99.999%) and 20 mL of H₃PO₄ (Sigma-Aldrich, cat #695017, ACS reagent, ≥99.999%) for 15 min. Then, the prepared acid mixture is added to 1.5 g of graphite powder (Sigma-Aldrich, cat #282863, powder, <20 μm, synthetic) and stirred overnight. In the next step, 9 g of KMnO₄ (Sigma-Aldrich, cat #223468, ACS reagent, ≥99.0%) is added to the prepared solution very slowly where solution temperature is kept between 0-5 °C using an ice bath to prevent the production of any strong exotherms. Afterward, the mixture is stirred for 12 h in an oil bath while the temperature is kept at 50 °C. Later, the solution temperature is lowered to 25 °C, and 200 mL of ultrapure water is added to the mixture conditions very slowly to prevent any sudden temperature increase. Finally, in stirring conditions, 1.5 mL of 30% H₂O₂ (Supelco, cat #107298) is added to the mixture very slowly, and the solution color changes to yellowish brown. Then, the obtained solution is centrifuged (Orto Elresa Digicen 21) at 2500 rpm for 30 min to decant the supernatant. The remaining product is washed with ultrapure water, 1.0 M of HCl (Supelco, cat #HX0607, 34-37% *Omni Trace*), and ethanol three times successively; after each wash, the solution is centrifuged, and the supernatant is decanted away. Finally, the product is freeze-dried to maintain the exfoliated nature of GO. GO is preserved in DI water to protect its unique structure[37].

2.1.2 Synthesis of NS-rGO and NS-rGO catalysts

Synthesis of NS-rGO starts with preparing a dispersion of 0.3 g of GO and 1.5 g of thiourea (Sigma-Aldrich, cat #527084, ReagentPlus®, ≥99.0%) in 150 mL of ultrapure water. The mixture is then autoclaved at 190 °C for 18 h. Also, to synthesize NS-rGO

900, NS-rGO is pyrolyzed at 900 °C for 1 h in an N₂ atmosphere with a ramp rate of 5 °C/min.

2.1.3 Synthesis of Co/NS-rGO, 2Co/NS-rGO and Co/NS-rGO 900 catalysts

To synthesize DMF-protected Co precursor solution, N, N-dimethylformamide (DMF)(Sigma-Aldrich, cat #319937, ACS reagent, ≥99.8%) is heated up to 140 °C, and 1 mL of 0.1 M CoCl₂ (Sigma-Aldrich, cat #255599, ACS reagent, ≥98.0%) solution is added to DMF at 140 °C in vigorous stirring conditions. Then, the mixture is refluxed for 12 h in an oil bath while maintaining its temperature at 140 °C. After 12 h, the refluxed solution is halved using a rotary evaporator to remove excess DMF. Co/NS-rGO synthesis begins with dispersing 5 mg of NS-rGO in 5 mL of DMF to prepare 1 M NS-rGO/DMF solution. Afterward, a mixture of 5 mL of DMF-protected Co precursor solution and 1 M NS-rGO/DMF solution is sonicated for 30 minutes. Then, the mixture is dried at low pressure at 80 °C for 8 h using a rotary evaporator, and finally, Co/NS-rGO is obtained. 2Co/NS-rGO is prepared using the same protocol except for adding a two-fold volume of DMF-protected Co precursor solution used to synthesize Co/NS-rGO. Furthermore, for the synthesis of Co/NS-rGO 900, Co/NS-rGO is pyrolyzed at 900 °C for 1 h in an N₂ atmosphere with a ramp rate of 5 °C/min.

2.2 Structural Characterizations

2.2.1 X-ray diffraction (XRD)

XRD is a structural characterization technique that adopts the principle of inspecting a material's elastic scattering capabilities excited by X-ray photons. The analysis of XRD data is often qualified and interpreted through the XRD patterns that the material exhibits through the constructive interference of X-ray photons excited by the material characteristics[148]. Crystallographic properties of the materials, such as lattice spacing and crystallite size, can be unraveled using XRD analyses. Lattice spacing provides information about the orientation of crystalline planes concerning each other, i.e., a specific plane being compact and reduced through lower lattice spacing or the exfoliated nature of the plane through higher lattice spacing. Crystallite size enables us to analyze the size of a crystallite or a particle assigned for that specific plane, such as metal oxides or carbon planes. Also, to identify the different planes of XRD patterns, one should

utilize the Powder Diffraction File (PDF) database and analyze the compatibility of one's XRD pattern with PDF results. Essential information of XRD can be listed as:

$$\text{Bragg's Law: } 2d \sin\theta = n\lambda \quad (\text{eq. 2.2.1})$$

$$\text{Scherrer Equation: } D = n\lambda\beta\cos\theta \quad (\text{eq. 2.2.2})$$

λ : X-ray wavelength; d : lattice spacing; θ = angle between the X-ray and the plane;
 β : Peak width half maximum.

2.2.2 Raman Spectroscopy

Raman spectroscopy is a method based on the concept of elastic scattering of monochromated photons that induces inelastic scattering on the material, also known as Raman scattering. Regarding this study, Raman spectroscopy can be used to determine the number of graphene planes using the following equation[149], [150]:

$$\omega_G(\text{G} - \text{peak wavelength number})(\text{cm}^{-1}) = 1581.6 + \frac{11}{1 + n^{1.6}} \quad (\text{eq. 2.2.3})$$

$$\omega_G = (\text{G} - \text{peak wavelength number})(\text{cm}^{-1}); n = \text{number of graphene planes}$$

2.2.3 X-ray Photoelectron Spectroscopy (XPS)

XPS is a surface-sensitive method that extracts essential information about the elemental composition, elemental bondings within the orbital, and the composition of the surface species on the material's surface. The XPS technique is based on Einstein's photoelectric effect concept, where the excitation of a photoelectron through interaction with incident photons and photoelectron attain kinetic energy associated with its binding energy[151], [152]. The equation for the photoelectric effect that XPS inherits from its foundations is:

$$E_k = h\nu - E_b - \varphi \quad (\text{eq. 2.2.4})$$

E_k : kinetic energy of the photoelectron; h : Planck's constant; E_b = Binding energy of the photoelectron ; φ = Work function loss between a sample and the device.

2.2.4 X-ray Absorption Spectroscopy (XAS)

XAS is a soft X-ray spectroscopy technique that provides information about the material's electronic and local geometric structure. In XAS, the photon energy is set so that the core electrons are excited to an unoccupied state where the significant

excitations of electrons introduce a region called an edge. Absorption edges of K, L, and M correspond to the principal quantum numbers of $n=1, 2$, and 3 . XAS spectra contain two different regions emerging from excitations to the lowest occupied state: the main edge and rising edge, where the former is for transitions from the Fermi level and the latter is for transitions induced by photon energies lower than the lowest unoccupied states[153]. With higher photon energies, two different spectral of X-ray Absorption Near-Edge Structure (XANES) and Extended X-ray Absorption Fine Structure (EXAFS) can be studied, providing valuable insights about electronic structures through symmetry analyses for the former and atomic distance and coordination environment for the latter.

Synchrotron facilitates that have an XAS device

XAS results can be quantified using Beer's Law:

$$I_t = I_0 e^{-\mu(E)t} \quad (2.2.5)$$

I_t : X-ray intensity by transmission; I_0 : Incident X-ray intensity; t : sample thickness; $\mu(E)$: absorption coefficient

2.3 Electrochemical Tests

2.3.1 Linear Sweep Voltammetry (LSV)

LSV is an electrochemical method where a specific potential range concerning a reference potential, i.e., vs. RHE vs. Ag/AgCl, is scanned with a selected scan rate using a potentiostat device, a two or three-electrode cell with respective acidic, basic, or neutral electrolytes with or without gas saturation. LSV results can be used to analyze reaction kinetics, current output, or electrode durability. LSV is a mass-transfer limited technique[154].

2.3.2 Cyclic Voltammetry (CV)

CV and LSV do not differ in how the experiment is performed. Unlike LSV, CV measurements are two-way measurements, meaning that a consecutive cathodic and anodic or anodic or cathodic scan direction is used. CV can be used to analyze the reversibility of redox reactions effectively by analyzing specific behaviors in given potentials after numerous cathodic and anodic cycles. CV measurements can also

introduce essential information about the charge storage properties of the material through electrochemical surface area (ECSA) and double-layer capacitance (C_{dl}) analyses. Furthermore, CV can be a vital electrochemical tool for analyzing the durability of materials and batteries through hours- or days-long cyclic experiments[154].

2.3.3 Rotating Disk Electrode (RDE)

RDE is a hydrodynamic electrochemical method that evaluates the performance of an electrode in a fixed potential range with a given scan rate as of LSV while eliminating mass-transfer limitations through the forced convection driven by rotation at a given rate. RDE plots give information about three different regions: (1) kinetic region, (2) mixed region, and (3) mass-transfer limited region. In the kinetic region, one can identify the efficiency of the reaction concerning side reactions through electron transfer number analysis[155]. By looking at the slope at the beginning of the mixed region, one can measure the material's Tafel slope, which shows how fast the reaction in the specified region is. Also, the mass-transfer limited region presents information about the mass-transfer limiting current of the catalyst and elucidates Levich analysis. Levich analysis equation is given below:

$$\text{Eq. 2: } i_l = 0.62nFAD^{\frac{2}{3}}\nu^{-\frac{1}{6}}c_0\omega^{\frac{1}{2}} \quad (\text{eq. 2.2.6})$$

i_l : limiting current; n number of electrons transferred; A : area of the electrode

F : Faradaic Constant; D : diffusion coefficient; ν : kinematic viscosity of the electrolyte

c_0 : solubility of O_2 in electrolyte; ω : rotation rate

2.3.4 Rotating Ring Disk Electrode (RRDE)

The RRDE technique utilizes the same method regarding the experiment, whereas the electrode design differs significantly from RDE's. In RDE, a glassy carbon working electrode is used, while in RRDE, a Pt or Gold ring is used to oxidize the material to the desired potential. By this method, one can measure the decomposition rate of specific

species, i.e., H_2O_2 , or determine the number of electrons transferred in a wide potential range [122], [156].

For RRDE measurements, first, a collection efficiency must be obtained using a potassium ferricyanide solution. The equation for collection efficiency ($N_{empirical}$) is given below:

$$N_{empirical} = \left| \frac{i_{L,RING}}{i_{L,DISK}} \left(\frac{n_D}{n_R} \right) \right| \quad (eq. 2.2.7)$$

$N_{empirical}$: collection efficiency; $i_{L,RING}$: disk current; $i_{D,RING}$: ring current; n_D : number of electrons exchanged at the disk; n_R : number of electrons exchanged at the ring where $n_D = n_R$

The number of electrons transferred during ORR is calculated using the equation below:

$$n_e^- = \frac{4|i_{L,DISK}|/N_{empirical}}{i_{L,DISK} + i_{L,RING}/N} \quad (eq. 2.2.8)$$

The selectivity of the peroxide species is found as shown below:

$$H_2O_2\% = 2 \times \frac{i_{L,RING}/N_{empirical}}{i_{L,DISK} + i_{L,RING}/N} \times 100 \quad (eq. 2.2.8)$$

2.3.5 Chronoamperometry (CA) and NH_3 , NO_2 , and H_2 quantification using the Indophenol Blue Method, Griess' Test, and Gas Chromatography

CA measurements are carried out by applying a desired constant potential for a given duration. NH_3 , NO_2 , and H_2 quantification are carried out using the equations below:

$$F \cdot E_{NH_3} = \frac{8 \times F \times C_{NH_3} \times V}{(MW_{NH_3} \times Q)} \quad (eq. 2.2.9)$$

$$F \cdot E_{NO_2^-} = \frac{2 \times F \times C_{NO_2^-} \times V}{(MW_{NO_2^-} \times Q)} \quad (eq. 2.2.10)$$

$$F \cdot E_{H_2} = (n_{H_2 flow} \times n_e \times F \times n \times V_{H_2 flow}) \quad (eq. 2.2.11)$$

$$r_{NH_3, NO_2^-, H_2} = \frac{C_{NH_3, NO_2^-, H_2} \times V}{t \times m_{cat}} \quad (eq. 2.2.12)$$

F : Faradaic constant; C_{NH_3} : NH_3 concentration; V : volume of the catholyte; MW_{NH_3} : molecular weight of NH_3 ; Q : amount of charge accumulated; C_{NH_3} : NH_3 concentration; $C_{NO_2^-}$: NO_2 concentration; $MW_{NO_2^-}$: molecular weight of NO_2 ; $n_{H_2 flow}$:

H₂ molar flow rate; n_e: number of electrons transferred; t: duration of CA; m_{cat}: the mass of catalyst loading

2.3.6 Electrochemical Analysis

2.3.6.1 Electrochemical Surface Area (ECSA)

ECSA analysis begins with a determination of double-layer capacitance (C_{dl}) by taking CV measurements in a specified non-Faradaic potential range with different scan rates and taking the slope of the graph between half of the difference between anodic and cathodic current at the middle potential of CV range[157]. Then, ECSA is calculated as follows:

$$ECSA = \frac{C_{dl}}{C_s} \quad (eq. 2.2.13)$$

C_{dl}: Double layer capacitance; C_s = Specific Capacitance = 0.022 mF/cm²

2.3.6.2 Electrochemical Impedance Spectroscopy (EIS)

The EIS technique uses disturbances induced by applied potentials in electrochemical systems through sinusoidal waves and an analysis of the input's sinusoidal responses. EIS can be used to determine the material's charge transfer resistance (r_{ct}), perform capacitive analyses, or investigate the material's electronic properties[158]. Total layer capacitance and space-charge capacitance of the material can be found using EIS techniques as such:

$$C_{tot} = -\frac{1}{2\pi f Z_{img}} \quad (eq. 2.2.14)$$

C_{tot}: Total double-layer capacitance; f: frequency; Z_{img}: the imaginary part of the Nyquist plot; C_{SC}: space-charge capacitance

$$Eq. 5: C_{tot,ECSA} = \frac{C_{tot}}{ECSA} \quad (eq. 2.2.15)$$

$$Eq. 6: \frac{1}{C_{tot,ECSA}} = \frac{1}{C_H} + \frac{1}{C_{SC}} \quad (eq. 2.2.16)$$

$$C_H = \text{Helmholtz capacitance} = 0.02 \text{ mF/cm}^2$$

Furthermore, the density of states of a material ($D(E_F)$), charge carrier density (N_D), and flatband potential (V_{fb}) can be measured using EIS techniques by taking an impedance measurement around the onset potential of the catalyst towards the reaction of interest, then carrying out the following algebra[159]:

$$D(E_f) = \frac{C_{sc}^2}{\varepsilon \varepsilon_0 e_0^2}$$

C_{sc} = Space – charge capacitance; ε = Dielectric constant = 3.3

$$\varepsilon_0 = \text{Vacuum permittivity} = 8.854 \times 10^{-12} \frac{F}{m}$$

$$e_0 = \text{Electronic charge} = 1.602 \times 10^{-19}$$

$$\text{Eq. 8: } D(E_f) = \frac{C_{sc}^2}{\varepsilon \varepsilon_0 e_0^2}$$

$$\text{Eq. 9: } N_D = \frac{2}{\varepsilon \varepsilon_0 e_0 ECSA^2 MS_{slope}}$$

$$\text{Eq. 10: } \frac{1}{C^2} = \frac{2}{\varepsilon \varepsilon_0 e_0 ECSA^2 N_D} \left(V - \left(V_{fb} + \frac{kT}{e} \right) \right)$$

MS_{slope} : Mott-Schottky slope; k: Boltzmann constant; T: absolute temperature

Chapter 3:

Elucidation of Activity and Selectivity of Co/NS-rGO Catalysts Towards ORR Using In-Depth Kinetic and Electronic Property Analysis

3.1 Introduction

3.1.1 Importance of ORR for Sustainable Energy Future

To reduce global fossil fuel dependence, robust and cost-effective renewable energy technologies should be widely used to meet increasing energy demand[2]. Proton exchange membrane fuel cells (PEMFCs) hold great importance because they compensate for the bottlenecks of recently emerging lithium-ion batteries and further enable full electrification of vehicle transportation[14], [160], [161], [162], [163]. It is widely known that PEMFC activity is primarily limited by sluggish oxygen reduction reaction (ORR) kinetics at the cathode despite the high loading of active expensive platinum group metal (PGM) electrocatalysts, i.e., Pt, at both cathode and anode[164], [165]. Scarcity and the high price of Pt are some of the most prominent issues facing the large-scale commercialization of PEMFCs[72], [79], [129], [166]. It is proposed that for economically advantageous PGM-free materials to replace Pt, they must have optimum durability, many available active sites[167], turnover frequency (intrinsic activity)[146], mass transport, and charge/electron transfer properties[168], [169]. Also, PGM-free catalysts need to avoid degradation during hours of continuous PEMFC operation[170]. Over the years, intensive effort has been put into synthesizing either low-PGM catalysts with alloying[24], [168], [171], [172], [173], [174], [175], atomically dispersed PGM catalysts[176], core-shell structures[23], [24], [177], [178], PGM-free catalysts, N-doped carbon materials[47], [63], atomically dispersed metal nitrogen carbon catalysts[62], [179], [180], [181], [182], [183], [184], [185], [186], [187] and heteroatom-doped metal catalysts[51], [188], [189].

PGM catalysts have activity setbacks such as demetallation, degradation, aggregation[190], sulfur deactivation[191], and the inability of superoxide protonation in fuel cell conditions[192]. High surface area metal nitrogen carbon (M-N-C) catalysts are proposed to overcome these setbacks as alternative ORR catalysts[186]. Previously, Fe-

N-C catalysts were proposed as an alternative to PGM catalysts due to their high stability and activity[179], [182], [187], [193]. Wu et al. synthesized PANI-Fe-C catalyst using ketjen black and polyaniline (PANI) as the carbon and nitrogen sources mixed with iron salts and then subjected to heat treatment[179]. Catalyst has shown 0.3 V of overpotential, 99% 4e⁻ selectivity, and durability of 700 hours at 0.4 V, a relatively low potential for PEMFC operations. However, like other Fe-N-C catalysts[194], PANI-Fe-C catalyst suffered from significant performance loss at 0.6 V, which can be due to the amorphous nature of the material[179] and demetallation during fuel cell operation in an acidic environment[194], [195], [196]. It is claimed that using Co instead of Fe for the catalyst would make ORR less dominated by Fenton reactions detrimental to PEMFC activity.

3.1.2 Aims of Study

Herein, we develop Co/NS-rGO and 2Co/NS-rGO catalysts with 1:2 Co content, CoS presence, and Co SA(single-atom) distribution along the graphene plane. The effect of Co amount in both catalysts is investigated in terms of overpotential R_{ct} (charge-transfer resistance), ECSA, current density, and capacitance values. Furthermore, the electronic properties of the catalysts are examined using EIS techniques, including the semiconductor properties of the material, flatband potential, density of states (DOS), and carrier concentration. The synergy between CoS, Co-N_x moieties, and Co SAs is studied to understand the origin of the superior ORR activity. Also, the benefits of using S as a co-dopant are studied to unravel its intervention in forming ORR-active structures, electronic properties, and superior kinetic assets.

3.2 Experimental Methods

3.2.1 Structural Characterizations

200 kV aberration-corrected FE-TEM/STEM Hitachi HF5000 equipment is used to perform transmission electron microscopy (TEM) images, where a 20 kV Zeiss Ultra Plus microscope is utilized to obtain the scanning electron microscopy (SEM) images. Bruker D2 Phaser X-ray Diffractometer records X-ray diffraction (XRD) results for the powder samples with Cu K α radiation having potential and current of 10 kV and 30 mA, respectively. The Renishaw INVIA instrument was used for Raman spectroscopy, performed at 532 nm of laser excitation wavelength. X-ray photoelectron spectroscopy

(XPS) was performed using a Thermo K-alpha with monochromatized Al K α (1486.7 eV) radiation.

3.2.2 Electrochemical Screenings

The electrochemical characterizations of the rotating disk electrode (RDE) and rotating ring disk electrode (RRDE) were carried out using a Pine Research WaveDriver 40 DC Biopotentiostat in an N₂ or O₂-purged 0.1 M KOH electrolyte with standard three-electrode cell configuration. H₂O⁻ selectivity measurements were performed using a rotating ring disk electrode (RRDE) with Pt ring, and a collection efficiency value of 27% is determined in potassium ferrocyanide (K₃[Fe(CN)₆]) (Sigma-Aldrich, cat #244023, ACS reagent, $\geq 99.0\%$) solution considering red-ox reaction between Fe(CN)₆³⁻ and Fe(CN)₆⁴⁻ species. The three-electrode cell configuration comprised a graphite rod or platinum coil used as a counter electrode, an Ag/AgCl reference electrode calibrated by a standard hydrogen electrode, and a coated glassy carbon electrode as the working electrode. The working electrodes were prepared by drop-casting the as-prepared ink onto the glassy carbon electrodes. The catalyst inks consisted of 5 mg of the regarding catalyst, 120 μ l of pure ethanol (Sigma-Aldrich, cat #32221-M, absolute, $\geq 99.8\%$), and 5 μ l NafionTM perfluorinated resin solution (Sigma-Aldrich, cat #510211). The mass loading of catalysts on the glassy carbon electrode was set to 0.204 mg cm⁻². The iR compensation for all electrochemical measurements was performed at 85% and 100 kHz using either a Biologic SP-200 or a VSP-300 Potentiostat. All the other electrochemical impedance spectroscopy (EIS) and voltammetric measurements are carried out using a Biologic VSP-300 Potentiostat. The double-layer capacitance (C_{dl}) values were calculated by taking half of the slope for the anodic and cathodic current density differences ((j_a-j_c/2)) versus the potential scan rate (mV/s) plot. Then, C_{dl} is divided by specific capacity (C_s) of 0.022 mF/cm² to find the electrochemical surface area (ECSA). Tafel analysis was carried out by plotting potential versus log (j) in the kinetic region.

3.3 Characterizations Of The Catalysts Using Imaging, Spectroscopy And Crystallographic Techniques

As described in the experimental methods part, first, NS-rGO is synthesized by autoclaving 1:5 %w/w thiourea, GO aqueous solution for 18 h at 190 °C. Then, cobalt

loading is carried out by intercalating Co/DMF precursor to NS-rGO/DMF dispersion at 80 °C for 12 h. We pursued a low-pressure, low-temperature, and long-duration approach to preserve graphene's defect-rich, porous, and layered structure (Fig. 3-1)[197]. Thiourea, as a doping reagent, is claimed to facilitate the interaction between GO and N/S due to its ability to provide the heteroatoms to the graphene plane simultaneously[76], [198]. For Co/NS-rGO, an excess amount of cobalt incorporation into the NS-rGO matrix is avoided; hence, the formation of neither large chunks of nanoparticles nor clusters is observed, which can be seen in high-resolution scanning transmission electron microscopy (HR-STEM) results (Fig. 3-3 a-d). Determination of the optimum Co-to-NS-rGO ratio has enabled the anchoring of many cobalt SAs and actualized even distribution of Co, N, and S atoms throughout the NS-rGO matrix, which is shown in EDX mapping (Fig. 3-2). It is speculated that during the incorporation of Co, the presence of S prevented the formation of CoO_x by facilitating Co-S interactions to form CoS.

The HR-STEM image (Fig. 3-3a) shows that we could preserve the layered, exfoliated, and porous nature of NS-rGO and obtain a wrinkle-like structure intrinsic to doped rGO materials[199]. Co SAs, which are homogeneously distributed throughout the graphene material, are claimed to boost ORR activity (Fig. 3-3 b, c, d). Also, HR-STEM images for 2Co/NS-rGO indicate the abundance of Co_xO_y species in the material (Fig. 3-3 e, f, g, h).

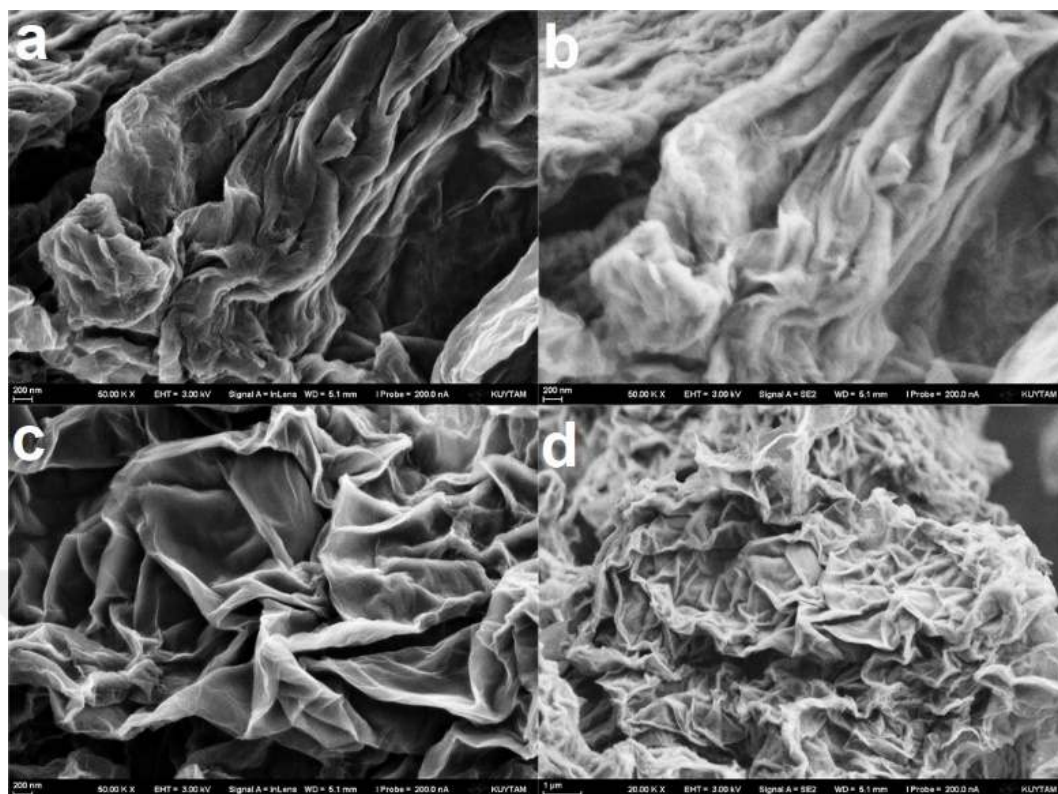


Figure 3-1 Scanning Electron Microscopy (SEM) images for NS-rGO

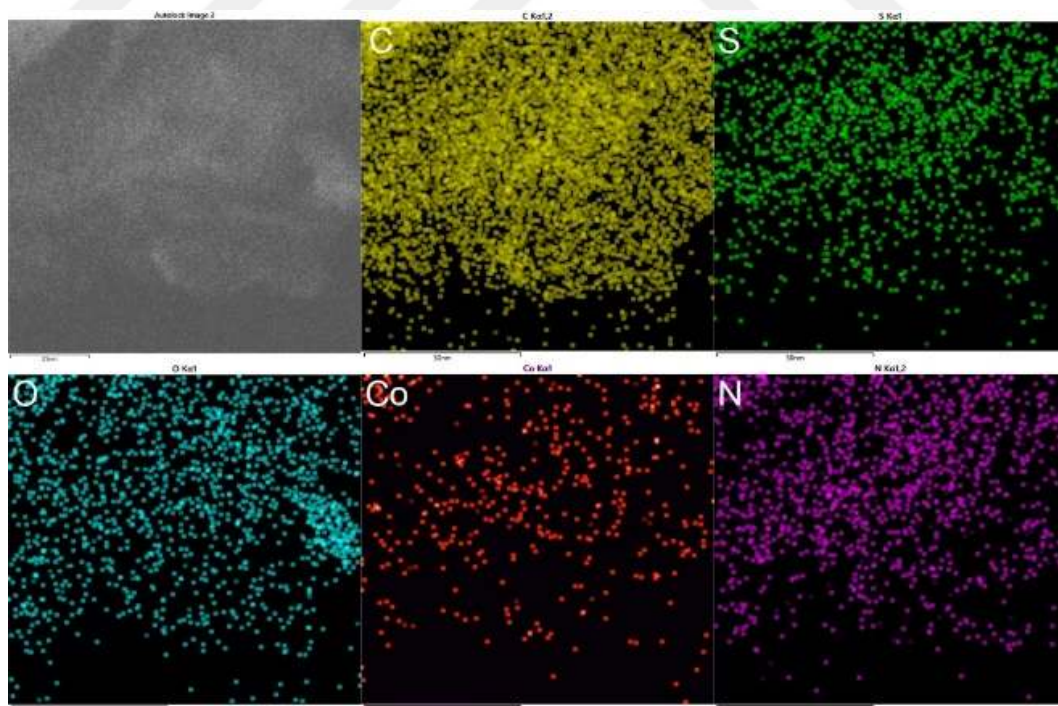


Figure 3-2 HR-STEM EDX Mapping for Co/NS-rGO

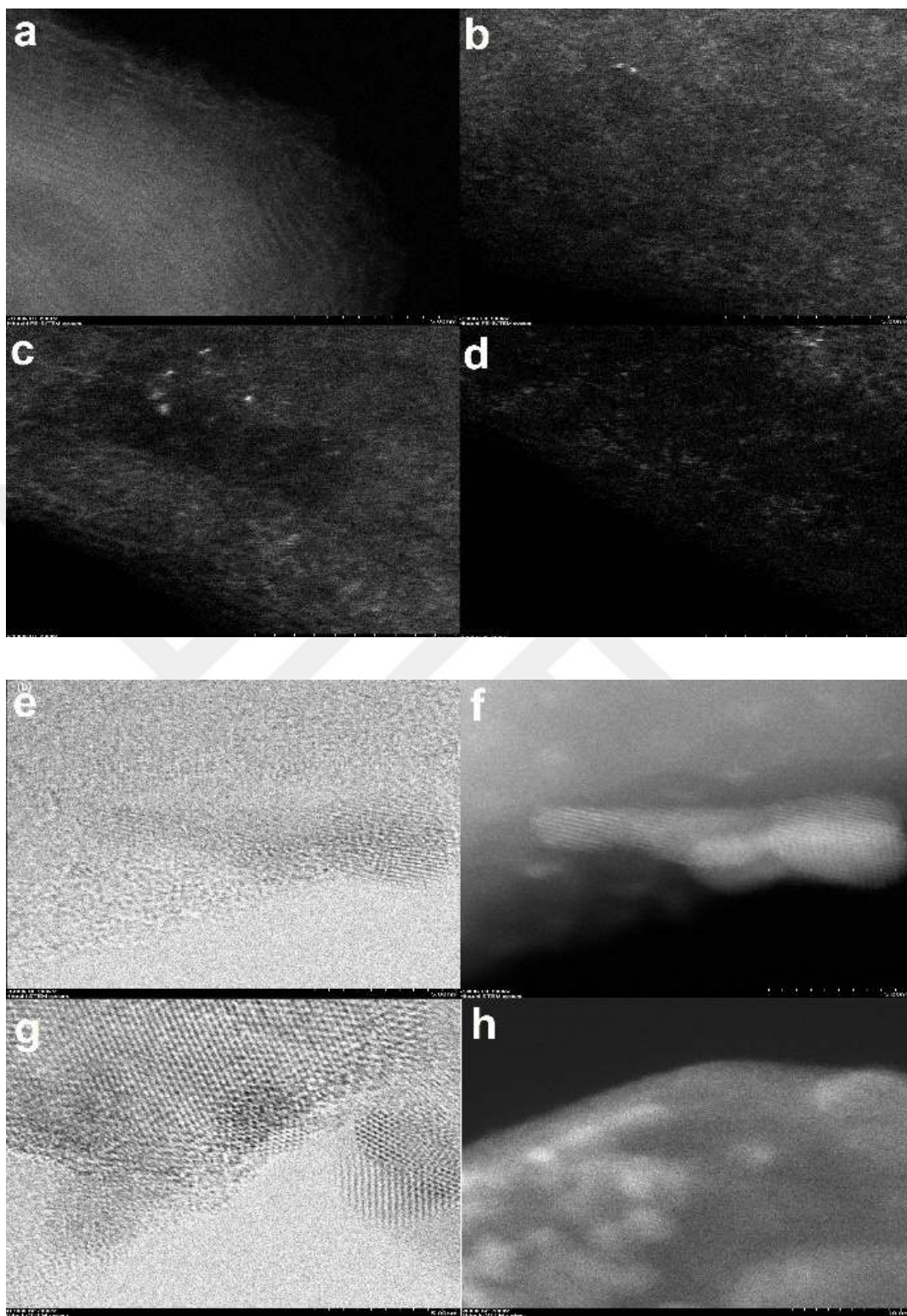
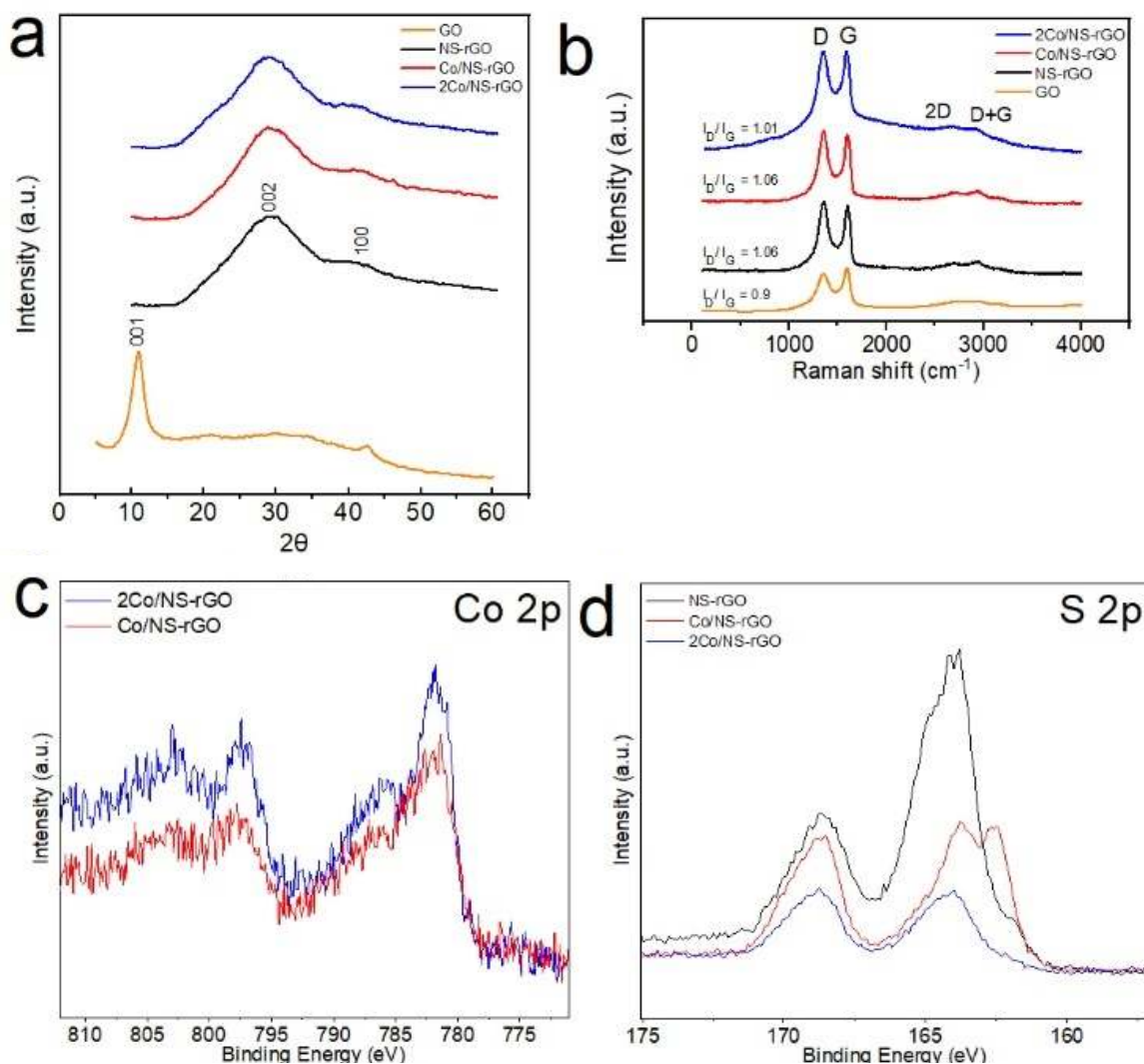


Figure 3-3. HR-STEM images. a, b, c, d) Co/NS-rGO. e, f, g, h) 2Co/NS-rGO

EDX analysis (Fig. 3-2) demonstrates the chemical bonding nature of Co SAs distributed over the region with a similar S, N, and O species distribution throughout the graphene plane. Interestingly, despite the relatively high amount of oxygen in the material, there is

no visible formation of cobalt oxide nanoparticles for Co/NS-rGO. Still, cobalt oxide formation is commonly observed in oxygen-containing cobalt catalysts[44], [59] and is present for 2Co/NS-rGO. X-ray photoelectron spectroscopy (XPS) S 2p (Fig. 3-4d) spectra show that S favors sulfate formation for Co/NS-rGO and suppresses cobalt oxide formation, possibly due to lower formation energy of SO_4^{2-} while the 2Co/NS-rGO exhibits low SO_4^{2-} formation, indicating the possibility of cobalt oxide presence. The presence of abundant, homogenously distributed single atoms in Co/NS-rGO (Fig. 3-2) and the simultaneous occurrence of CoS particles (Fig. 3-4d) is speculated to be the cause of the catalyst's superior ORR performance. CHNS results show that S and N content is 17% (Fig. 3-5b) and 9% in the catalyst Co content is found as 1.8% for Co/NS-rGO and 5.11% for 2Co/NS-rGO using ICP-MS (Fig. 3-5c)



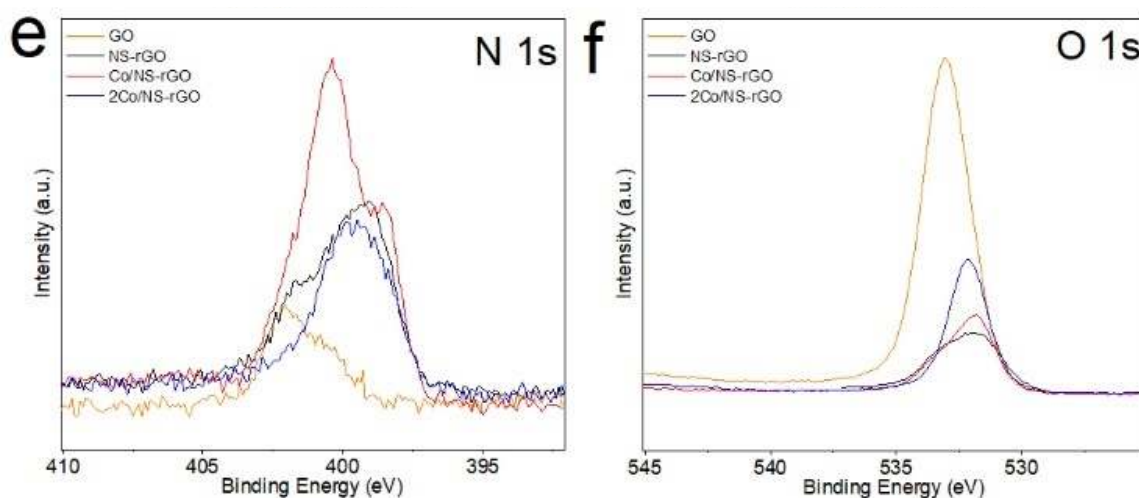


Figure 3-4 Structural Characterizations. a) XRD. b) Raman Spectra. c) XPS Co 2p. d) XPS S 2p. e) XPS N 1s. f) XPS O 1s.

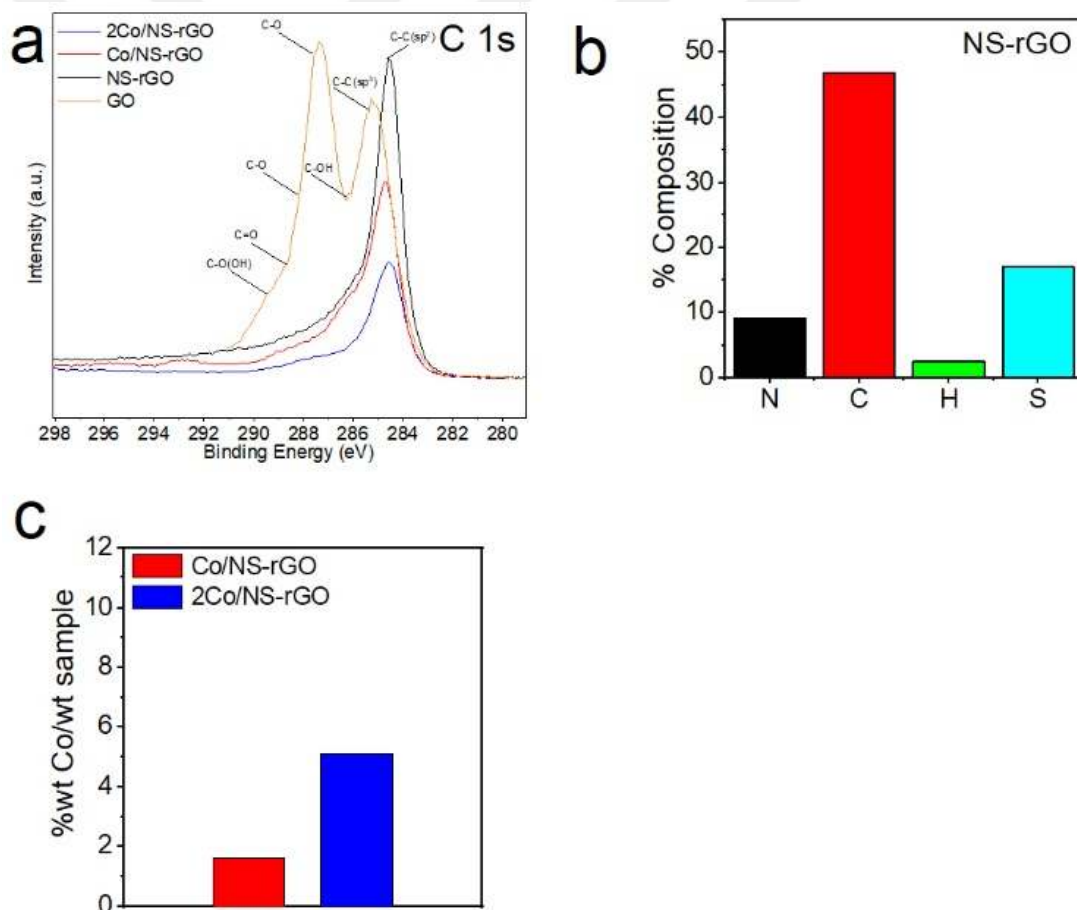


Figure 3-5 Structural Characterizations. a) XPS C 1s. b) CHNS for NS-rGO. c) ICP-MS for Co/NS-rGO and 2Co/NS-rGO

X-ray diffraction (XRD) patterns (Fig. 3-4a) show the emergence of a sharp peak at $2\theta = 10.9^\circ$ for GO (001), suggesting successful oxidation of graphite (characteristic peak at $2\theta = 26^\circ$). GO XRD reflections indicate that a significant fraction of graphite to GO

conversion has succeeded; however, a small peak at $2\theta = 43^\circ$ attributed to a graphite impurity is still visible[200], [201]. The peak assigned to the (001) plane in GO appears in NS-rGO and Co/NS-rGO (Fig. 3-4a), manifesting a substantial reduction for both materials[202]. Crystalline planes of (002) and (100) present in graphene-like carbon materials are aligned at $2\theta = 28.8^\circ$ and 41.5° values, respectively[76], [92], [203]. The reflections from (002) and (100) planes are deviated to $2\theta = 26^\circ$ and 43° . This shift is possibly due to the presence of oxygen in the sample and convolution due to the presence of CoS species[76]. Weak intensity and broad peaks at crystalline planes of carbon (002) and (100) in both Co/NS-rGO and NS-rGO show the formation of loosely stacked graphene layers that are different from graphite, which has narrow and sharp peaks[92], [204]. Nearly the same XRD pattern of Co/NS-rGO and NS-rGO and the same d-spacing values (d_{002}) of the carbon planes suggests that the Co incorporation did not cause a change in crystalline properties of NS-rGO in large size and further actualized formation of CoS moieties.

Raman spectra (Fig. 3-4b) show carbon's disorder-induced D and G bands at 1354 cm^{-1} and 1598 cm^{-1} , respectively[205]. Higher intensity ratio for D and G bands (I_D/I_G) of Co/NS-rGO and NS-rGO (1.06) compared to GO (0.9) proves the reduction of GO, incorporation of N and S to graphene lattice, and increase in defect density induced by the two atoms for Co/NS-rGO and NS-rGO[95], [203] while the 2Co/NS-rGO lacks defect-rich properties possibly linked to its inferior activity. I_D/I_G ratio for NS-rGO (1.06) does not change after cobalt incorporation, which emphasizes the inability of Co to induce carbon-related defects. It is proposed that the G band in Raman spectra arises from the presence of sp^2 rings in the material, and G-peak intensity increases with increased doping levels; we verify the presence of sp^2 rings in each sample by existing G band and successful doping with the increased G-peak intensity[206]. For the analysis, we used intensity ratios (I_D/I_G) rather than area (A_D/A_G) ratios due to the high disorder of the materials in this study[149]. Therefore, we did not carry out any peak deconvolutions deriving from the area of each band and stuck to intensity values in Raman analysis. Also, the number of graphene layers is calculated using G peak position values[207], and it is found to be approximately 4 in all three of the samples, which verified the preserved nature of layered graphene[207]. The 2D peak claimed to be affected by doping levels, and the number of graphene layers is present at $2678\text{ (cm}^{-1})$ for all three samples[206],

[208]. Considering peak intensity at D + G combination mode (2950 cm^{-1}), induced by disorder, CoNS-rGO and NS-rGO have the same and higher disorder than GO, which is the observed pattern throughout the size of the $\sqrt{\frac{I_D}{I_G}}$ value is proportional to the average size of sp^2 ring clusters (L_A) ($L_a(\text{nm}) = \frac{560}{E_{\text{laser}}^4} (\frac{I_G}{I_D})$). L_A for NS-rGO does not change after cobalt incorporation, which lets us suggest that either pyridinic and graphitic N (Fig. 3-4e) substitutions to sp^2 hybridized graphene do not cause a significant size change due to similar size of C/N or low pyridinic, graphitic N / total N ratio or binding of nitrogen to deoxygenated sites to form pyrrolic N sp^3 hybridization is the reason for keeping the of sp^2 species with the same size [65], [205]. Finally, Raman spectroscopy results (Fig. 3-4b) show that C-(S-S)_n bonds exist in NS-rGO, verified by the corresponding peak at 660 cm^{-1} [93]. In contrast, in Co/NS-rGO, the regarding peak does not exist, which allows us to suggest that C-S bonds in NS-rGO are broken and the remaining sulfur atoms are either bound to oxygen to form sulfate or bound to Co to form atomic scale CoS moieties.

Co 2p spectra (Fig. 3-4c) results also show a higher amount of Co is present on the material's surface, which agrees with ICP-MS measurements for bulk. Co 2p peaks characteristic of CoO are present for 2Co/NS-rGO, which supports our findings in HR-STEM.

In S 2p spectra (Fig. 3-4d), peaks of 168.7, 165.4, 163.8, and 162.4 eV are attributed to sulfate or thiosulfate (SO_4^{2-} , $\text{S}_2\text{O}_3^{2-}$) thiolic S (C-S-H), thiophenic-S (C-S-C), and S_n^{2-} (surface) respectively for both sulfur-containing samples[78], [94], [209], [210]. Shift to lower binding energies after incorporation of cobalt suggests the emergence of new functional groups and species[211]. Interestingly, in Co/NS-rGO, a single peak at 162.4 eV appears, and its emergence is derived from Co-S bonding[209]. It is claimed that S_n^{2-} (surface) species react first, and its corresponding CoS formation is achieved through the breaking up of the S-S bond, i.e., the peak intensity at 165 eV decreases and stabilization of S^- through electron transfer from Co to S_n^{2-} (surface) [212]. Also, peaks at 168.7 to 170.8 eV are assigned to sulfate species formed with exposure to air[213]. The most apparent difference between NS-rGO and Co/NS-rGO in S 2p spectra is the decrease in peak intensity of thiophenic (C-S-C) and thiolic S (C-S-C) with Co incorporation. This result agrees with Raman's results, where C-S bonding is only observed for NS-rGO.

These results support the hypothesis that C-S bonds are broken down with Co introduction to the catalyst, and the remaining sulfur is bound to Co to form CoS. Also, thiophenic-S (C-S-C), along with pyrrolic and pyridinic N species, is claimed to be another factor indicating that N and S doping on graphene is carried out effectively[94].

The peaks in XPS N 1s (Fig. 3-4e) at 398.4, 399, 400.4, and 401.8 eV are assigned to pyridinic N, amide N, pyrrolic N, and quaternary N[64], [214]. Along with CoS formation, the lower amount of Co incorporation enables the formation of pyridinic N, contributing to the superior ORR activity of Co/NS-rGO. Amide N formation for NS-rGO originated from heating thiourea and sulfuric acid at 190 °C. Also, the peak at 400.5 eV for pyrrolic N emerges for Co/NS-rGO. However, pyrrolic N is claimed not to be a ORR-active species since its presence does not affect the carrier concentration of graphene[63], [215]. Furthermore, the peak intensity for quaternary N is similar for both GO and NS-rGO, while Co/NS-rGO has a higher and 2Co/NS-rGO has a lower quaternary N content. It is claimed that the oxygen content in Co/NS-rGO is bound to N-species and suppresses CoO formation. At the same time, the low concentration of quaternary N for 2Co/NS-rGO could have promoted the CoO formation for the catalyst. Also, O 1s spectra (Fig. 3-4f) show that after the hydrothermal treatment, oxygen content in the material decreases while 2Co/NS-rGO obtains high oxygen content, which could be attributed to CoO formation.

XPS C 1s spectra (Fig. 3-5a) of GO show different high-intensity peaks starting from 286.2 eV to 290 eV that are attributed to C-OH, C-O, C=O, and C-O(OH) species, which shows the abundance of oxygen and oxygenated groups in GO. However, peaks of these oxygenated groups in NS-rGO, Co/NS-rGO, and 2Co/NS-rGO samples shift to lower binding energies but do not disappear after reduction, which shows the partial reduction and presence of pre-doped oxygenated species in all samples. We observe broader and more asymmetric C 1s peak of sp^2 C towards higher binding energies from NS-rGO to Co/NS-rGO due to increasing functional groups with Co introduction, such as CoS. The graphene-like structure is achieved, which is observed considering the presence of sp^2 hybridization in C-C bonds. We see a clear reduction in peak intensity of C-O / C-OH bonds after reduction to NS-rGO. We see that the peak for C=C sp^2 bonds obtain high intensity in NS-rGO, whereas after Co incorporation, regarding double bond peak intensity decreases and the relative intensity of single-bonded carbon increases, which is

most probably due to breaking of the double bond, functionalization of the surface species with their interaction with Co and alignment of pyridinic, pyrrolic and quaternary N. However, according to Raman Spectra analysis, we should also note that the size of sp^2 ring clusters stays the same after cobalt incorporation. This enables us to propose that the alignment of pyridinic and graphitic N does not cause a size change in graphene. Also, the significant increase in pyrrolic N content for Co/NS-rGO in N 1s spectra (Fig. 3-4e) causes us to suggest that the size of sp^2 hybridized ring clusters does not change since N species do not preferentially locate through sp^2 hybridization. Still, nitrogen may bind to edge sites of graphene where -COOH and -OH groups are left, and pyrrolic N is formed, which claims the fixed size of sp^2 ring clusters.

3.4 Electrochemical Activity and Selectivity Analysis: Unraveling Kinetic and Electronic Properties of the Catalyst

Cyclic voltammetry (CV) measurements in N_2 and O_2 -purged electrolytes show the presence of two peaks at 0.4 V and 1.1 V, which can be assigned to CoS or quinone red-ox peaks[179], [216]. However, these peaks' irreversibility eliminates the possibility of quinone presence and allows us to assign them to CoS peaks. Bănică et al. showed the presence of CoS by showing the reduction in labile CoS peaks in time[216]. After several cycles, N_2 -purged CV measurements (Fig. 3-7a) show a similar reduction trend in assigned CoS peaks of Co/NS-rGO. Also, N_2 CV (Fig. 3-6a) results indicate the superior capacitive/pseudocapacitive behavior after cobalt incorporation for Co/NS-rGO. In contrast, the excessive cobalt loading resulted in lower capacitance values for 2Co/NS-rGO, possibly due to agglomeration and cobalt oxide (CoO) formation. O_2 CV (Fig. 3-6b) measurements show the same CoS peaks for Co/NS-rGO, which aligns with N_2 CV. Also, the exact figure shows that 2Co/NS-rGO has $Co^{2+} \rightleftharpoons Co^{3+}$ peaks at 0.9 V at both anodic and cathodic scans[217], [218]. These oxidation-reduction peaks are present in CoO-derivative electrocatalysts[217], [218], [219], and we suggest that the relatively high cobalt loading leads to the formation of CoO for 2Co/NS-rGO. In contrast, a lower amount of Co in Co/NS-rGO leads to CoS formation for Co/NS-rGO.

O₂-purged rotating disk electrode (RDE) voltammetry results show a pseudo-limiting current plateau for all three samples. The materials studied in this work have porous characteristics. It is claimed that the supply of reactant (O₂) for porous materials is not the limiting parameter, but the product (HO₂⁻) decomposition rate[220], [221]. It is observed that Co/NS-rGO has the highest current density at -0.15 V, followed by NS-rGO and 2Co/NS-rGO, where the high current of Co/NS-rGO possibly originated from better oxygen adsorption on the catalyst surface[122]. Since 2Co/NS-rGO, Co/NS-rGO, and NS-rGO give -3.8, -5.6, and -5.8 mA/cm² current density at the potential of pseudo-limiting current, Co/NS-rGO and NS-rGO approach to 4e⁻ transfer whereas 2Co/NS-rGO is closer to 2e⁻ transfer because at given 1600 rpm rotation rate and 0.196 cm² surface area 4e⁻ transfer requires -58 mA/cm² of current density according to Levich equation[222] (eq. 2 Supplementary). As discussed by Chlistunoff, we do not link the comparably high Tafel slope (150-210 mV/dec) of our catalysts with either non-efficient catalyst utilization, conductivity problems, or mass transport losses. Still, we associate it with their intrinsic ORR activities[122]. It is postulated that the porous electrodes can exhibit a two-fold Tafel slope of the non-porous electrodes, and peroxide presence can promote high Tafel slope values, too[122], [223]. Considering the high porosity of our materials and current density limitation by peroxide decomposition rate, we postulate that the Tafel slopes of our materials either need to be compared to other porous electrodes or evaluated with an internal reference. We observed the lowest Tafel slope (Fig. 3-7b) for Co/NS-rGO, which refers to the fastest electron transfer rate and better oxygen adsorption properties[122], [224]. In contrast, the low Tafel slope of NS-rGO negligibly changes with Co incorporation to 2Co/NS-rGO.

To elucidate the electron-transfer mechanism (n_{e^-}) for the reaction, the rotating ring disk electrode (RRDE) method is utilized, considering the critical role of RRDE on kinetic investigations in porous electrodes (Fig. 3-6d)[145]. RRDE technique uses the principle of oxidizing and quantifying HO₂⁻ that is produced during ORR by setting the ring potential to 1.1 V. RRDE results (Fig. 3-6d, 3-7c) show that NS-rGO requires high current loads to drive 4e⁻ electron transfer whereas Co/NS-rGO shows Pt/C-like behavior, and the number of transferred electrons is in the 4e⁻ vicinity for all the studied potential range for this catalyst, which makes it a competitive catalyst for PEMFC operations[156]. Also, while the ring current decreases for NS-rGO and Co/NS-rGO below 0.5 V, it increases

for 2Co/NS-rGO, indicating that the catalyst facilitates HO_2^- production in this potential range (Fig 3-7c). Also, 2Co/NS-rGO produces neither a high amount of HO_2^- nor follows a $4e^-$ pathway in the whole potential range, making it an insufficient catalyst for PEMFC operation and peroxide generation. Since Co/NS-rGO behaves like an n-type semiconductor and is not dominated by metallic properties, the catalyst may follow a non-coupled proton-electron transfer (non-CPET) mechanism. Also, on the semiconducting surface, the initial reduction of $\text{O}_{2(\text{aq})}$ to $\text{O}_{2(\text{aq})}^-$ through the outer-sphere mechanism might be the rate-determining step followed by adsorption of O_2^- on a semi-metallic surface[65].

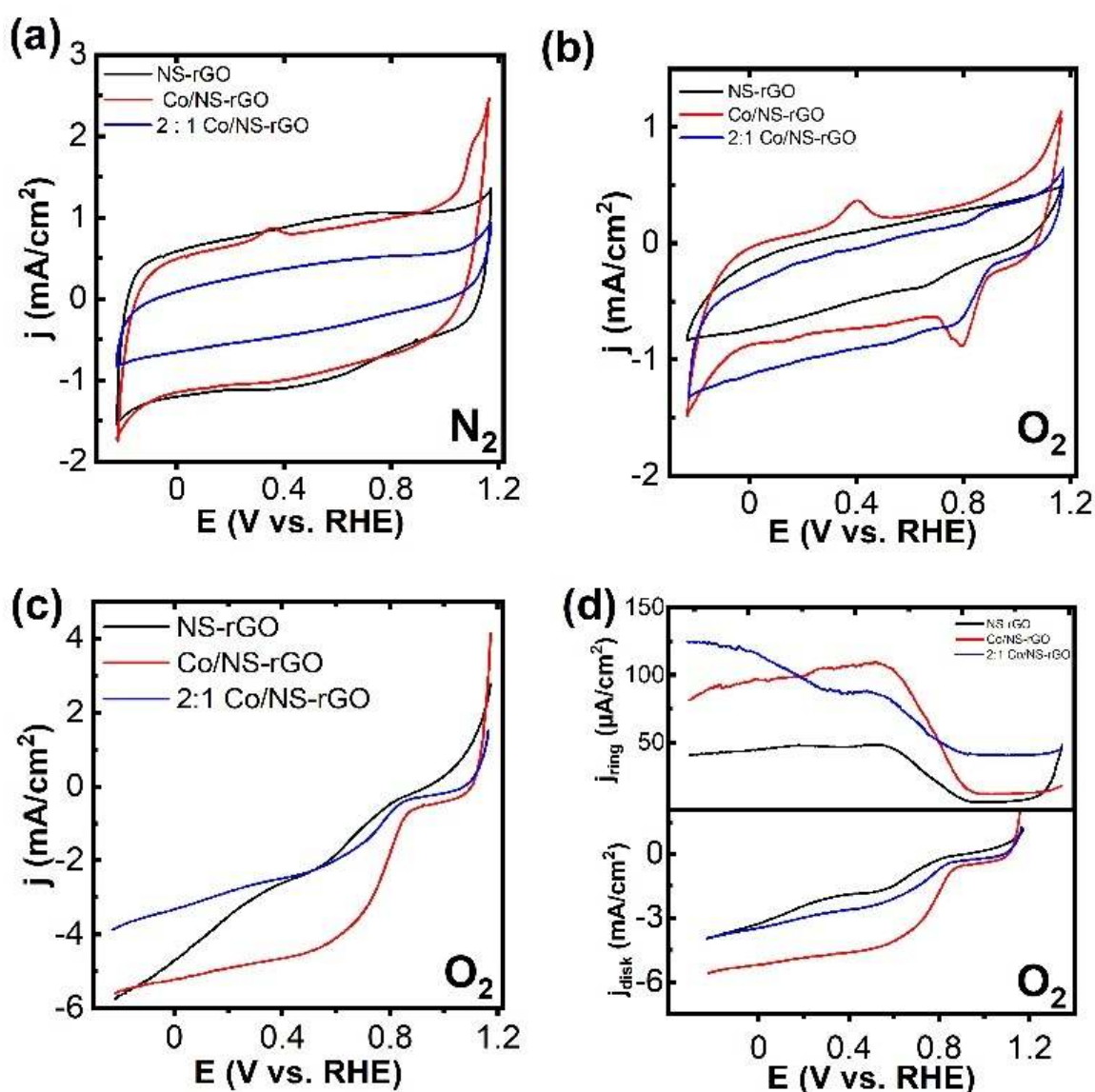


Figure 3-6 Electrochemical Characterizations. Electrochemical Characterizations. a) CV in N_2 -saturated electrolyte. (b) CV in O_2 -saturated electrolyte. (c) RDE in O_2 -saturated electrolyte. (d) RRDE in O_2 -saturated electrolyte..

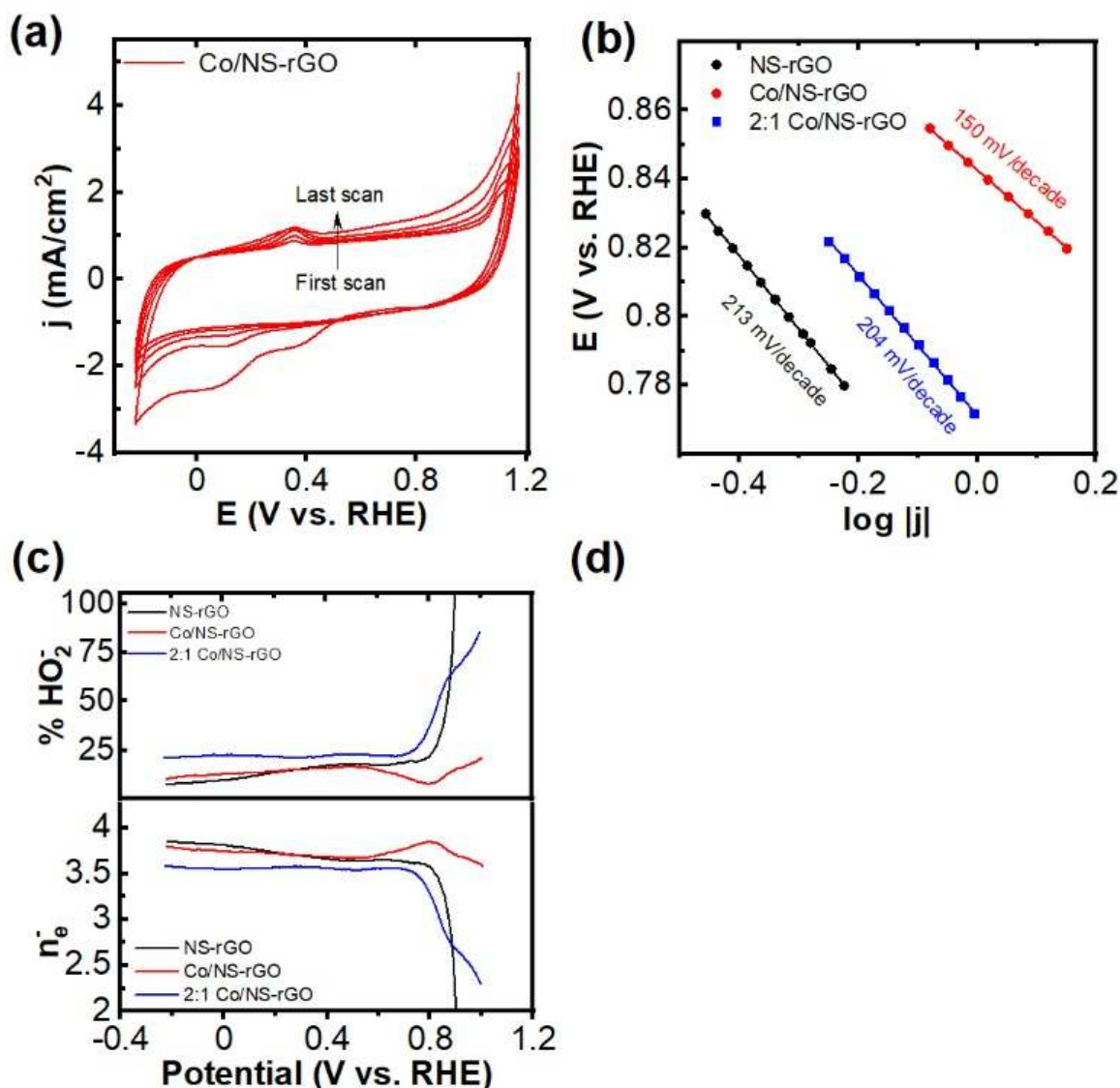


Figure 3-7 Electrochemical activity tests. a) CV in in O_2 -saturated electrolyte. b) Tafel slope analysis of RDE in O_2 -saturated electrolyte. c) RRDE in O_2 -saturated electrolyte.

Capacitive measurements further unravel the electrocatalysts' high porosity and unique ORR behavior. Double-layer capacitance (C_{dl}) values are measured with different scan rates, and the materials' electrochemical surface area (ECSA) is calculated by taking the C_s value as 0.022 mF/cm² [157]. We observed that the ECSA after cobalt incorporation is decreased for both catalysts (Fig. 3-8a). Still, Co/NS-rGO obtains a higher ECSA than 2Co/NS-rGO, which can be linked to its superior ORR activity (Fig. 3-9a). Furthermore, the ECSA normalized RDE and RRDE results show that the Co/NS-rGO has the superior intrinsic activity for ORR (Fig. 3-9a, b). In contrast, despite NS-rGO having the highest ECSA, the worst ECSA normalized ORR activity indicates poor ORR performance.

Furthermore, metal mass activity normalized RDE and RRDE graphs (Fig. 3-9 c, d) show that the Co/NS-rGO catalyst has far superior ORR activity compared to 2Co/NS-rGO originating from the superior metal utilization of the catalyst. Since ORR activity originates from the electronic properties of the material, change in ORR behavior of the electrodes is elucidated by the density of states (DOS), flat-band potential (V_{fb}), and doping density (N_D) values, which are derived from electrochemical impedance spectroscopy (EIS) measurements[131], [159].

EIS measurements in Fig. 4b are performed around the onset potential in N_2 -saturated electrolytes. R_1 is assigned to equivalent series resistance (ESR), which emerges from solution resistance and electrical connections. Constant-phase element (CPE) Q is used for modeling instead of C owing to the non-ideal behavior of the materials[158]. Q_1 is assigned to the total double-layer capacitance build-up in the electrode-electrolyte interface, where R_2 is the corresponding resistance. Bending in the high-frequency region is denoted as equivalent distributed resistance (EDR) that arises from ion transport to the electrode surface[225]. 2Co/NS-rGO holds a significantly higher EDR value than other samples, possibly due to the stacking emerging from the high amount of cobalt in the catalyst. W_1 , Warburg impedance, is assigned to the low-frequency behavior of the samples associated with the mass transfer process in the catalysts[158]. Since the selected potential is in close vicinity with the potential of electron transfer for ORR, the density of states around the Fermi level ($D(E_F)$) is calculated for each catalyst at a given potential (eq. 8 supplementary) [159], [226], [227]. The highest ($D(E_F)$) value of $2.75 \times 10^{22} \text{ eV}^{-1} \text{ cm}^{-3}$ (Table 1) is observed for Co/NS-rGO, indicating the highest concentration of electronic states around the Fermi level possibly due to the presence of CoS species[228] and agrees with the superior kinetic behavior of the sample towards ORR. However, a high amount of Co for 2Co/NS-rGO decreased $D(E_F)$, which led to slower ORR kinetics.

Mott-Schottky measurement is carried out to analyze the samples' capacitive behavior further (Fig. 3-8c). Firstly, it is found that all the materials show n-type semiconductor behavior. Materials have attained n-type semiconductor properties, possibly due to the presence of N, S, and O species along the graphene basal plane[65]. N_D normalization uses ECSA instead of BET surface area to include active sites specific to electrochemical activity. It is observed that the Co/NS-rGO has the highest N_D value, indicating that the catalyst obtains the highest carrier concentration to perform ORR,

which signifies its superior ORR kinetics (Table 3-1). According to eq. 10, the highest V_{fb} is observed for Co/NS-rGO, whereas a sharp decrease in V_{fb} is detected for 2Co/NS-rGO (Table 3-1). The V_{fb} trend agrees with RDE results, which show the lower overpotential of Co/NS-rGO required to perform oxygen electroreduction[159].

The capacitance versus potential graph (Fig. 3-8d) shows the highest total double-layer capacitance (C) for Co/NS-rGO and the lowest C value for 2Co/NS-rGO. A decrease in capacitance in potentials positive of V_{fb} verifies the materials' n-type semiconductor properties[132].

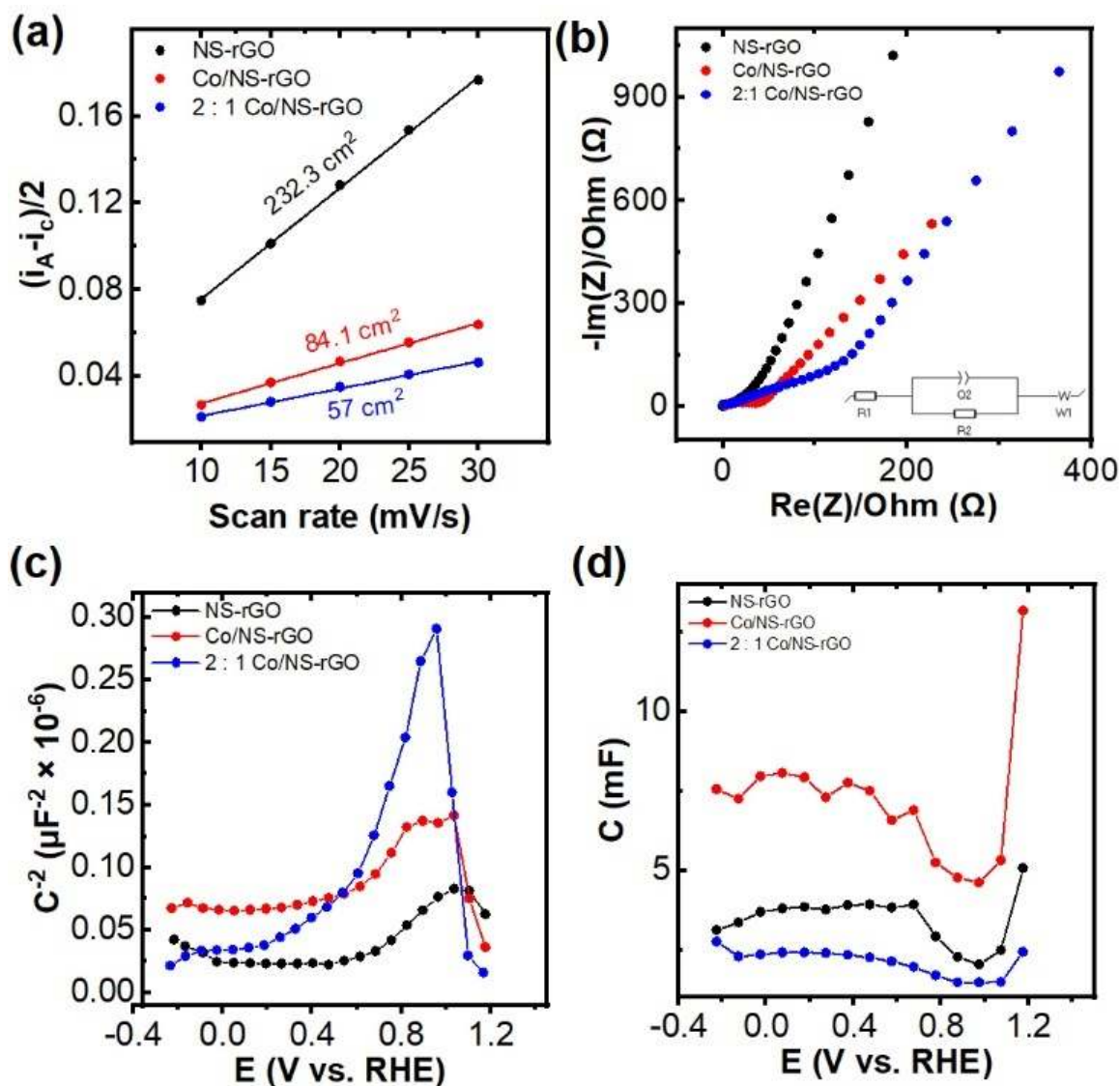


Figure 3-8 Electrochemical screenings using voltammetry and impedance techniques. a) ECSA in N_2 -saturated electrolyte. b) EIS in N_2 -saturated electrolyte. c) Mott-Schottky in N_2 -saturated electrolyte. d) Capacitance measurements in N_2 -saturated electrolyte.

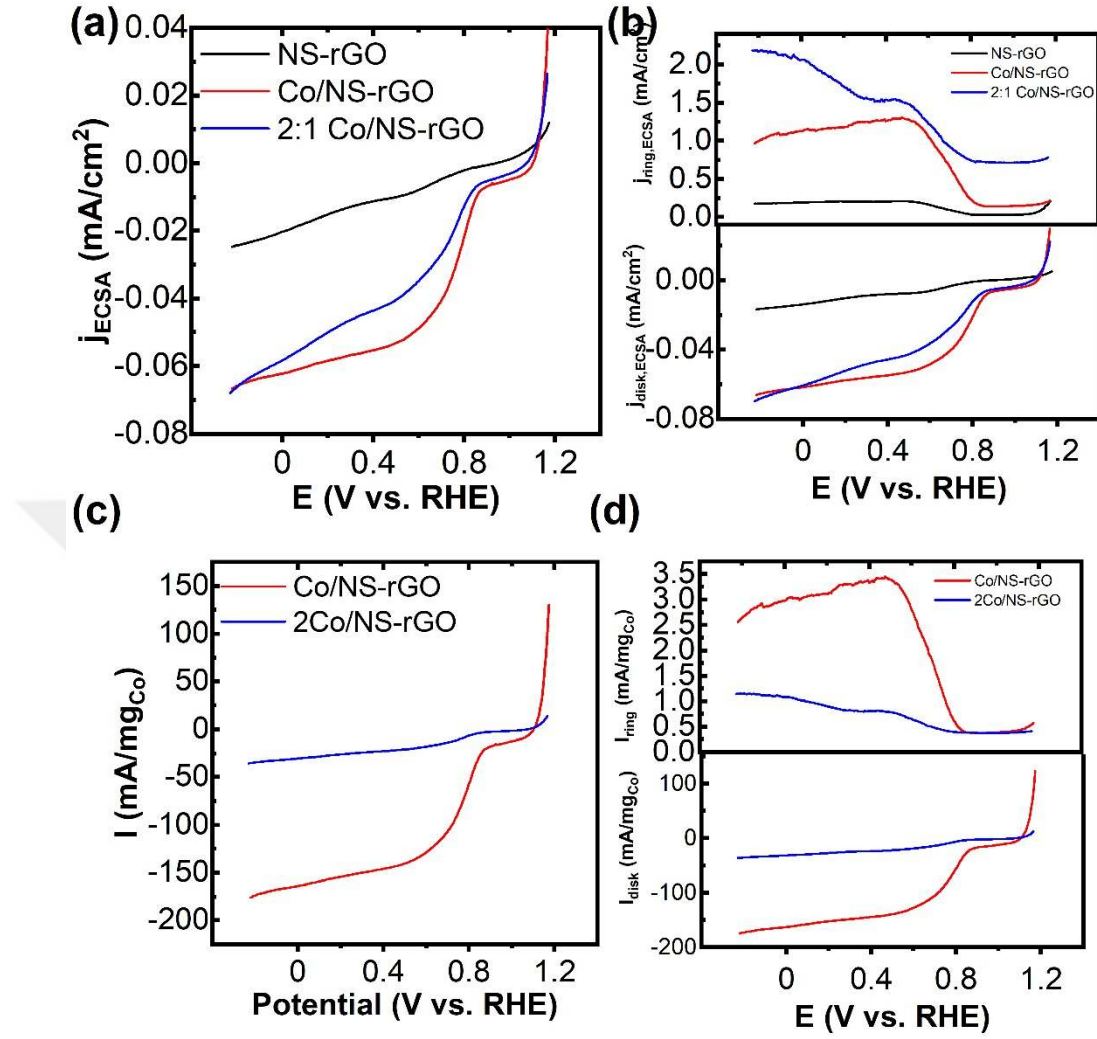


Figure 3-9 ECSA normalized electrochemical activity tests. a) ECSA normalized RDE in O₂-saturated electrolyte. b) ECSA normalized RRDE in O₂-saturated electrolyte. c) Metal mass activity normalized RDE. d) Metal mass activity normalized RRDE.

	$D(E_F)$ [eV ⁻¹ cm ⁻³]	Carrier Density (N_D) [cm ⁻³]	Flatband Potential (V_{fb}) [V]
NS-rGO	2.43×10^{22}	4.76×10^{25}	0.865
Co/NS-rGO	2.75×10^{22}	2.66×10^{26}	0.891
2Co/NS-rGO	2.17×10^{22}	2.2×10^{26}	0.666

Table 3-1 Electronic properties of density of states, carrier density, and flatband potential for the materials

3.5 Conclusion

Herein, we develop Co/NS-rGO and 2Co/NS-rGO catalysts with 1:2 Co content, CoS presence, and Co SA distribution along the graphene plane. With an optimum cobalt amount, the Co/NS-rGO catalyst would show lower overpotential and R_{ct} , higher ECSA, current density, and capacitance values. In contrast, the opposite happens when more cobalt is used for 2Co/NS-rGO. When the electronic properties of the catalysts are investigated through EIS techniques, it is found that catalysts show n-type semiconductor behavior and higher flatband potential, density of states, and carrier concentration. CoS formation is found using the XPS technique, while HR-STEM shows the distribution of Co SAs on the N, S-rGO matrix. Synergistic effects between CoS, Co SAs, and the NS-rGO basal plane are related to the catalyst's higher ORR activity.

Chapter 4:

Mechanistic Insights into Nitrate-to-Ammonia Electroreduction Through Constant and Pulsed Electrolysis Using Co/NS-rGO Catalysts

4.1 Introduction

4.1.1 Electrification of NH₃ Production Using Waterborne N-Species

Ammonia (NH₃) is a vital carbon-free energy carrier and a chemical commodity for its intensive use in (1) fertilizers to supply nutrition for the increasing human population[229], (2) pharmaceutical and heavy industry[9], [230], and (3) energy storage owing to its high energy density[13], [231]. However, the energy-intensive Haber-Bosch (H-B) process accounts for 90% of the annual global NH₃ production[232] and approximately 1.8% of the worldwide CO₂ emission[233]. The disruptive impact of the H-B process on the N-cycle due to anthropogenic activities has led to a significant shift in focus towards the green electrochemical synthesis of NH₃ through the reactive N-species, offering a promising alternative[133], [234], [235]. The electrochemical nitrogen (N₂) reduction reaction (NRR) presents an alternative route for sustainable and environment-friendly NH₃ production[236]. However, despite the intensive efforts[126], [237], [238], [239], [240], [241], [242], [243], [244], NRR suffers from poor NH₃ selectivity and two orders of magnitude lower yield rate than the H-B process[236] due to the high disassociation energy of N≡N bond (945 kJ mol⁻¹) in N₂[245], the low solubility of N₂ in aqueous electrolytes, and the presence of the competing hydrogen evolution reaction (HER) around the similar thermodynamic reduction potential[246]. Also, the quantification challenges in NRR experiments due to NH₃ contamination raise the question of the reliability and reproducibility of the NRR literature[247]. Furthermore, using Ultra-high vacuum systems in high-performance NRR studies may reflect the difficulty of producing scalable NH₃ at room T by reducing N₂[237], [243], [244]. When the bottlenecks of NRR are considered, potential solutions using different reactive N-species, i.e., NO₃⁻, NO₂⁻[248], for green and scalable ambient

electrochemical NH_3 synthesis, have emerged, offering an optimistic outlook for a more sustainable NH_3 production[249].

An abundance of nitrate (NO_3^-) ions exists in (1) agricultural runoffs due to overfertilization practices, (2) sewage, and (3) industrial wastewater streams, which cause significant pollution of surface water and underground aquifers[233]. Excessive nitrate contamination significantly threatens human health and the ecosystem[234], [250]. It causes the formation of eutrophication in wildlife, methemoglobinemia, and cancer in humans with their presence in drinking water[251]. Being hazardous to human health and the environment, there has been immense research on eliminating waterborne N-species through water remediation and denitrification[252]. These studies consider NH_3 an unwanted product since they aim to produce N_2 by reducing nitrate and purifying the water. On the other hand, one can have a sustainable NH_3 production route by capturing and isolating NO_x^- species abundant in wastewater and converting them into economically valuable NH_3 . This autonomous system can effectively eliminate nitrate species' detrimental effects and produce NH_3 to compensate for the H-B process[253]. On top of the environmental merits, NO_3^- has a significantly lower disassociation energy of N=O bond (240 kJ mol^{-1}) and very high solubility in aqueous solutions[249], [254]. Furthermore, the recent low-energy plasma technologies that produce NO_3^- from the air add to its widespread availability[255].

4.1.2 Biological Insights into Electrochemical NO_3RR

In microorganisms, NO_3RR is carried out by a tandem process where nitrate-to-nitrite reduction is carried out by S-containing Mo-cofactors and nitrite-to-ammonia reduction is carried out by S-containing Fe and Cu-cofactors separately[256]. Many studies that mimic biological NO_3RR can achieve high NH_3 selectivity and yield. Ren et al. used a bioinspired approach where they studied Cd single-atom anchored $\text{Mo}_2\text{TiC}_2\text{T}_x$ catalyst to achieve >95% Faradaic efficiency (F.E.) and $48.5 \text{ mg h}^{-1} \text{ cm}^{-2}$ NH_3 yield rate by activating NO_3^- by Mo active center as of nitrate reductase and suppressing HER simultaneously[108]. Murphy et al. utilize a FeMo-N-C atomically dispersed catalyst that follows a cascade NO_3RR pathway through associative and disassociative adsorption of NO_3^- . They could effectively adsorb and reduce NO_3^- at Mo- N_4 sites to desorbable NO_2^- and $^*\text{NOOH}$ to NH_3 on Fe-N-C. The catalyst achieved 94% F.E. and a

high stability of 90% F.E. for 60 h[109]. Abbott et al. achieved 41% F.E. and $12.9 \mu\text{mol h}^{-1}\text{mg}^{-1}$ using Mo_2CT_x : Fe through NO_3^- binding to vacancies promoted by Fe sites and further reduction to NH_3 [257].

4.1.3 Electrochemical NO_3RR : Electronic Properties and The Effect of Pre-reduction

The material's electronic properties are significant for NH_3 production, especially in NO_3^- adsorption and water disassociation to form protons. Li et al. investigated the adsorption characteristics of the intermediates on Cu sites by projected partial density of states (PDOS) calculations. PDOS results showed that the Cu- NO_2 model exhibited higher energies of electrons than Cu- NO_3 and Cu- H_2O by the positive shift in Cu d orbital. This shift signified the better activation of the NO_2 adsorbed species for NO_3RR [258]. Similarly, Zhou et al. examined the material's electronic structure using PDOS, where the interaction between Cu and Co in the alloy allows spontaneous NO_3^- reduction and showed that H_2O disassociation is the rate-determining step for the reaction[144]. Carvalho et al. described through the electronic structure analysis that the NH_3 selectivity increases when H^* binding affinity is enough to reduce the NO_3^- while not contributing to HER. Also, materials' selectivity towards NH_3 increases when d-band center energy (E_d) approaches or overcomes the Fermi level E_f [259].

The NO_3RR literature does not comprehensively examine pre-reduction effects except for synthesis[99] or control experiments[260]. In contrast, pre-reduction is extensively studied on CO_2RR [127], [261], [262], [263] and ORR[264], which is applied to remove the oxide layer or O content, high surface roughness, and better kinetics.

4.1.4 Pulsed Electrolysis of NO_3RR -to- NH_3

Recently, pulsed electrolysis has been commonly studied to increase NH_3 yield and F.E. by causing morphological or oxidation state changes in the material[265] or modulating the NO_3^- to NH_3 mechanism. Pulsed electrolysis was used with cathodic and anodic potentials in multiple articles to denote the contribution of reconstruction of the catalyst to higher nitrate-to-ammonia activity[266]. In one study, oxidative and

reductive potentials are applied in a pulsed fashion to lower interfacial pH and replenish the electrical double-layer (EDL). However, they found that the lower interfacial pH promoted HER and caused nitrite selectivity to increase. Also, they argue that the N-selectivity can be changed due to nitrite desorption and prevent NH_3 electrosynthesis[134]. Another article proposes that the positive bias in electrode potential allows more nitrate accumulation on the electrode and more NH_3 production[267]. On the other hand, Li et al. applied negative potentials to modulate $\text{NO}_3^- - \text{NO}_2^- - \text{NH}_3$ modulation. They could confine NO_2^- in the high -0.8 V pulsed potential, suppress HER by NO_2^- to NH_3 reduction, and achieve high NH_3 F.E. through tandem NO_3^- to NH_3 reduction[258].

4.1.5 Aims of Study

Here, we use a cobalt-decorated Nitrogen, Sulfur-doped reduced Graphene Oxide (Co/NS-rGO), and Co/NS-rGO 900 catalyst. The former consists of CoS catalytic pairs in an N, S doped graphene structure, while the latter has a disrupted carbon structure enriched with CoO species. We investigated the effect of CoS and CoO species to promote NO_3RR and NO_2RR through potentiostatic and pulsed electrolysis experiments. We utilized pulsed potentials to examine the impact of more negative potentials to drive confined NO_2^- to NH_3 reduction by enabling water disassociation via inferior H^* adsorption without promoting HER or less negative potentials to investigate the dominance of N-O bond cleavage and NO_2^- accumulation. We used electrochemical Tafel analysis to understand whether a chemical, adsorption, or electron transfer step occurs on different potentials. Furthermore, we proposed two different mechanisms for two different Co catalysts using XAS (X-ray absorption spectroscopy), EIS (Electrochemical impedance spectroscopy), and DRT (Distribution of relaxation times)[268] model principles.

4.2 Experimental Method

4.2.1 Synthesis of NS-rGO, NS-rGO 900, Co/NS-rGO, Co/NS-rGO 900 Catalysts

Graphene Oxide (GO) is synthesized using improved Hummer's method. Nitrogen-Sulfur doped reduced Graphene Oxide (NS-rGO) catalyst is synthesized by autoclave of 1:5 GO and Thiourea mixture at 190 °C for 18 h through a hydrothermal method.

Cobalt decorated NS-rGO (Co/NS-rGO) catalyst is synthesized by drying the 1 mL Co-DMF precursor and 1:1 NS-rGO and DMF solution at 90 °C. NS-rGO 900 and Co/NS-rGO 900 electrocatalysts are synthesized by the 1 hour pyrolysis of NS-rGO and Co/NS-rGO in N₂ environment for 1 h.

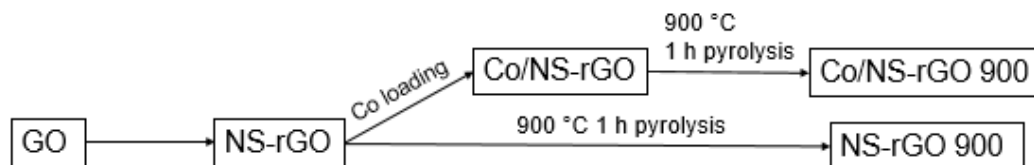


Figure 4-1: Classification of Electrocatalysts

4.2.2 Structural Characterizations

X-ray diffraction (XRD) measurements are performed using a Bruker D2 Phaser X-ray Diffractometer records equipped with a Cu K α radiation with 10 kV potential and 30 mA current. Transmission electron microscopy (TEM) images are taken using 200 kV aberration-corrected FE-TEM/STEM Hitachi HF5000 device. Scanning electron microscopy (SEM) images are attained using A 20 kV Zeiss Ultra Plus microscope. Thermo K-alpha instrument that has monochromatized Al K α (1486.7 eV) radiation is used to carry out X-ray photoelectron spectroscopy (XPS) results. The Raman spectroscopy is carried out using a Renishaw INVIA device at the laser wavelength of 532 nm. XAS measurements are carried out in SESAME (Synchrotron-light for Experimental Science and Application in the Middle East) XAS device with energy range of 3-30 keV and counting rate of minimum 3.2 Mcps.

4.2.3 Electrochemical Tests

NO₃RR and NO₂RR experiments are carried out in 0.1 M KOH + 0.1 M KNO₃ or KNO₂ electrolytes in both anodic and cathodic compartments that are separated by a Nafion membrane. An H-cell with conventional three electrode configuration having Ag/AgCl reference electrode, Pt counter electrode and coated Glassy Carbon electrode with catalyst loading of 2.122 mg/cm² is used for electrochemical NH₃ production experiments. H₂ detection experiments are carried out using a Gas Chromatography device with thermal conductivity detector (TCD) while NH₃ and NO₂⁻ detection experiments are performed using UV-Vis Spectroscopy by Indophenol Blue Method

and Griess' Test respectively⁸. The iR compensation and all EIS measurements was carried out at 85% and 100 kHz using either a Biologic SP-200 or VSP-300. Also, pulsed CA measurements are carried out at a fixed potential of -0.9 V or -0.4 V and one potential between -0.2 V to -1.2 V at each potential for 2 s by summing up to 30 mins.

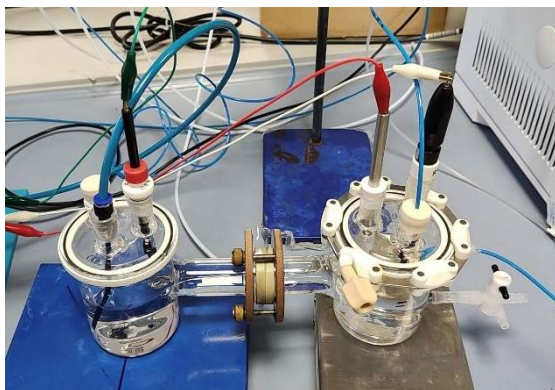


Figure 4-2: H-Cell Configuration

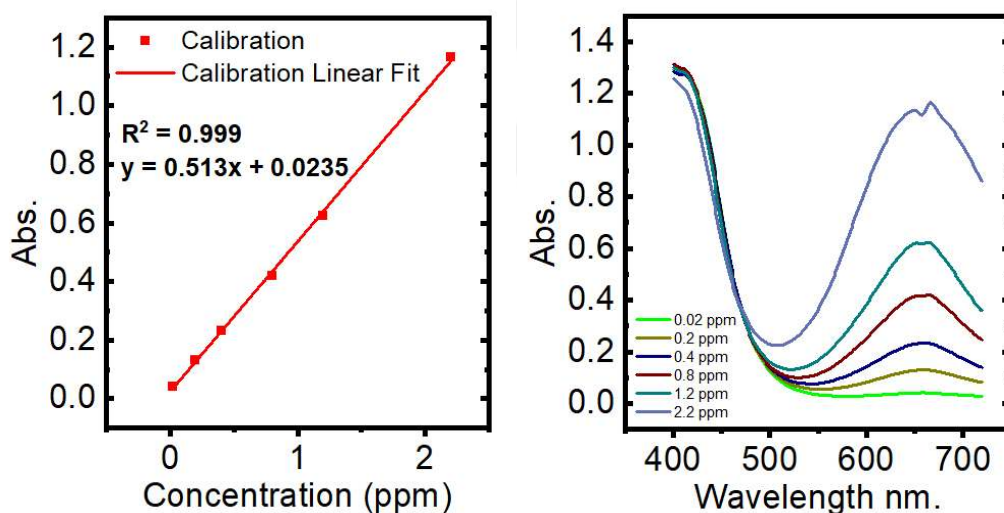


Figure 4-3: NH₃ Calibration using Indophenol Blue Method

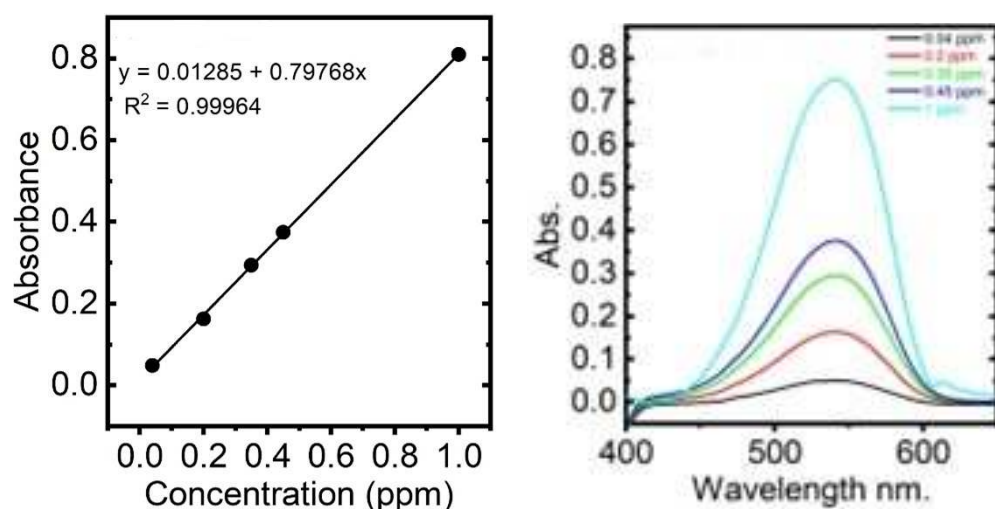
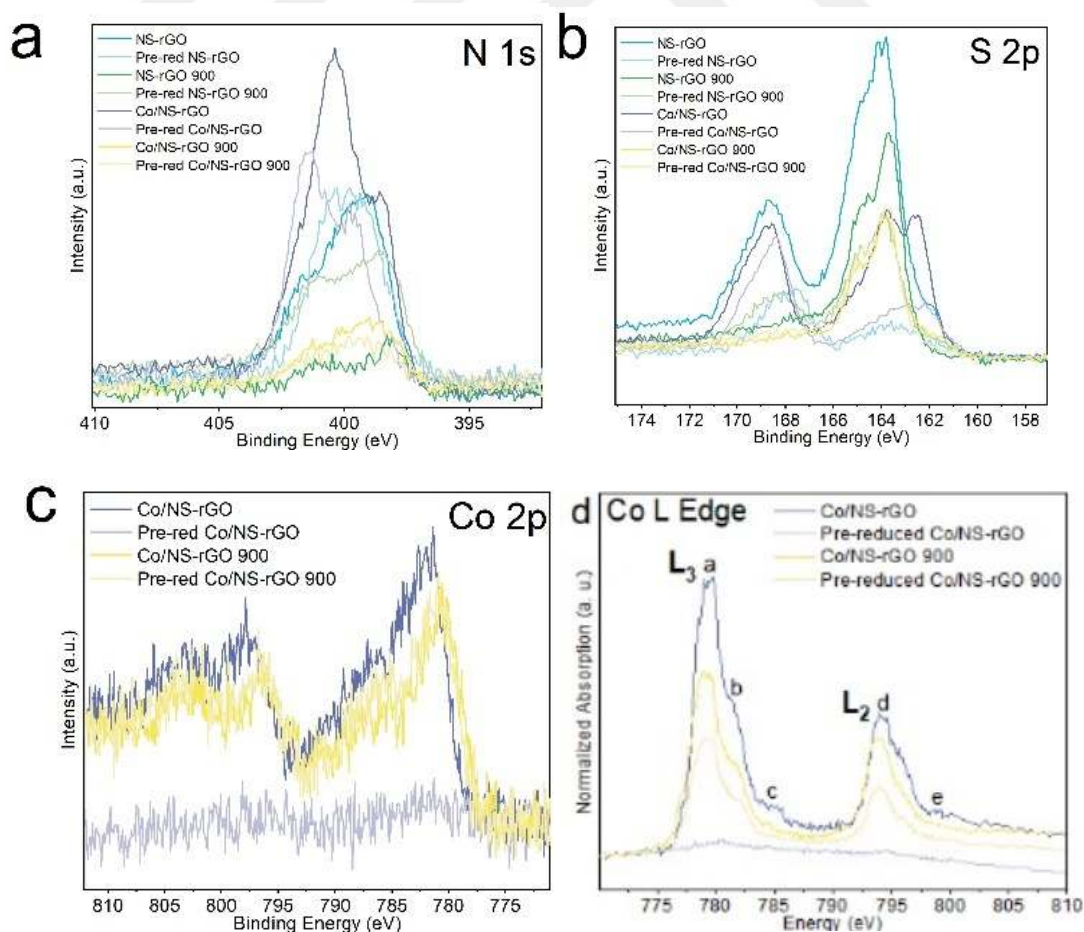
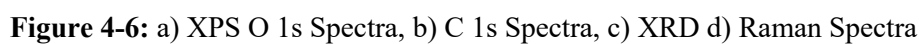


Figure 4-4: NO₂ Calibration using Griess' Test

4.3 Surface Characterization of Co/NS-rGO Through XPS and XAS Techniques





				I(L ₃)/ I(L ₃)+
Sample	I(L ₃)	I(L ₂)	I(L ₃)/ I(L ₂)	I(L ₂)

				Branching Ratio
Co/NS-rGO	0.535	0.274	1.953	0.661
Pre-red Co/NS-rGO	0.0278	0.00742	3.747	0.789
Co/NS-rGO 900	0.352	0.226	1.558	0.609
Pre-red Co/NS-rGO 900	0.262	0.131	2	0.667

Table 4-1: Co L-Edge Peak Intensity and Branching Ratio

The N 1s spectra show that Co/NS-rGO obtains pyridinic and pyrrolic N after Co incorporation to NS-rGO through surface functionalization. At the same time, the other samples primarily inherited their N-character from NS-rGO. Furthermore, the pre-reduction induced N-species with higher oxidation energies for all samples (Fig. 4-5a). The S 2p spectra show the emergence of a doublet peak for Co/NS-rGO around 162-164 eV, which indicates the formation of CoS in the catalyst. On the other hand, the S behavior of NS-rGO stays the same after the sample is subjected to either T or Co incorporation, only T treatment eliminating the sulfate content. Furthermore, S 2p spectra show that Co/NS-rGO preserves its CoS species even after pre-reduction (Fig. 4-5b). The Co 2p spectra in XPS show that the $\text{Co}^{3+}/\text{Co}^{2+}$ ratio increases after T treatment even though the catalyst is Co^{2+} rich. An increase in C 1s peak intensities (Fig. 4-6a.) and I_D/I_G ratio in Raman Spectra and a decrease in O 1s peak intensities (Fig. 4-6b.) is observed after T treatment. Also, peak intensities in O 1s for pre-reduced samples are attributed to the presence of H_2O in the sample after the interaction with the electrolyte. The I_D/I_G ratio increase shows the presence of defects in the graphene plane. XRD patterns (Fig. 4-6c.) show the presence of (002) graphene plane for all samples. However, narrower peaks for NS-rGO 900 and Co/NS-rGO 900 imply the existence of a more graphitized structure, while broader peaks for the remainder show the presence of a few-layer graphene for these materials[269], [270]. The presence of (102) graphene layers, which indicates the short-range order in stacked layers, is also present in all samples except Co/NS-rGO 900. This finding also agrees with the disruption of the graphene plane for Co/NS-rGO 900. On the other hand, the presence of (220), (222), and (400) planes of Co_3O_4 [271], (101), (012) planes of $\text{Co}(\text{OH})_2$, and (002), (101) planes of Co[272] and disappearance of (100) plane is observed for Co/NS-rGO 900.

In Co L edge XAS (Fig. 4-5d), an increase in the $I(L_3)/I(L_2)$ peak intensity ratio and the emergence of c and e peak for Co/NS-rGO shows the presence of CoS in the catalyst[153]. Furthermore, a higher $I(L_3)/I(L_2)$ ratio indicates higher electron occupancy in the d orbital, and higher branching ratios refer to a high spin state (HS) while a low spin state (LS) for the opposite case[273]. It is important to note that the occupied d orbitals can cause weak binding of NO_x^- species due to high electron-electron repulsion[274]. Also, for example, the HS Co^{2+} in Co/NS-rGO can induce weak Co-O bond formations and suffer from low NO_x^- adsorption, while LS Co^{2+} in Co/NS-rGO 900 can facilitate strong Co-O bonds and facilitate NO_x^- adsorption, but the protonation becomes difficult. In addition, pre-reduction increases d-orbital occupancy and allows transition to higher spin states for both electrocatalysts.

O K-Edge XAS spectra (Fig. 4-5e) refer to the hybridization between Co 3p orbitals and O 2p unoccupied density of states by the fact that as peak (t_{2g}) intensity at 531 eV increases the peak intensity at (e_g) decreases and vice versa. For Pre-red Co/NS-rGO, a decrease in t_{2g} peak intensity induces an increase in the, e.g., peak intensity and an electron exchange from, e.g., orbitals to t_{2g} , which causes higher d orbital occupancy, which is also shown by Co L-Edge peak intensity analysis[275]. A similar phenomenon of electron exchange between N_{py} and, e.g., orbitals is observed for N K-Edge, corresponding to peaks of 399.8 and 401 eV consecutively (Fig. 4-5f)[276].

4.4 Structural Characterizations of Catalysts

It is observed that NS-rGO has a well-preserved layered graphene structure after N and S bonding, where characteristics of graphene flakes and paper-like structure are visible (Fig. 4-7). Even though subjected to 900 C pyrolysis for 1 h, NS-rGO 900 also has graphene layers, and the effects of carbonization are visible (Fig. 4-8) which is further supported by CHNS analysis (Fig 4-10 a, b.) In Fig. 4-9b, the Co/NS-rGO HR-STEM image shows the graphene layers and distribution of Co atoms in the basal plane. Furthermore, Co/NS-rGO SEM EDX (Fig. 4-11 a, c.) shows the presence of CoS pairs, also proven by XPS S 2p spectra. Co/NS-rGO 900 SEM (Fig. 4-5d) shows the disruption of the graphene plane for Co/NS-rGO and the presence of cobalt oxide particles shown by SEM EDX (Fig. 4-11 b, d). Also, ICP-MS results show the ultra-low

Co content in the catalysts, which are 1.6% for Co/NS-rGO and 1.9% for Co/NS-rGO 900 (Fig. 4-10 c).

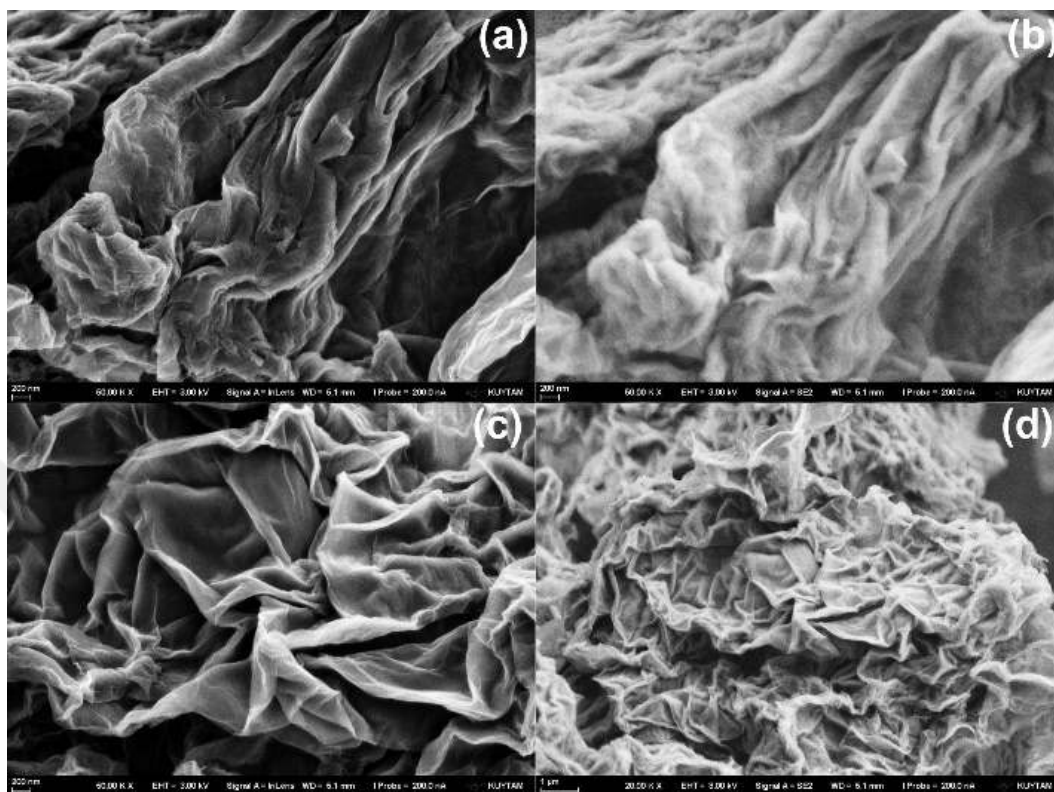


Figure 4-7: SEM images for NS-rGO

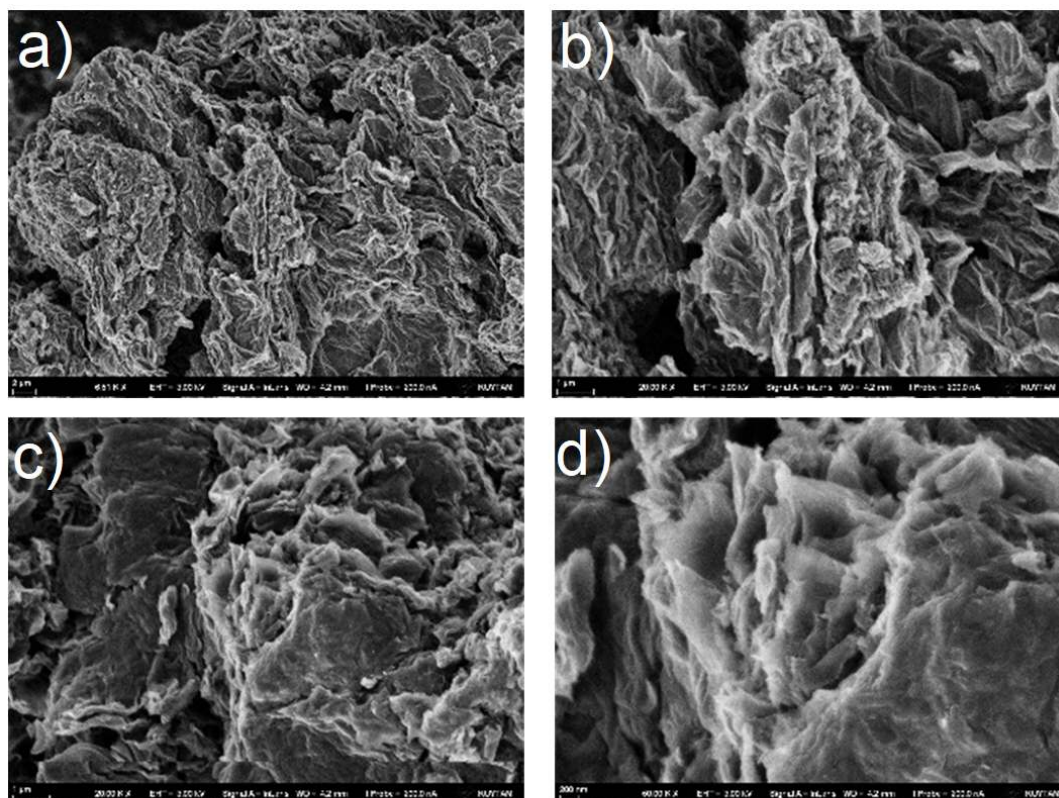


Figure 4-8: SEM images for NS-rGO 900

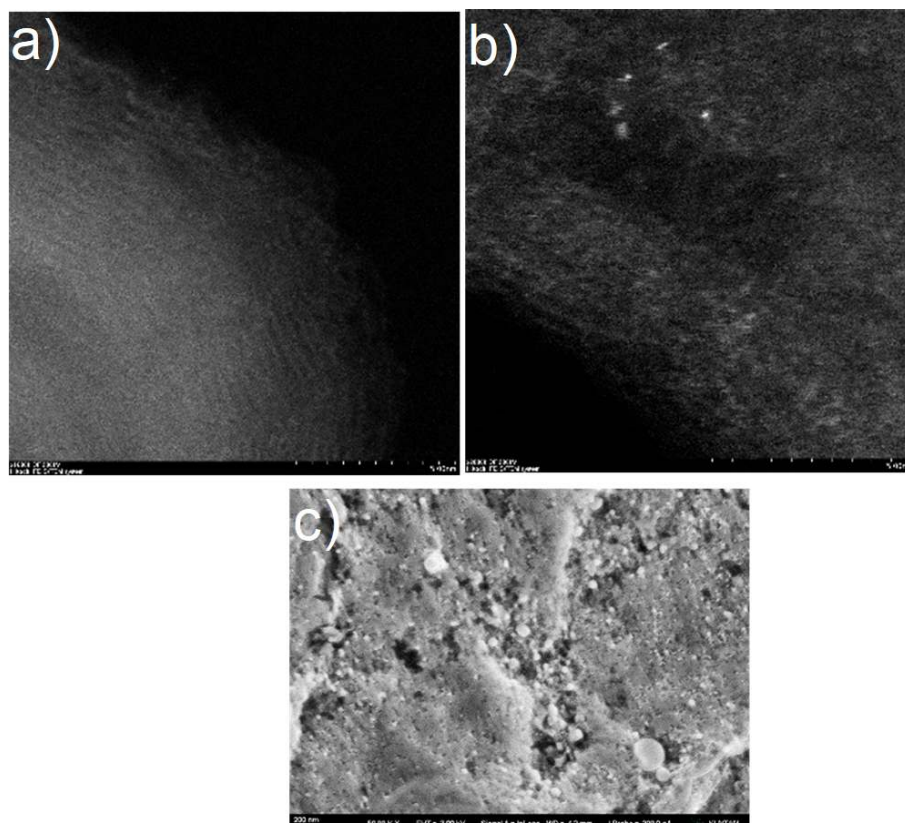


Figure 4-9: a, b) HR-STEM images for Co/NS-rGO. c) Co/NS-rGO 900 SEM

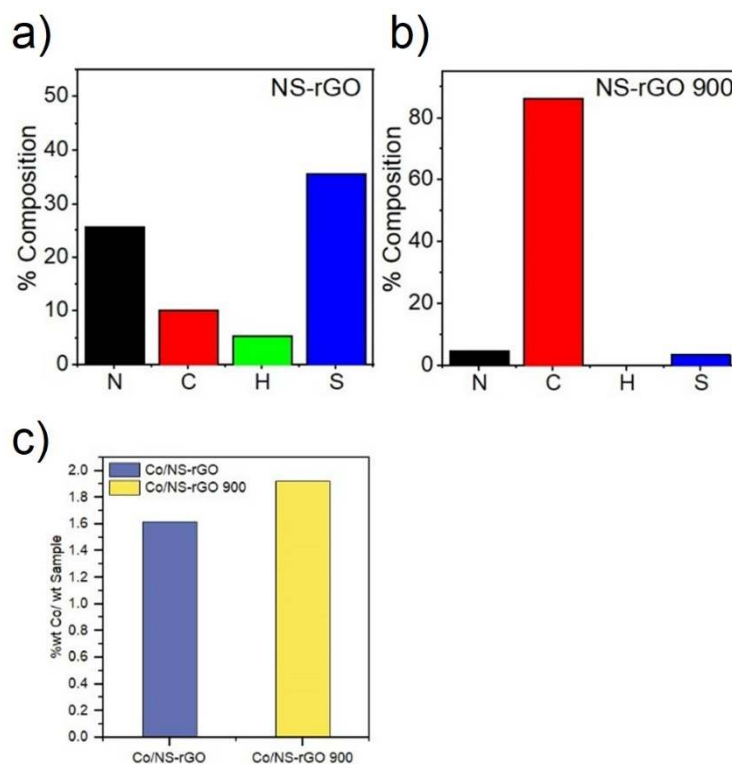


Figure 4-10: a) CHNS Analysis for NS-rGO and NS-rGO 900 b) ICP-MS for Co/NS-rGO and Co/NS-rGO 900

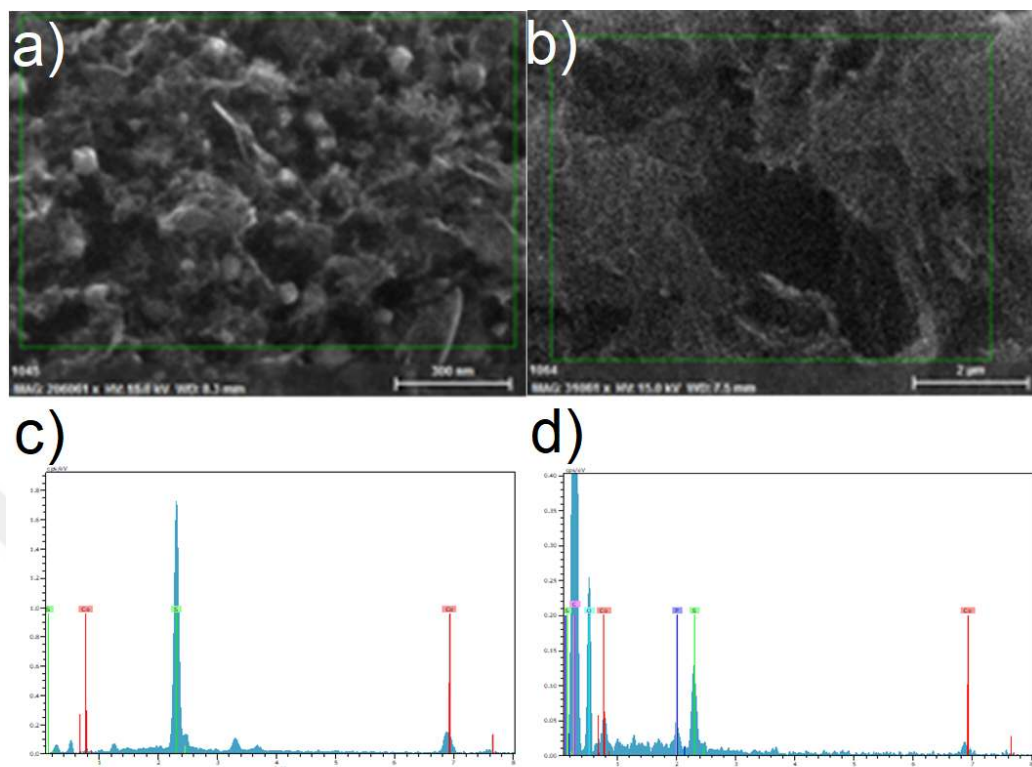


Figure 4-11: EDX Spectra, Microscopical Images, and ICP-MS. a) Co/NS-rGO area-specific SEM c) EDX b) Co/NS-rGO 900 area-specific SEM d) EDX

4.5 Tafel Analysis

LSV measurements show that except NS-rGO 900 NO₃RR, an already strongly reduced sample, pre-reduction allows a considerable decrease in overpotential. Furthermore, pre-reduction enables a plateau in the kinetic region for NS-rGO and Co/NS-rGO samples (Fig. 4-12a). It is observed that Co/NS-rGO 900 and Co/NS-rGO have the lowest overpotential for NO₃RR and NO₂RR respectively. Also, the lowest overpotential for HER belongs to Co/NS-rGO 900 and the highest to Co/NS-rGO (Table 2).

Tafel slope values indicate that only for pre-reduced NS-rGO and Co/NS-rGO 900 samples, RDS for NO₃RR is 1 e⁻ transfer for NO₃⁻ to NO₂⁻ reduction due to the slope being slightly lower than 120 mV⁻¹/decade. NO₃RR RDS for other catalysts is determined as the initial adsorption and activation of NO₃⁻ due to their higher Tafel slope values. For NO₂RR, NS-rGO and Co/NS-rGO 900 are limited by the binding of NO₂⁻, while for NS-rGO 900, RDS is the NO₂⁻ to NO e⁻ transfer. Furthermore, the 72.08

mV/decade Tafel slope of Co/NS-rGO indicates that for Co/NS-rGO, RDS is a chemical step that can be the coupling of strong adsorption of intermediate, i.e., NO^* , $^*\text{H}$ species. Tafel slopes for HER show that Co/NS-rGO 900 has the fastest kinetics for HER, while Co/NS-rGO has a significantly higher Tafel slope value[99]. Therefore, HER can suppress the NO_3RR activity of Co/NS-rGO 900 at lower overpotentials, while slower kinetics of HER for Co/NS-rGO can facilitate NO_3RR protonation.

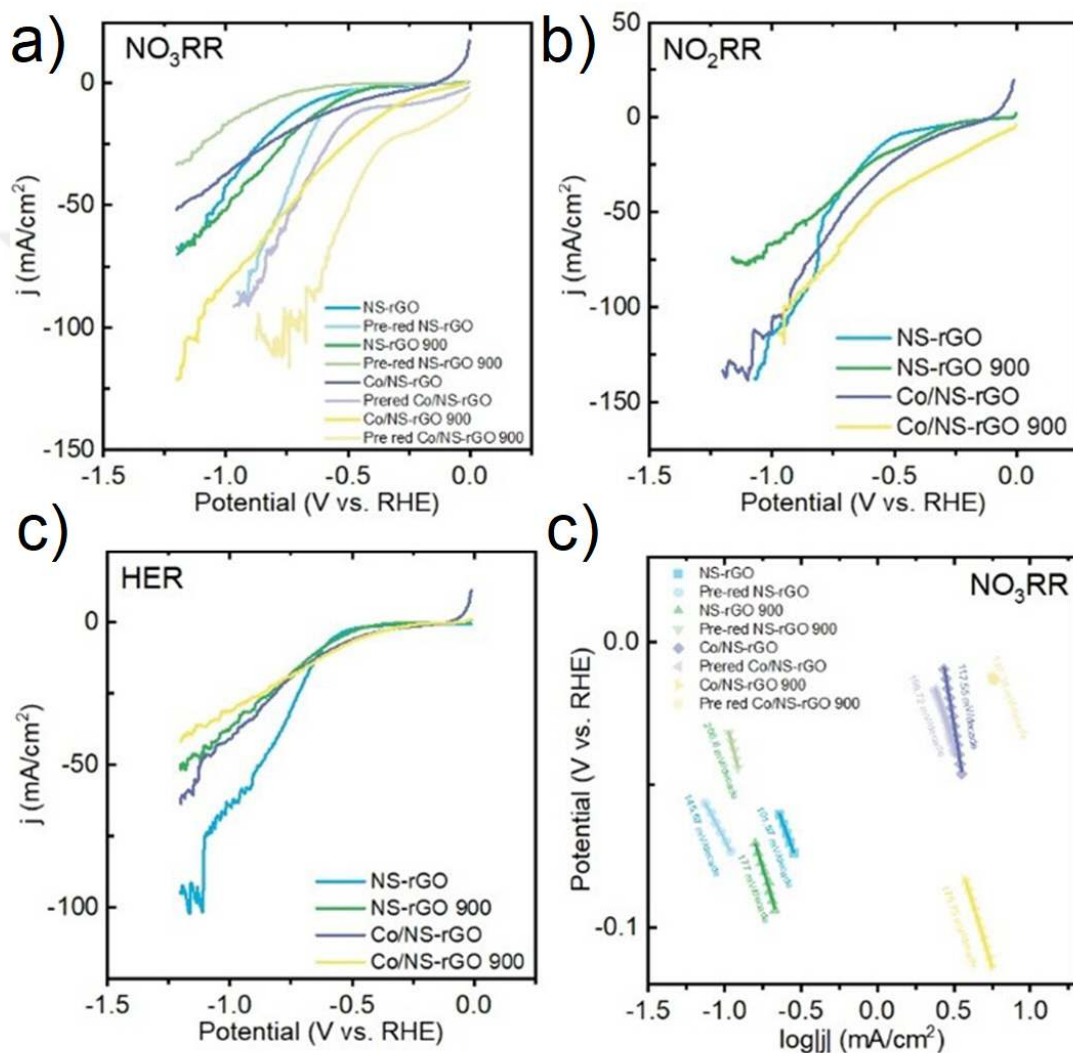


Figure 4-12: LSV Measurements. a) LSV for NO_3RR , b) LSV for NO_2RR , c) LSV for HER, d) Tafel Plot for NO_3RR

Sample	NO ₃ RR	TS	OP	NO ₂ RR	TS	OP	HER	TS	OP
NS-rGO		145.67	-0.67		161.19	-0.48		199.34	-0.54
Pre-red NS-rGO		101.87	-0.55						
NS-rGO 900		177	-0.53		115.73	-0.5		156.85	-0.53
Pre-red NS-rGO 900		206.8	-0.6						
Co/NS-rGO		317.87	-0.65		72.08	-0.27		563.75	-0.56
Pre-red Co/NS-rGO		169.72	-0.45						
Co/NS-rGO 900		175.75	-0.38		210.47	-0.52		145.03	-0.32
Pre-red Co/NS-rGO 900		107.38	-0.38						

Table 4-2: Tafel Slope (TS) Values (mV/decade) and Onset Potentials (OP) (V) of the samples for NO₃RR, NO₂RR, and HER

4.6 Proposing Two Different NO₃RR Mechanisms Through Tafel Analysis and XAS

When Tafel slope values, d-orbital occupancy, and spin state analysis from XAS are considered, one can propose two different mechanisms for Co/NS-rGO and Co/NS-rGO 900. It is observed that high d-orbital occupancy of Co/NS-rGO will repulse incoming NO₃⁻ molecules, and the HS state of Co/NS-rGO will induce weak Co-O bonds and will make the NO₃⁻ adsorption very challenging for the catalyst while ensuring better protonation properties which is in excellent agreement with Tafel analysis. On the other hand, low d-orbital occupancy and LS state of Co/NS-rGO 900 will allow better NO₃⁻ adsorption, and the catalyst will suffer from low protonation capabilities. Also, the faster kinetics of HER for Co/NS-rGO 900 shows the difficulty of protonating the sample from the Tafel analysis perspective.

4.7 NH₃ Production Performance Tests Using Constant CA

The high NO₂ yield rate and F.E. values for NS-rGO are shown by adding inductance to equivalent circuit modeling of Nyquist plots and the emergence of a new peak for inductance in DRT models and an inductive loop at EIS measurements NO₃RR after pre-reduction (Fig. 4-16) and NO₂RR [109], [158], [277], [278], [279], [280]. Fig. 4-13e shows that the electrolyte impedance (Z_e) is high for NS-rGO for NO₃RR without pre-reduction and charge transfer resistance (R_{ct}) (Fig. 4-14e) is high after pre-reduction,

which allows us to signify the inability of NS-rGO for NO_3 activation and adsorption for the former and NO_3 -to- NH_3 e^- transfer for the latter which is in line with TS analysis. The fact that Z_e values are not very high for NO_2 RR can show the NO_2 activation and adsorption ability of NS-rGO and R_{ct} values being comparably low (Fig. 4-14e) can be related to the comparably high NH_3 F.E. at more positive potentials. The decline in $\text{F.E.}_{\text{NH}_3}$ for NO_2 RR in more negative potentials may be due to the relaxation of adsorbed NO_2 species shown by the emergence of the inductive loop (Fig. 4-16) at more negative potentials.

NO_3 RR experiments for NS-rGO (Fig. 4-15 a, c) show that NH_3 production starts at -0.5 V and -0.6 V for NO_3 RR and Pre-red NO_3 RR, respectively, which is at the subsequent potential to the formation of the semicircle in EIS analysis (Fig. 4-16) and compliance with LSV onset potentials (Fig. 4-12a) for these samples. NS-rGO has the highest NO_2 yield rate and the lowest NH_3 yield rate at more negative potentials for all experiments. Also, pre-reduction facilitates NO_2 yield significantly while causing an inductive loop (Fig. 4-16), possibly due to the desorption of NO_2 . Furthermore, the highest $\text{F.E.}_{\text{NH}_3}$ of NS-rGO is observed at -0.5 V for NO_3 RR after pre-reduction, and $\text{F.E.}_{\text{NH}_3}$ is decreased significantly at more negative potentials. Even though pre-reduction enables the adsorption and activation of NO_3 , high R_{ct} (Fig. 4-14b) at lower potentials shows the inability of NS-rGO to perform NO_3 to NH_3 e^- transfer. The fact that the higher frequency peak to low-frequency peak ratio is increasing for NS-rGO with increasing potential observed in the DRT model (Fig. 4-16) further enables us to claim that the reaction becomes kinetically limited after -0.6 V for NO_3 RR with or without pre-reduction. This finding agrees with NS-rGO having the lowest NO_3 RR and pre-red NO_3 RR Tafel slope values. However, low NH_3 F.E. and yield rate for NO_3 RR allow us to claim that the reaction is 1 e^- reduction from NO_3 to NO_2 rather than 8 e^- transfer for NO_3 to NH_3 electroreduction.

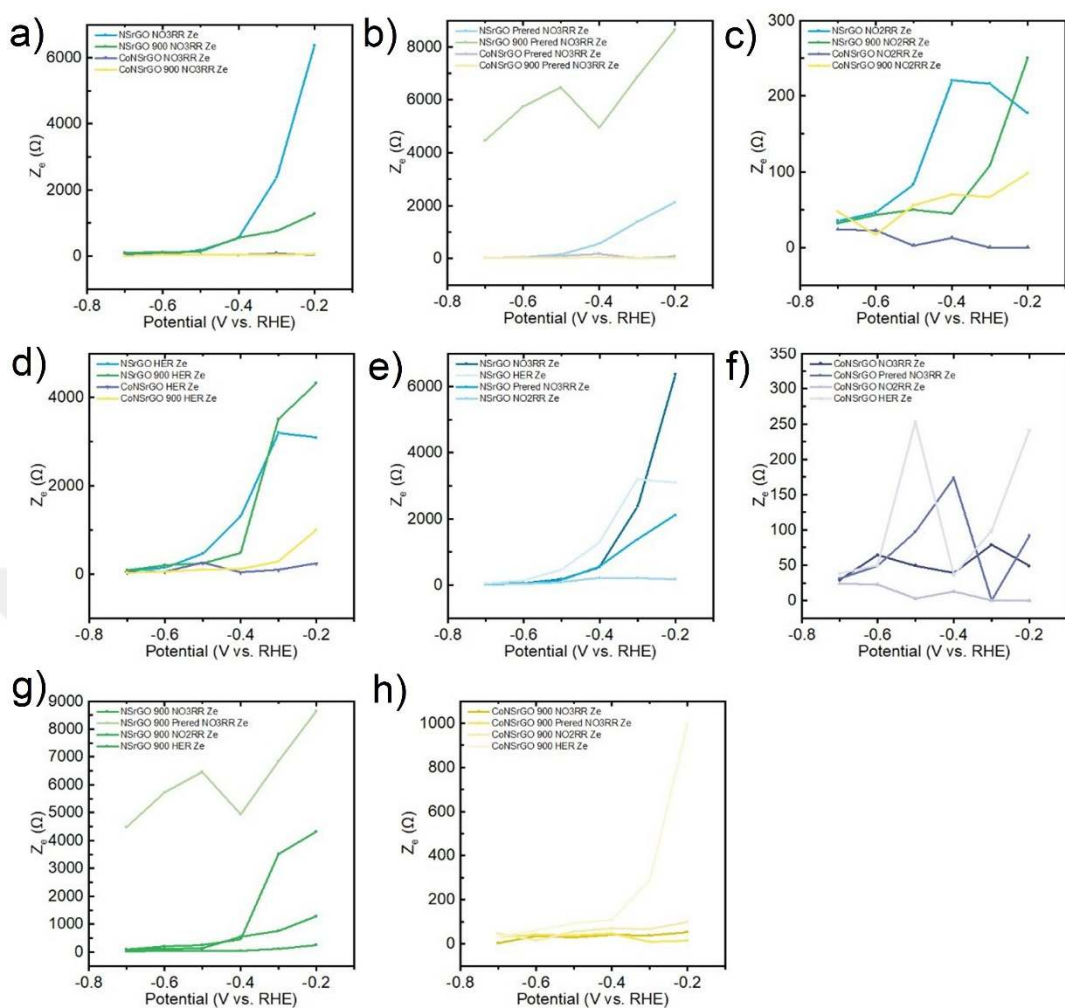


Figure 4-13: Electrolyte impedances (Z_e) for a) NO_3RR , b) Pre-red NO_3RR , c) NO_2RR , d) HER, e) NS-rGO, f) NS-rGO 900, g) Co/NS-rGO, h) Co/NS-rGO 900

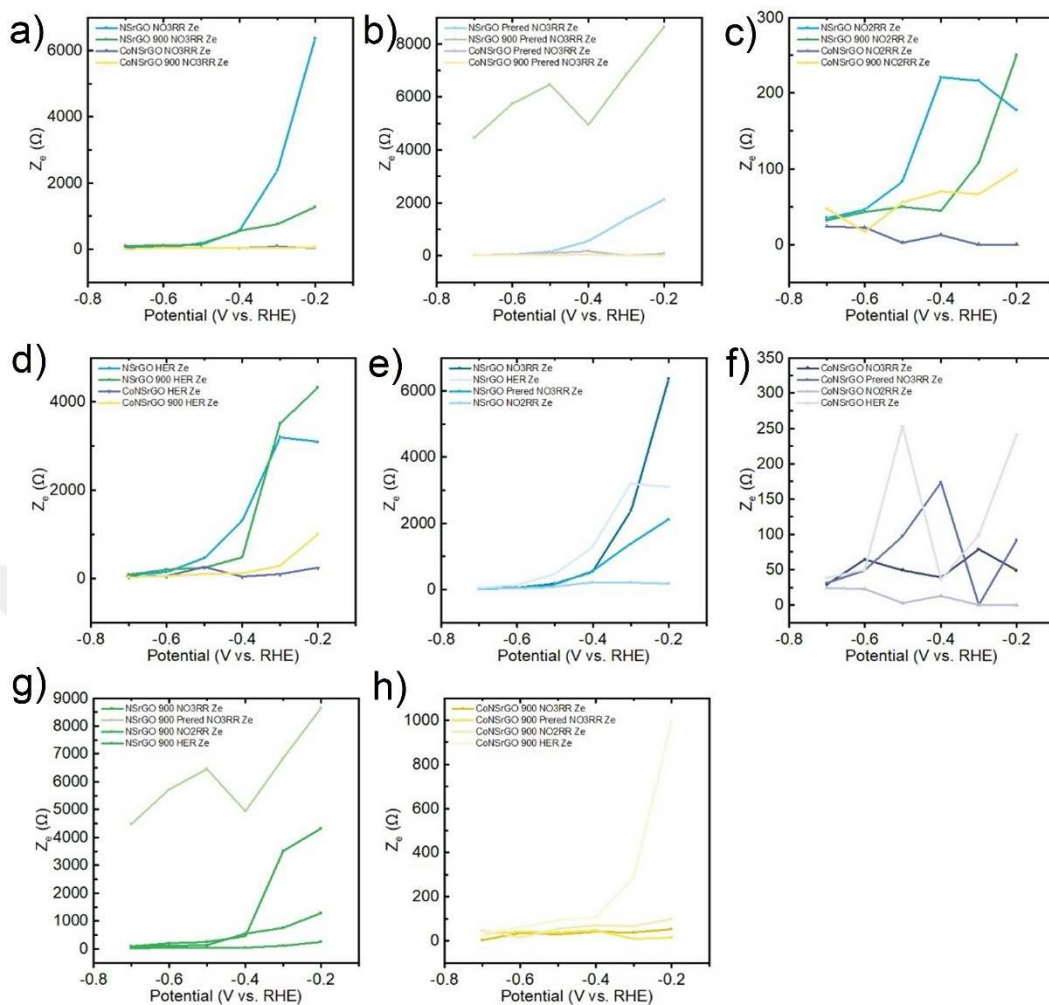


Figure 4-14: Charge Transfer Resistances (R_{ct}) for a) NO_3RR , b) Pre-red NO_3RR , c) NO_2RR , d) HER, e) NS-rGO, f) NS-rGO 900, g) Co/NS-rGO, h) Co/NS-rGO 900

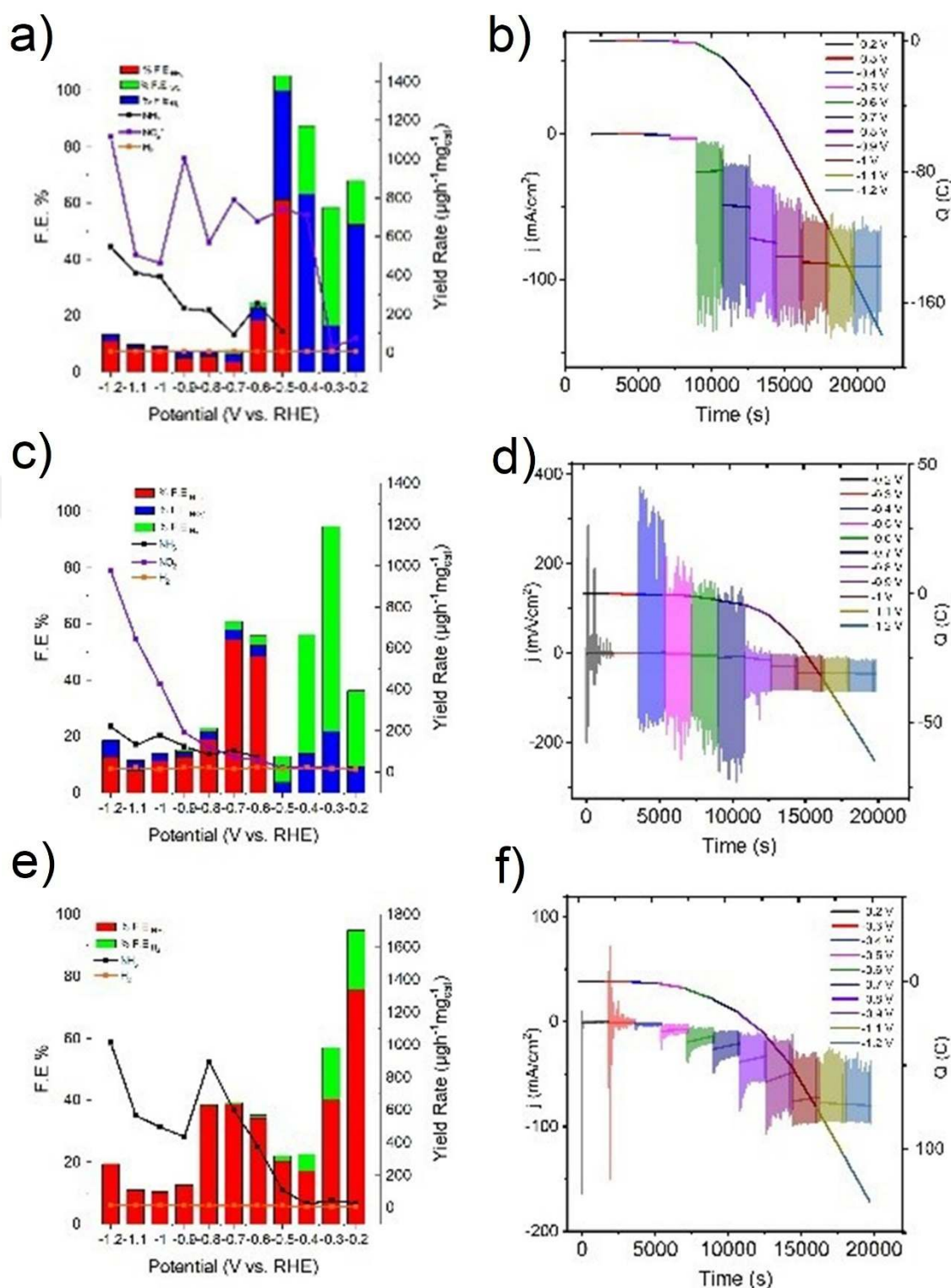


Figure 4-15: Nitrate to ammonia electrosynthesis activity of NS-rGO. a) NO₂⁻, NH₃, H₂ F.E., and Y.R. of NS-rGO after pre-reduction at -1.5 V in 0.1 M KOH + 0.1 M KNO₃. b) j and q values for NS-rGO NO₃RR after pre-reduction at -1.5 V. c) NO₂⁻, NH₃, H₂ F.E. and Y.R. of NS-rGO in 0.1 M KOH + 0.1 M KNO₃ d) j and q values for NS-rGO NO₃RR e) NO₂⁻, NH₃, H₂ F.E. and Y.R. of NS-rGO in 0.1 M KOH + 0.1 M KNO₂ f) j and q values for NS-rGO NO₂RR

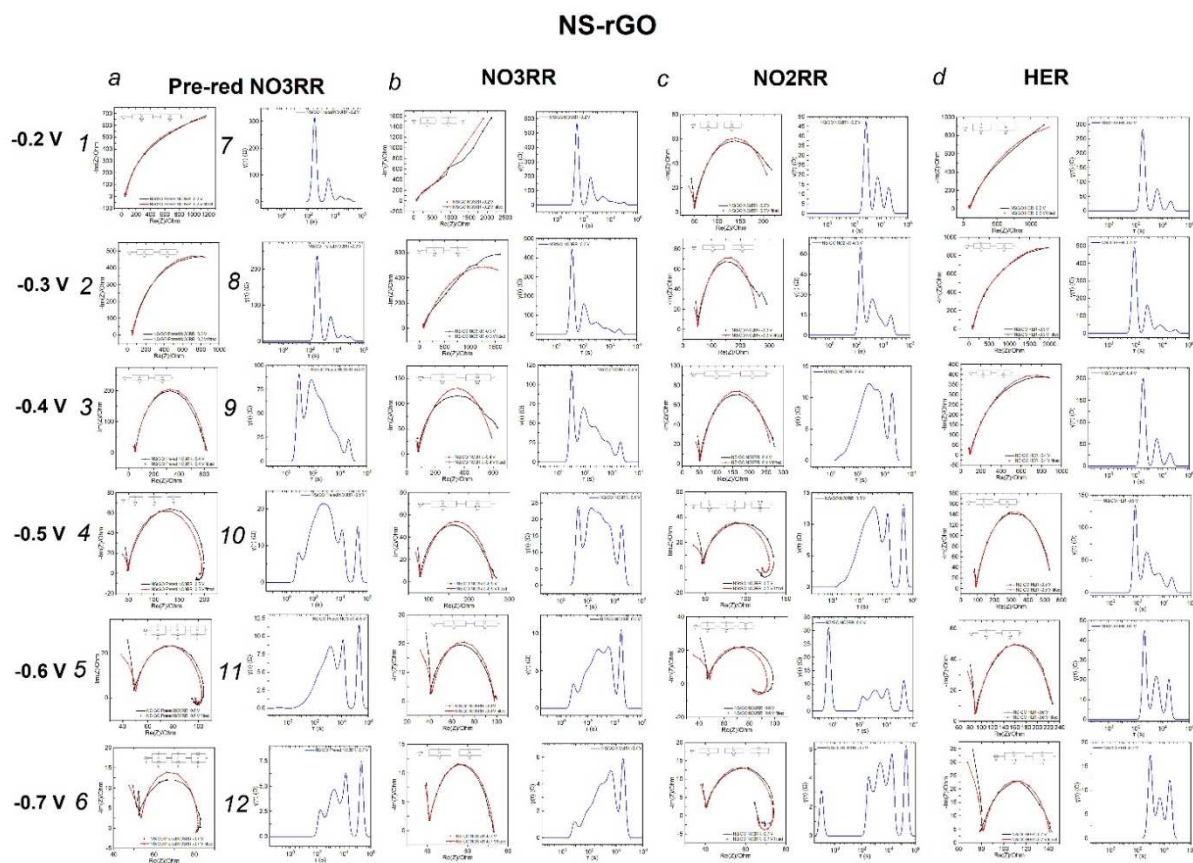


Figure 4-16: NS-rGO EIS and DRT Models

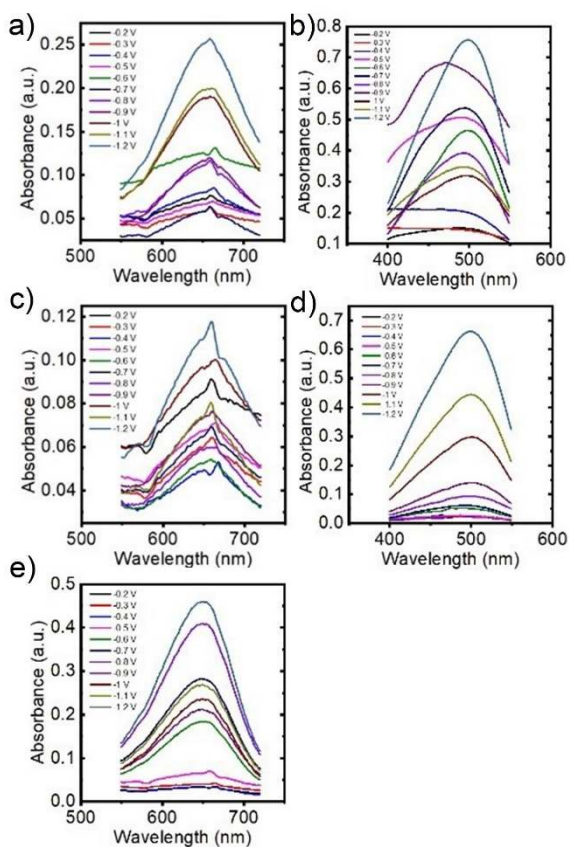


Figure 4-17: UV-vis spectroscopy results after NO₃RR and NO₂RR experiments. a) NH₃ quantification using the Indophenol Blue Method for NS-rGO NO₃RR after pre-reduction at -1.5 V. b) NO₂ quantification using the Griess Test for NS-rGO after pre-reduction at -1.5 V. c) NH₃ quantification using the Indophenol Blue Method for NS-rGO NO₃RR without pre-reduction at -1.5 V. d) NO₂ quantification using the Griess Test for NS-rGO without pre-reduction at -1.5 V. e) NH₃ quantification using the Indophenol Blue Method for NS-rGO NO₂RR.

High TS values for NO₃RR and NO₂RR indicate that NS-rGO 900 follows an adsorption-limited mechanism (Table 4-2). Surprisingly, the activity of the NS-rGO 900 catalyst is diminished after pre-reduction, where an increase in both overpotential and NO₃RR TS is observed (Fig. 4-12a). Also, HER TS values indicate faster kinetics for pyrolyzed samples as of NS-rGO 900 having 199.34 mV/decade. Superior HER kinetics are also observed in high H₂ yield in NH₃ activity measurements (Fig. 4-18). The highest Z_e (Fig. 4-13f) values for NS-rGO, among other samples, especially after pre-reduction, signify the mass-transfer limited NO₃RR activity of the catalyst. Also, the highest R_{ct} values (Fig. 4-14f) show that the subsequent charge transfer has inferior properties, which explains the low NH₃ activity of the catalyst suppressed by superior HER. High peak intensity of low frequencies in the DRT Model and the formation of semicircles in later potentials also emphasize the catalyst's inferior NO₃RR and NO₂RR activity (Fig. 7).

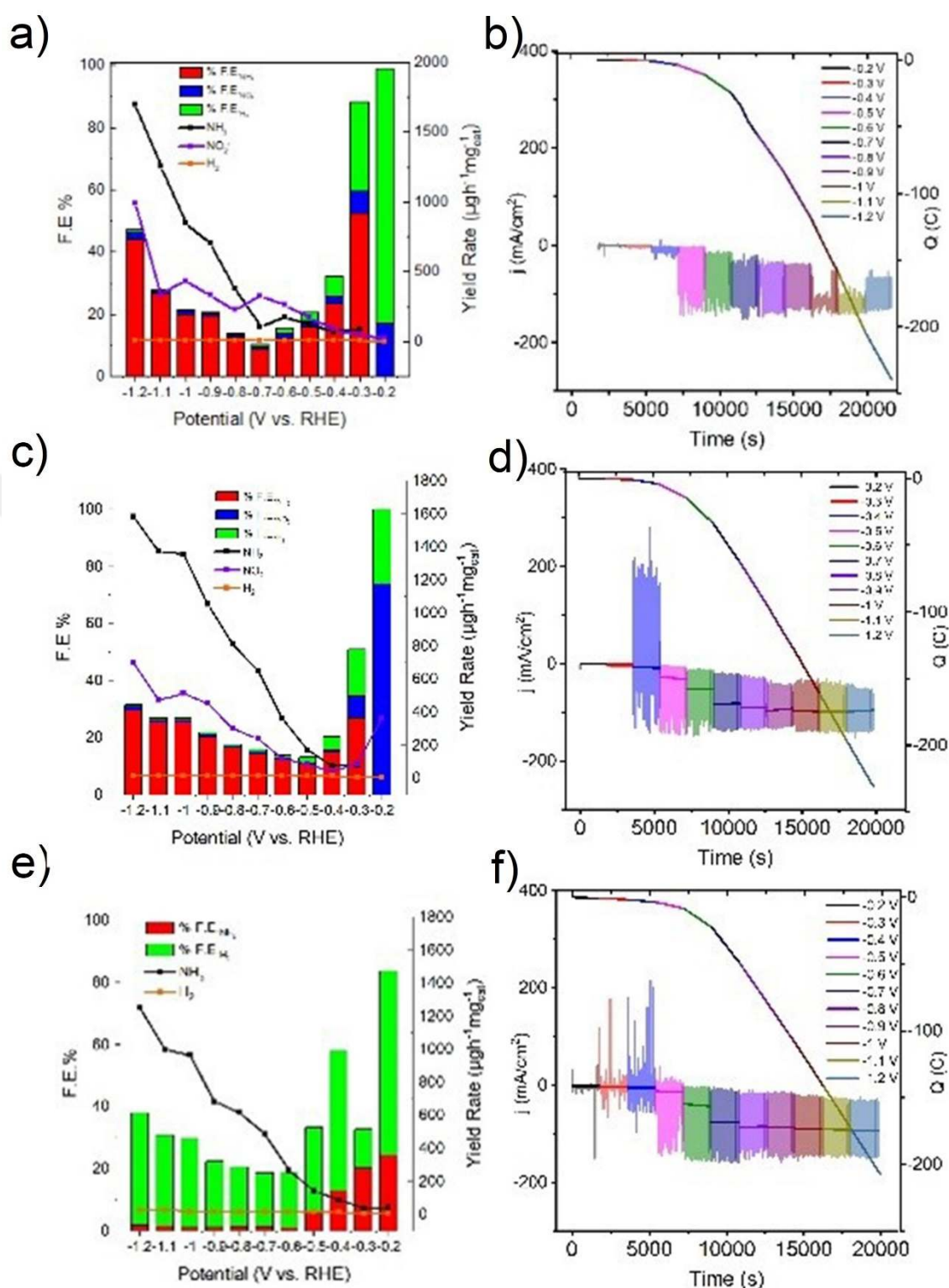


Figure 4-18: Nitrate to ammonia electrosynthesis activity of NS-rGO 900. a) NO₂⁻, NH₃, H₂ F.E., and Y.R. of NS-rGO 900 after pre-reduction at -1.5 V in 0.1 M KOH + 0.1 M KNO₃. b) j and q values for NS-rGO 900 NO₃RR after pre-reduction at -1.5 V. c) NO₂⁻, NH₃, H₂ F.E. and Y.R. of NS-rGO 900 in 0.1 M KOH + 0.1 M KNO₃. d) j and q values for NS-rGO 900 NO₃RR. e) NO₂⁻, NH₃, H₂ F.E. and Y.R. of NS-rGO 900 in 0.1 M KOH + 0.1 M KNO₃. f) j and q values for NS-rGO 900 NO₂RR.

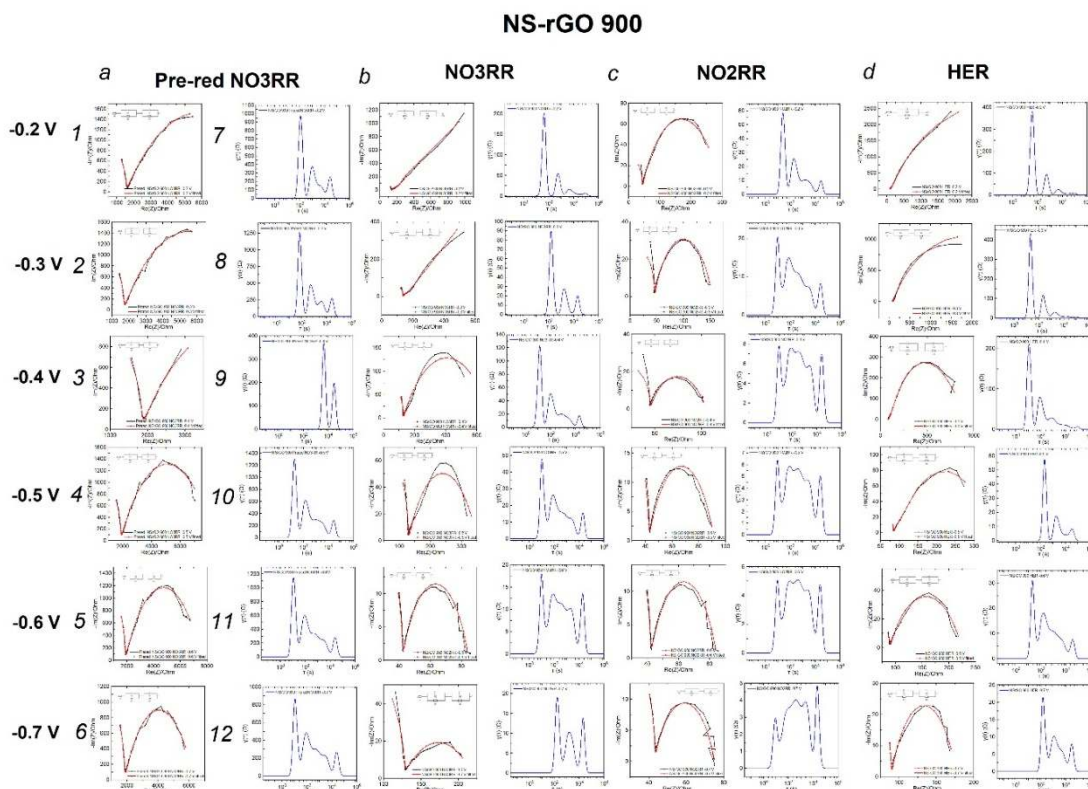


Figure 4-19: DRT and EIS analysis for NS-rGO 900

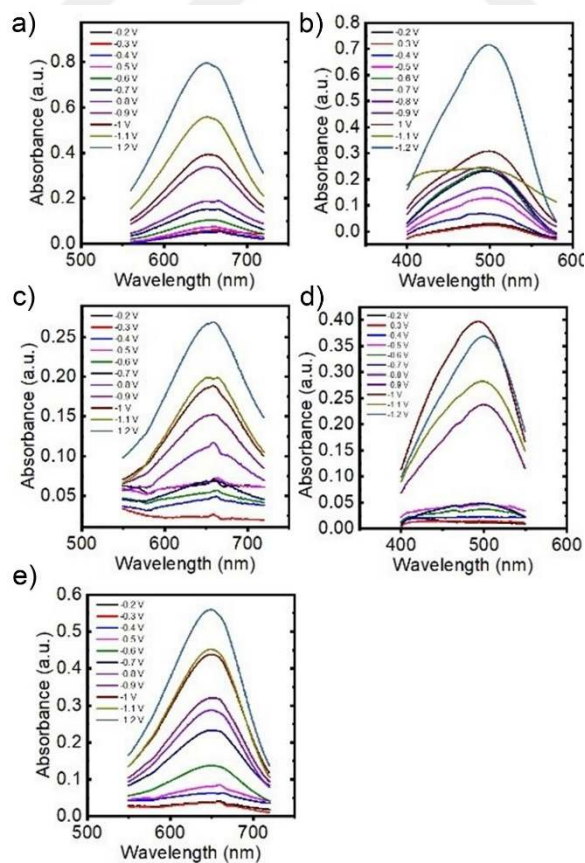


Figure 4-20: UV-vis spectroscopy results after NO₃RR and NO₂RR experiments. a) NH₃ quantification using the Indophenol Blue Method for NS-rGO 900 NO₃RR after pre-reduction at -1.5 V. b) NO₂ quantification using the Griess Test for NS-rGO 900 after pre-reduction at -1.5 V. c) NH₃ quantification using the Indophenol Blue Method for NS-rGO 900 NO₃RR without pre-reduction at -1.5 V. d) NO₂ quantification using the Griess Test for NS-rGO 900 without pre-reduction at -1.5 V. e) NH₃ quantification using the Indophenol Blue Method for NS-rGO 900 NO₂RR.

Even though NO₃RR TS is significantly lowered after the pre-reduction from 317.87 to 169.72 mV/decade, high TS values for Co/NS-rGO indicate a mass-transport-limited NO₃RR mechanism (Table 4-2). Furthermore, the highest HER TS of 563.75 mV/decade is achieved for Co/NS-rGO, indicating suppressed HER activity. This agrees with the low H₂ yield obtained from NH₃ activity measurements and XAS data, which shows that Co/NS-rGO facilitates protonation better. It is observed that the pre-reduction lowers the potential for NH₃ F.E. maxima from -0.9 V to -0.4 V (Fig. 4-21a). Also, the highest F.E._{NH₃} for Co/NS-rGO in the pre-reduced case is achieved at NO₃RR onset potential, while without pre-reduction, on top of a -200 mV increase in overpotential, the catalyst does not show its NH₃ F.E. onset potential, possibly due to the absence of a well-defined kinetic, mixed-kinetic and mass-transfer limited region (Fig. 4-12a). Since the NO₃RR for Co/NS-rGO is limited by the NO₃ adsorption, lower Z_e values (Fig. 4-13g) for Co/NS-rGO than NS-rGO can explain the increase in NH₃ production activity with Co incorporation.

As TS results suggest, the high peak intensity ratio between high-frequency and low-frequency peaks in the DRT model indicates that NO₃RR and HER reactions signify the adsorption-driven mechanism of Co/NS-rGO (Fig. 4-22). It is reported that high-frequency inductance behavior is associated with the relaxation of surface coverages or species desorption[158]. The same phenomenon is observed for pre-red NO₃RR starting from -0.5 V, increasing inductive loop sizes in EIS, and the emergence of a new peak in the DRT model. After pre-reduction, a 2-fold increase in NO₂ yield rate and higher NO₂ F.E. values allows us to claim the possibility of NO₂ being the desorbed species. Pre-reduction lowered the potential for forming the semicircle associated with the NO₃RR charge transfer. EIS and DRT behavior after pre-reduction can point out a change in the NO₃RR mechanism. The increase in NH₃ yield after pre-reduction may be explained by the more facile reduction of desorbed NO₂ at the next more negative potential in a mass-transfer-limited fashion owing to superior NO₂RR properties of Co/NS-rGO.

We see an exceptional NO₂RR activity for Co/NS-rGO in NH₃ yield and F.E. values (Fig. 4-21). Similar superior NO₂RR activity of Co catalysts is reported in the literature, too. Very high NH₃ F.E. values are observed for Co/NS-rGO throughout the studied potential range, and we see a near 100% F.E. at more negative potentials owing to the superior protonation ability of Co/NS-rGO. Surprisingly, the low NO₂RR TS of Co/NS-rGO 72.08 mV/decade shows that neither NO₂ adsorption nor charge transfer processes are RDS but a chemical step. Tafel analysis signifies the facile adsorption and e⁻ transfer processes for Co/NS-rGO. Furthermore, a significantly high HER TS of 563.75 mV/decade aligns with a low H₂ yield rate and F.E. observed for NO₃RR and NO₂RR experiments. This finding shows that HER cannot suppress NO₃RR and NO₂RR, and protonation is highly facilitated for the proposed mechanism. For NO₂RR, the inductive behavior starts at lower frequencies, which can be attributed to electrical connection disturbances or EDL interactions[281]. Because low-frequency behavior is only observed for the NO₂RR experiment of Co/NS-rGO, we claim that the behavior is due to EDL contributions induced by NO₂⁻. Furthermore, the affinity of NO₂⁻ species to interact with Co/NS-rGO EDL can indicate the facile movement of NO₂ along EDL, supporting the easier NO₂ desorption on EDL.

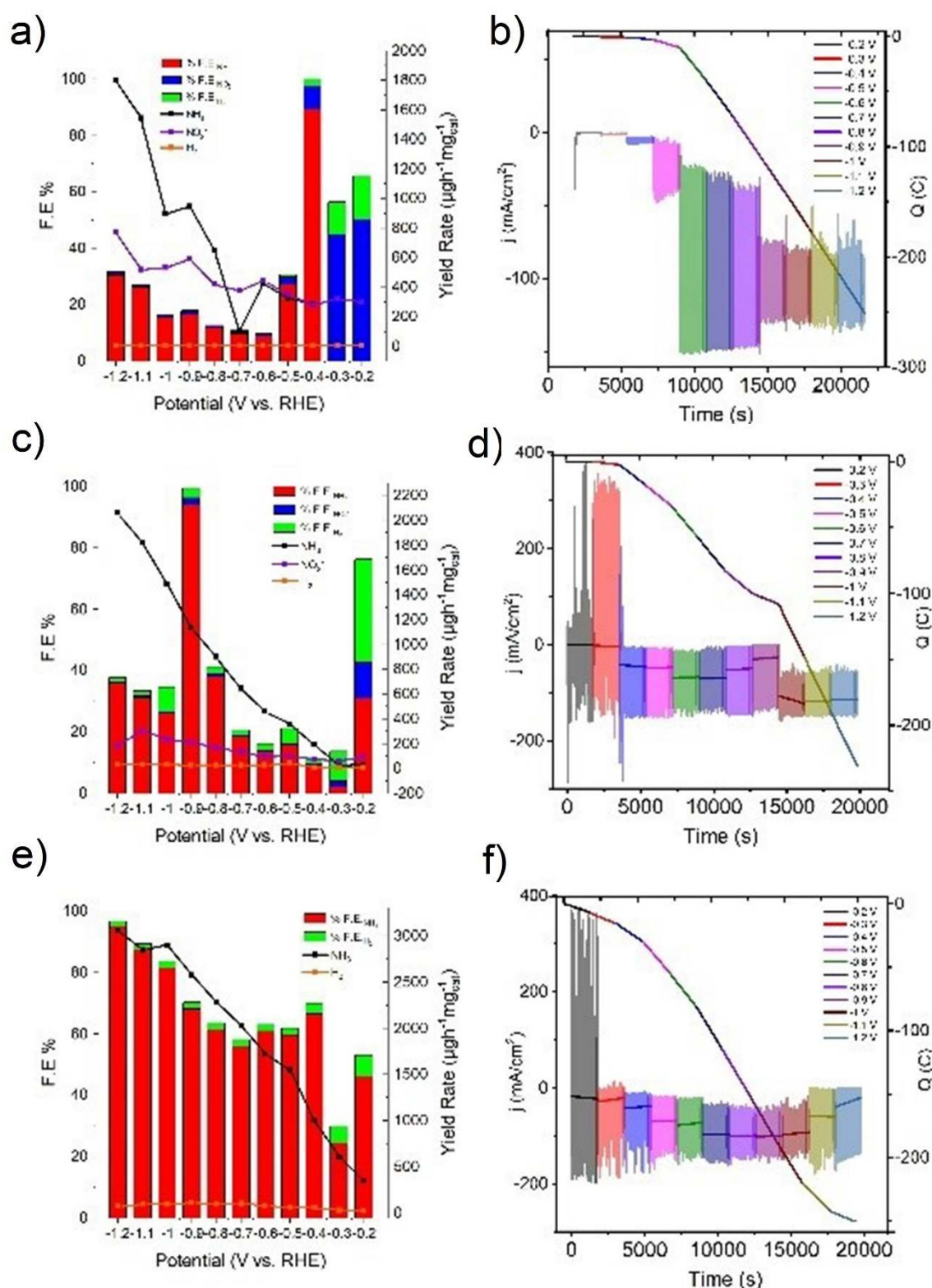


Figure 4-21: Nitrate to ammonia electrosynthesis activity of Co/NS-rGO. a) NO₂⁻, NH₃, H₂ F.E., and Y.R. of Co/NS-rGO after pre-reduction at -1.5 V in 0.1 M KOH + 0.1 M KNO₃. b) j and q values for Co/NS-rGO NO₃RR after pre-reduction at -1.5 V. c) NO₂⁻, NH₃, H₂ F.E. and Y.R. of Co/NS-rGO in 0.1 M KOH + 0.1 M KNO₃ d) j and q values for Co/NS-rGO NO₃RR e) NO₂⁻, NH₃, H₂ F.E. and Y.R. of Co/NS-rGO in 0.1 M KOH + 0.1 M KNO₂ f) j and q values for Co/NS-rGO NO₂RR

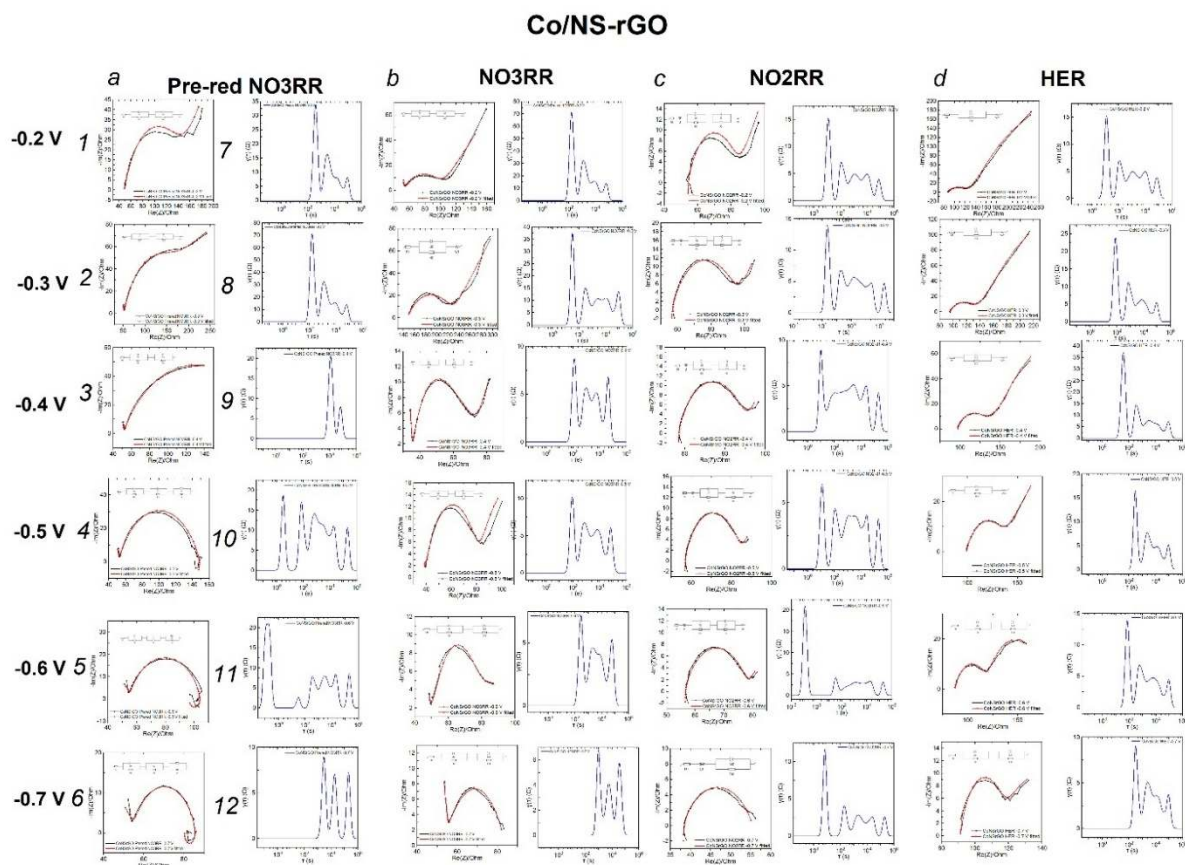


Figure 4-22: Potential dependent EIS and DRT analysis for Co/NS-rGO

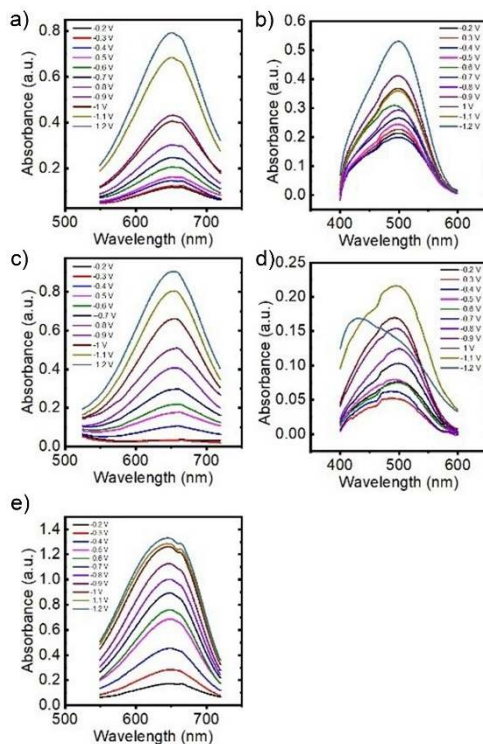


Figure 4-23: UV-vis spectroscopy results after NO₃RR and NO₂RR experiments. a) NH₃ quantification using the Indophenol Blue Method for Co/NS-rGO NO₃RR after pre-reduction at -1.5 V. b) NO₂ quantification using the Griess Test for Co/NS-rGO after pre-reduction at -1.5 V. c) NH₃ quantification using the Indophenol Blue Method for Co/NS-rGO NO₃RR without pre-reduction at -1.5 V. d) NO₂ quantification using the Griess Test for Co/NS-rGO without pre-reduction at -1.5 V. e) NH₃ quantification using the Indophenol Blue Method for Co/NS-rGO NO₂RR.

TS values of Co/NS-rGO 900 indicate that after pre-reduction, the reaction mechanism becomes NO₃-NO₂ 1e⁻ transfer from being adsorption limited (Table 4-2). Also, high NO₂RR TS and high overpotential indicate a NO₂ adsorption-limited mechanism and explain the catalyst's inferior NO₂RR activity (Fig. 4-12b). Furthermore, low HER TS indicates faster kinetics, which agrees with the high H₂ yield in NO₃RR experiments (Fig. 4-24). Also, TS findings align with the proposed mechanism inspired by XAS analysis where facile NO₃ adsorption properties for pre-red NO₃RR and low protonation properties due to fast HER kinetics. On the other hand, low Z_e (Fig. 4-13h) and R_{ct} (Fig. 4-14h) values for NO₃RR and high values for corresponding indicators for NO₂RR support our claim for the Co/NS-rGO 900 activity. NH₃ electrosynthesis activity measurements (Fig. 4-24) show the high NH₃ F.E. and yield rates, the positive effect of pre-reduction in lowering the potential for NH₃ production maxima, high H₂ yield, and inferior NO₂RR activity of the catalyst. EIS analysis (Fig. 4-25) shows a first bump in both NO₃RR and NO₂RR experiments, indicating the emergence of a semicircle for the former and a capacitive behavior for the latter. Different from other samples, instead of two peaks, adsorption and charge transfer processes are carried out in nearly the same semicircle with lower time constants. This finding shows the facile NO₃RR adsorption properties and charge-transfer limitation of the catalyst where NO₃-NO₂ e⁻ transfer is carried out easier, demonstrated by the formation of a semicircle but O₂-NO e⁻ transfer being not favored due to the absence of a semicircle. DRT model supports this claim by demonstrating higher frequency peaks and singular peaks in later potentials.

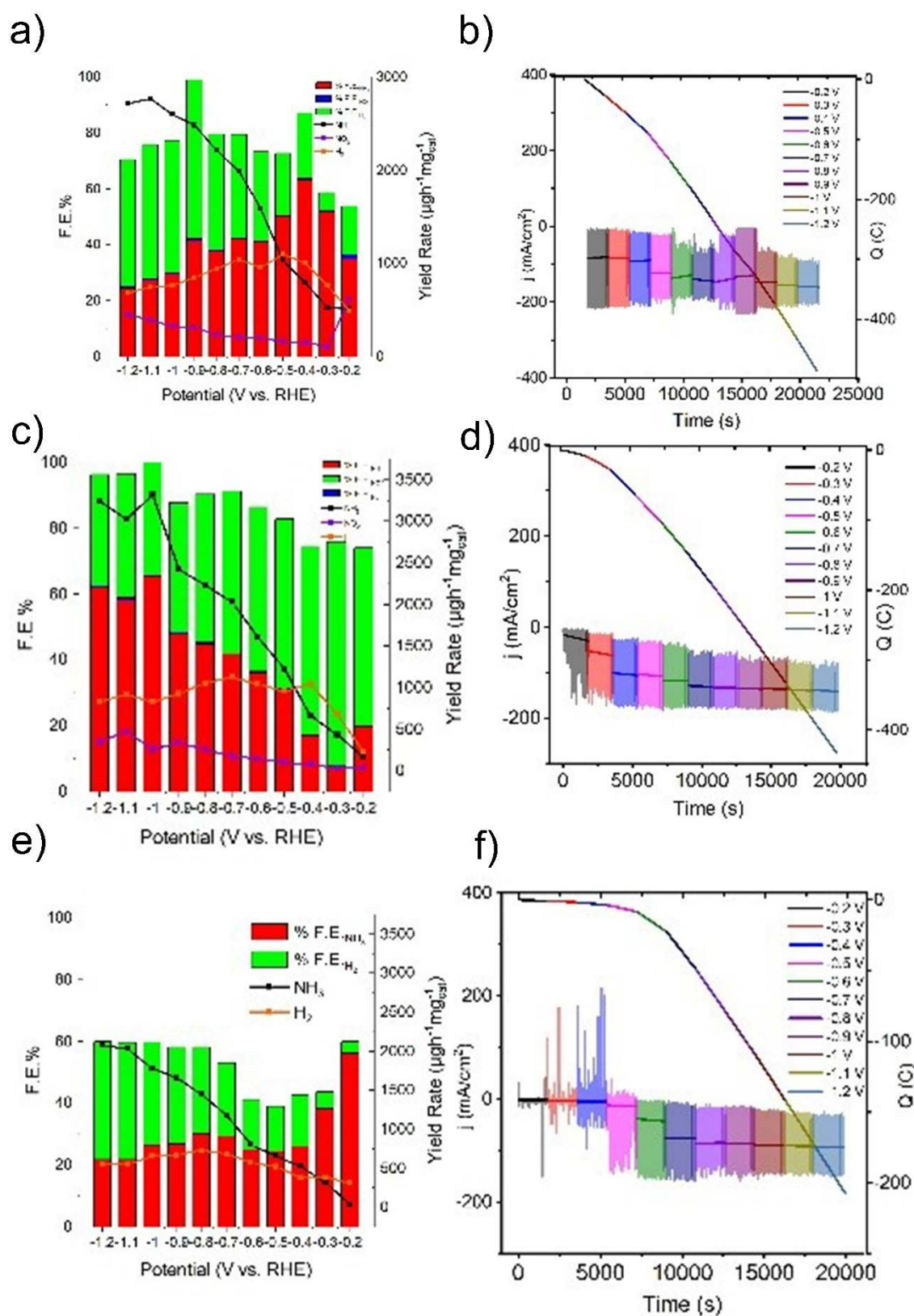


Figure 4-24: Nitrate to ammonia electrosynthesis activity of Co/NS-rGO 900. a) NO_2^- , NH_3 , H_2 F.E., and Y.R. of Co/NS-rGO 900 after pre-reduction at -1.5 V in 0.1 M KOH + 0.1 M KNO_3 . b) j and q values for Co/NS-rGO 900 NO_3RR after pre-reduction at -1.5 V. c) NO_2^- , NH_3 , H_2 F.E. and Y.R. of Co/NS-rGO 900 in 0.1 M KOH + 0.1 M KNO_3 d) j and q values for

Co/NS-rGO 900 NO₃RR e) NO₂⁻, NH₃, H₂ F.E. and Y.R. of Co/NS-rGO 900 in 0.1 M KOH + 0.1 M KNO₂ f) j and q values for Co/NS-rGO 900 NO₂RR

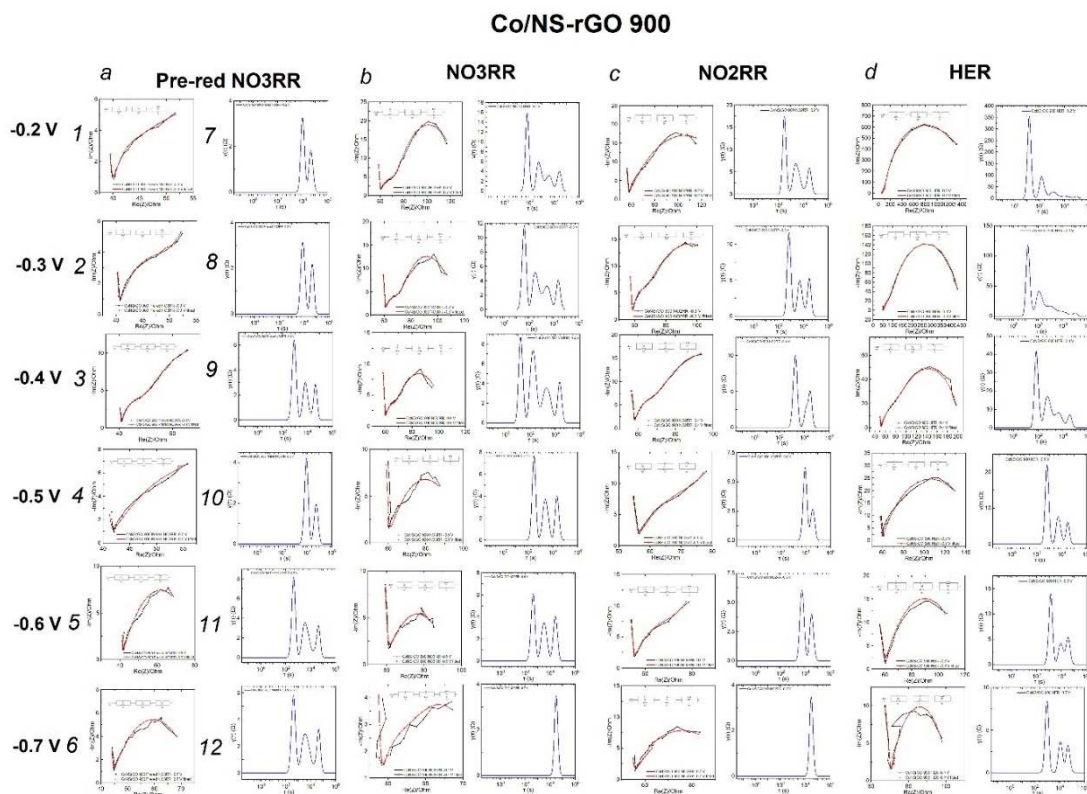


Figure 4-25: DRT and EIS analysis for Co/NS-rGO 900

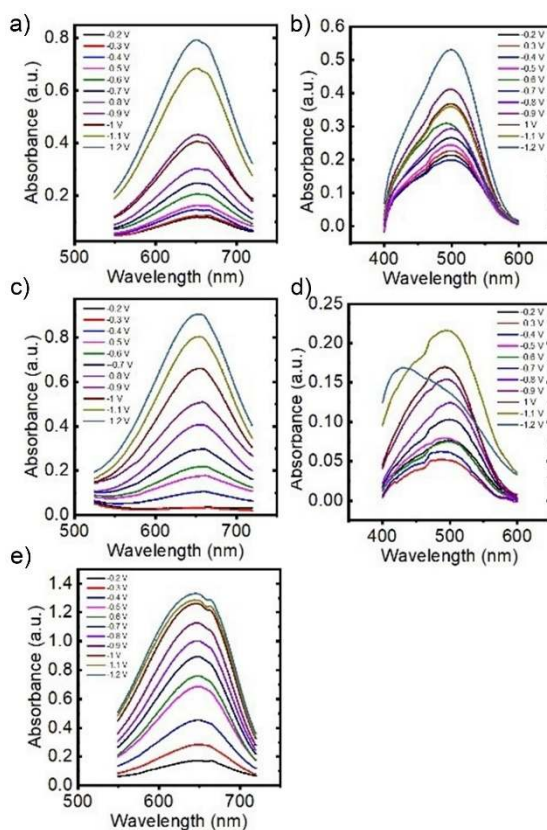


Figure 4-26: UV-vis spectroscopy results after NO₃RR and NO₂RR experiments. a) NH₃ quantification using the Indophenol Blue Method for Co/NS-rGO 900 NO₃RR after pre-reduction at -1.5 V. b) NO₂ quantification using the Griess Test for Co/NS-rGO 900 after pre-reduction at -1.5 V. c) NH₃ quantification using the Indophenol Blue Method for Co/NS-rGO 900 NO₃RR without pre-reduction at -1.5 V. d) NO₂ quantification using the Griess Test for Co/NS-rGO 900 without pre-reduction at -1.5 V. e) NH₃ quantification using the Indophenol Blue Method for Co/NS-rGO 900 NO₂RR.

4.8 NH₃ Production Performance Tests Using Pulsed CA

Considering the high NH₃ yield rates and F.E.s of Co/NS-rGO for NO₂RR, which is also reported in the literature, the observed mass transport limitations, to further investigate the actual behavior of adsorption limitations and better protonation properties of proposed mechanism and presence of NO₂⁻ desorption, a pulse strategy is proposed in both mass transfer limited and charge transfer dominated potentials of -0.4 V and -0.9 V consecutively. Also, the pulse method is utilized to examine the effect of more positive or negative potentials than the studied potential on NO₃RR activity through changes in the NO₃ concentration on the electrode. Furthermore, as suggested in other studies, the presence of a tandem NO₃RR electrolysis is investigated through a pulsed strategy owing to the desorption of NO₂⁻ species, surprisingly low Tafel Slope for Co/NS-rGO indicating a chemical step, and uniquely high NO₂⁻ to NH₃ conversion activity of Co/NS-rGO. Also, the behavior of non-metal and pyrolyzed metal-loaded catalysts is investigated to examine the mechanism of different catalysts.

Pulsed NO₃RR for NS-rGO (Fig. 4-27) is carried out to drive the reaction into becoming more mass-transfer limited by applying pulses at -0.4 V and the studied potential for 2 s each. In pre-reduced NO₃RR, NH₃ production starts at a later potential, possibly due to the more intensified mass-transport limitations or less NO₃ concentration on the electrode. Furthermore, the NO₂ F.E. nearly becomes zero when NH₃ F.E. increases owing to the consecutive reduction of desorbed NO₂ species identified in EIS analysis for pre-reduced NO₃RR through pulsed reduction. For NO₃RR, overall NH₃ and NO₂ F.E. is increased while H₂ F.E. is diminished. NO₂RR activity measurements show that for pulsed measurements, F.E._{NH3} increases at less negative potentials and decreases at more negative potentials due to mass-transfer properties being more dominant at more positive potentials.

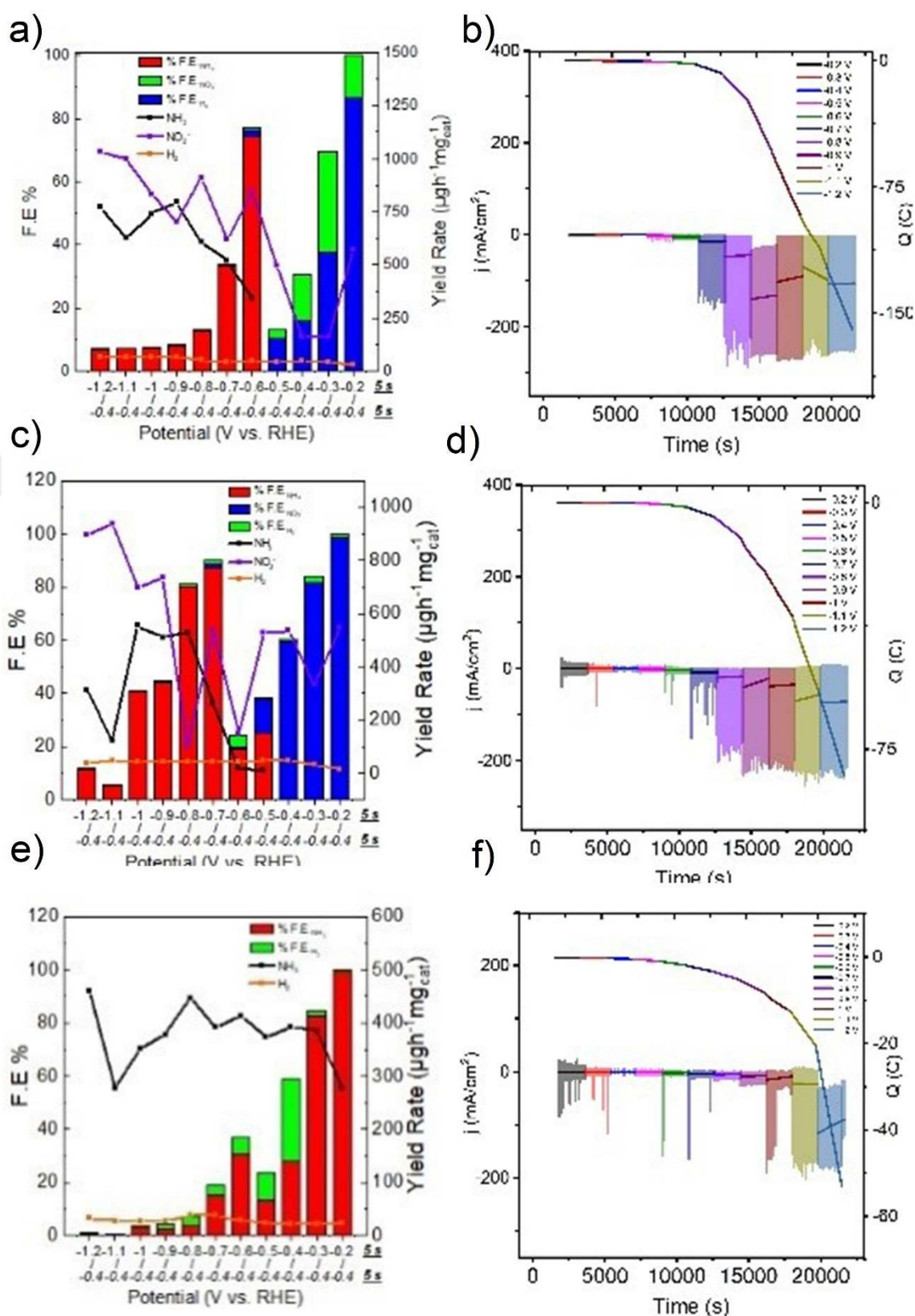


Figure 4-27: Pulsed Nitrate to ammonia electrosynthesis activity of NS-rGO for 2 s at constant electrolysis potentials and pulsed potential of -0.4 V. a) NO_2^- , NH_3 , H_2 F.E., and Y.R. of NS-rGO after pre-reduction at -1.5 V in 0.1 M KOH + 0.1 M KNO_3 . b) j and q values for NS-

rGO NO₃RR after pre-reduction at -1.5 V. c) NO₂⁻, NH₃, H₂ F.E. and Y.R. of NS-rGO in 0.1 M KOH + 0.1 M KNO₃ d) j and q values for NS-rGO NO₃RR e) NO₂⁻, NH₃, H₂ F.E. and Y.R. of NS-rGO in 0.1 M KOH + 0.1 M KNO₂ f) j and q values for NS-rGO NO₂RR

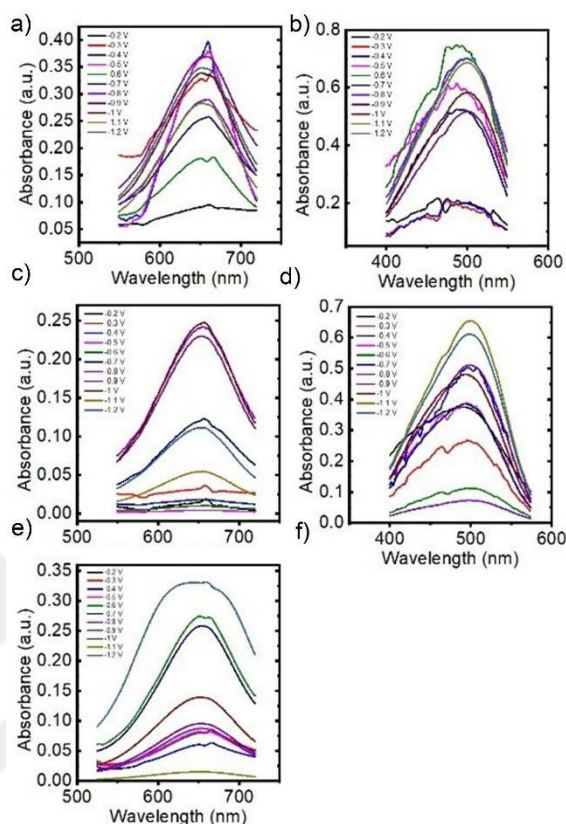


Figure 4-28: UV-vis spectroscopy results after pulsed NO₃RR and NO₂RR experiments at constant electrolysis potentials and pulsed potential of -0.4 V for 2 s. a) NH₃ quantification using the Indophenol Blue Method for NS-rGO NO₃RR after pre-reduction at -1.5 V. b) NO₂ quantification using the Griess Test for NS-rGO after pre-reduction at -1.5 V. c) NH₃ quantification using the Indophenol Blue Method for NS-rGO NO₃RR without pre-reduction at -1.5 V. d) NO₂ quantification using the Griess Test for NS-rGO without pre-reduction at -1.5 V. e) NH₃ quantification using the Indophenol Blue Method for NS-rGO NO₂RR.

Higher NH₃ F.E. values are obtained for pre-reduced Co/NS-rGO NO₃RR (Fig. 4-29) pulsed electroreduction experiments at -0.4 V, possibly due to the consecutive reduction of the desorbed NO₂ species. Furthermore, it is observed that at more negative potentials, F.E._{NH3} values steadily increase owing to the preservation of the material's mass-transfer-limited properties and superior protonation assets. Also, for NO₃RR, F.E._{NH3} is increased compared to constant CA measurements throughout the potential range except for the exceptionally high performance at -0.9 V NO₃RR of Co/NS-rGO. However, it is observed that the pulsed CA at -0.4 V for 2 s does not promote constant NO₂RR, possibly due to no adsorption limitations for NO₂RR.

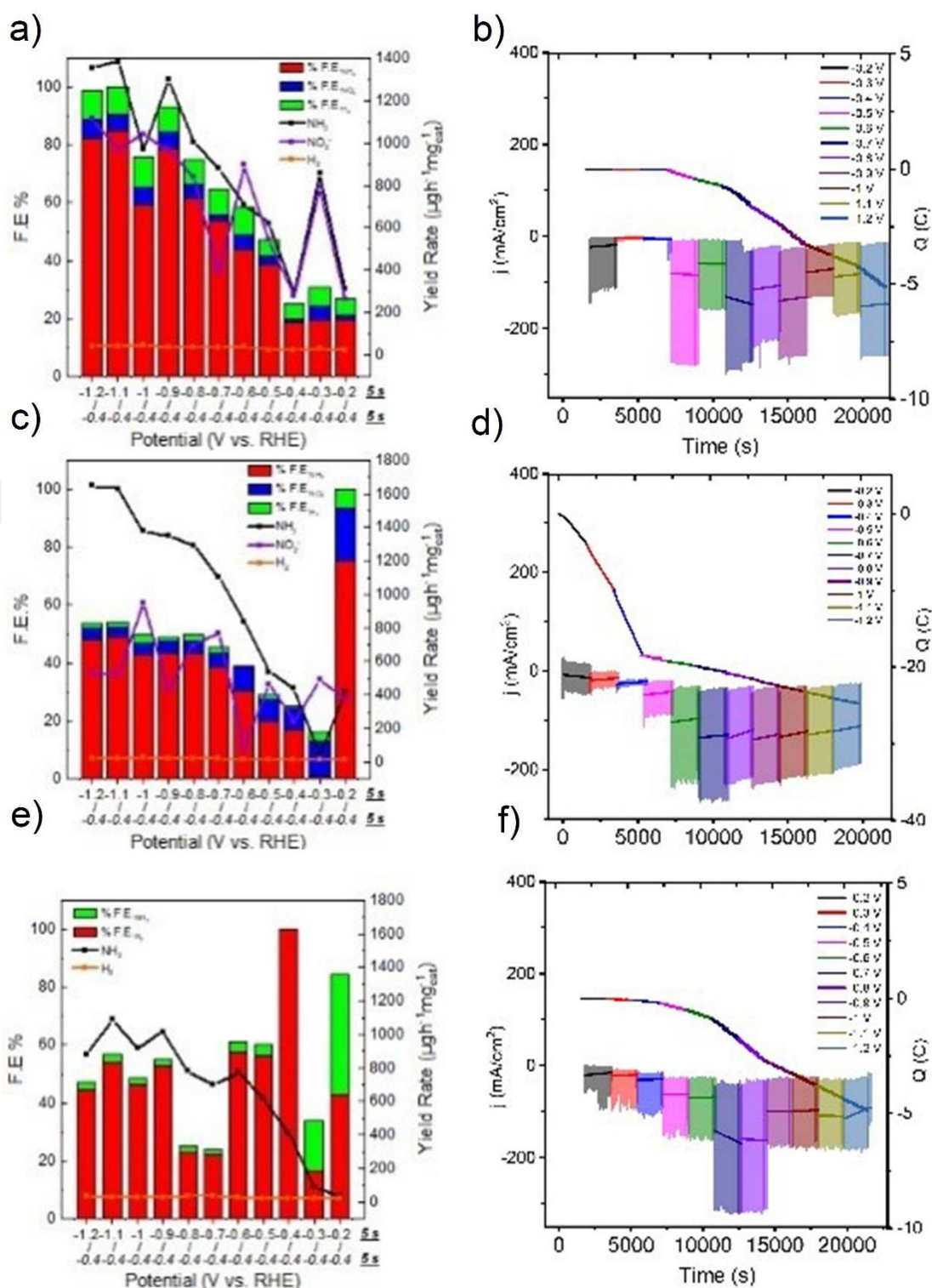


Figure 4-29: Pulsed Nitrate to ammonia electrosynthesis activity of Co/NS-rGO for 2 s at constant electrolysis potentials and pulsed potential of -0.4 V. a) NO_2^- , NH_3 , H_2 F.E., and Y.R. of Co/NS-rGO after pre-reduction at -1.5 V in 0.1 M KOH + 0.1 M KNO_3 . b) j and q values for Co/NS-rGO NO_3RR after pre-reduction at -1.5 V. c) NO_2^- , NH_3 , H_2 F.E. and Y.R. of Co/NS-rGO in 0.1 M KOH + 0.1 M KNO_3 d) j and q values for Co/NS-rGO NO_3RR e) NO_2^- , NH_3 , H_2 F.E. and Y.R. of Co/NS-rGO in 0.1 M KOH + 0.1 M KNO_2 f) j and q values for Co/NS-rGO NO_2RR

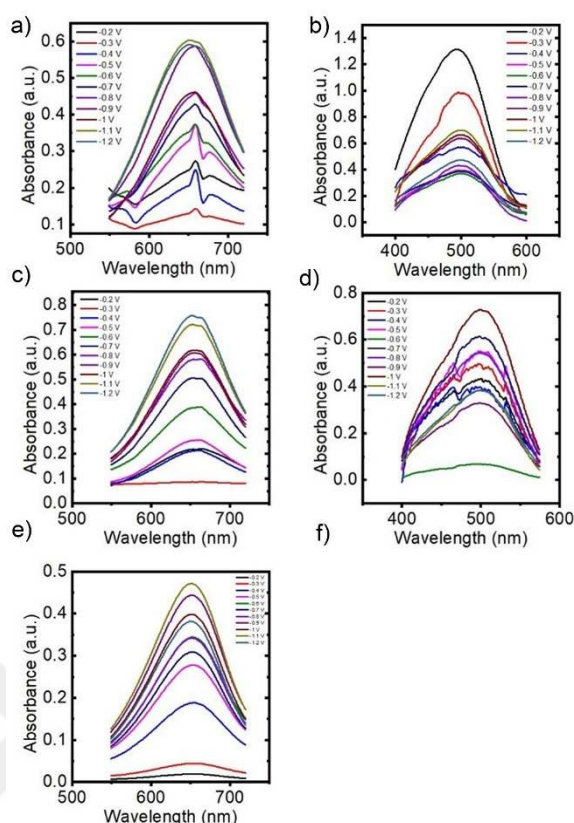


Figure 4-30: UV-vis spectroscopy results after pulsed NO_3RR and NO_2RR experiments at constant electrolysis potentials and pulsed potential of -0.4 V for 2 s. a) NH_3 quantification using the Indophenol Blue Method for Co/NS-rGO NO_3RR after pre-reduction at -1.5 V. b) NO_2 quantification using the Griess Test for Co/NS-rGO after pre-reduction at -1.5 V. c) NH_3 quantification using the Indophenol Blue Method for Co/NS-rGO NO_3RR without pre-reduction at -1.5 V. d) NO_2 quantification using the Griess Test for Co/NS-rGO without pre-reduction at -1.5 V. e) NH_3 quantification using the Indophenol Blue Method for Co/NS-rGO NO_2RR .

Even though NH_3 production starts at earlier potentials for pulsed electroreduction at -0.9 V, possibly owing to higher negative bias on the electrode, overall NH_3 F.E. is lower than the pulse at -0.4 V due to the superior catalytic properties of the material in mass-transfer-limited potentials. In the -0.9 V pulse, NO_3RR and NO_2RR for NS-rGO are heavily suppressed by HER, as shown by increasing H_2 yield rate and F.E. Interestingly, the highest NO_2 yield rate is achieved for NS-rGO at more negative potentials, possibly owing to increasing desorption at these potentials.

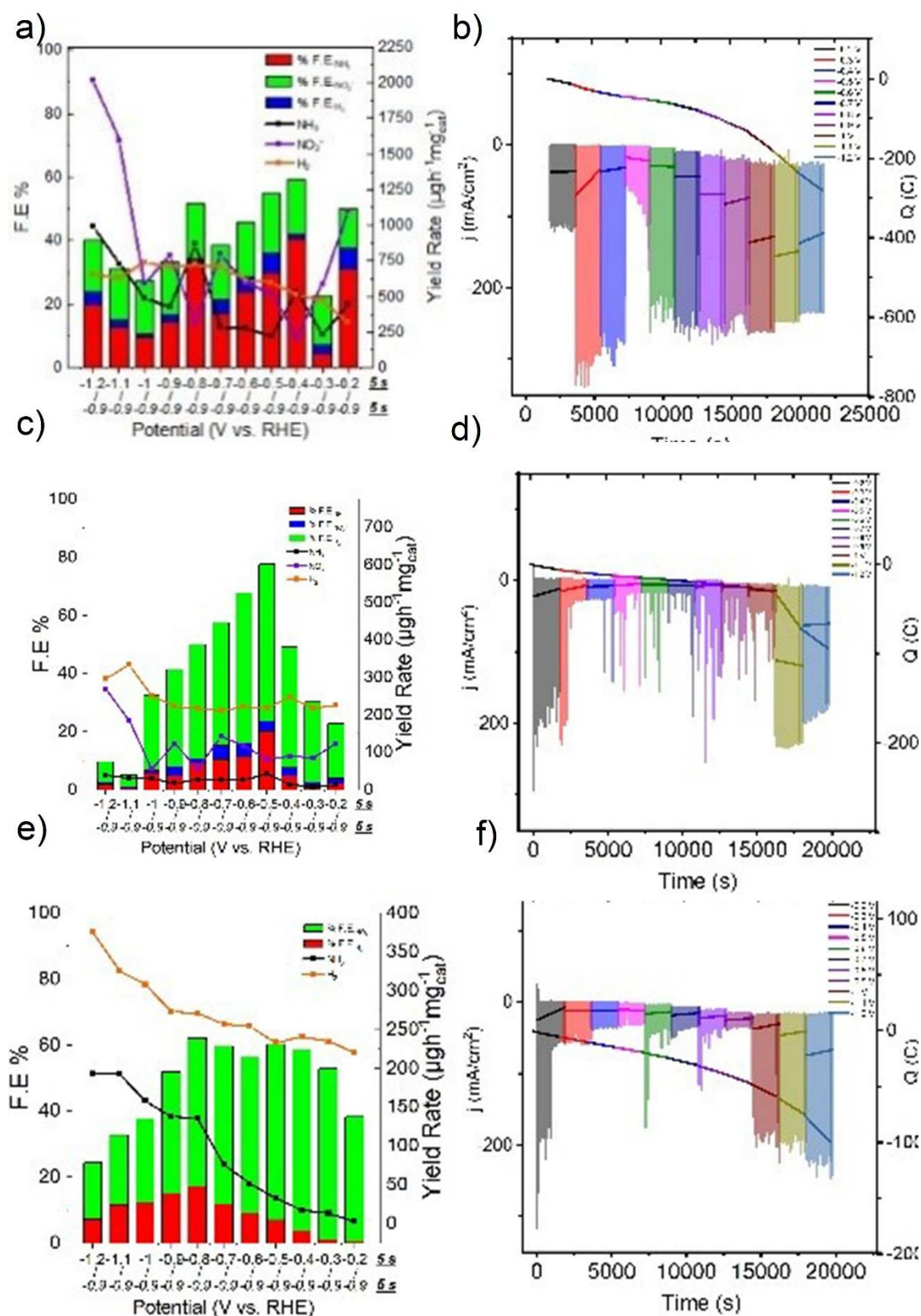


Figure 4-31: Pulsed Nitrate to ammonia electrosynthesis activity of NS-rGO for 2 s at constant electrolysis potentials and pulsed potential of -0.9 V. a) NO₂⁻, NH₃, H₂ F.E., and Y.R. of NS-rGO after pre-reduction at -1.5 V in 0.1 M KOH + 0.1 M KNO₃. b) j and q values for NS-rGO NO₃RR after pre-reduction at -1.5 V. c) NO₂⁻, NH₃, H₂ F.E. and Y.R. of NS-rGO in 0.1 M

KOH + 0.1 M KNO₃ d) j and q values for NS-rGO NO₃RR e) NO₂⁻, NH₃, H₂ F.E. and Y.R. of NS-rGO in 0.1 M KOH + 0.1 M KNO₂ f) j and q values for NS-rGO NO₂RR

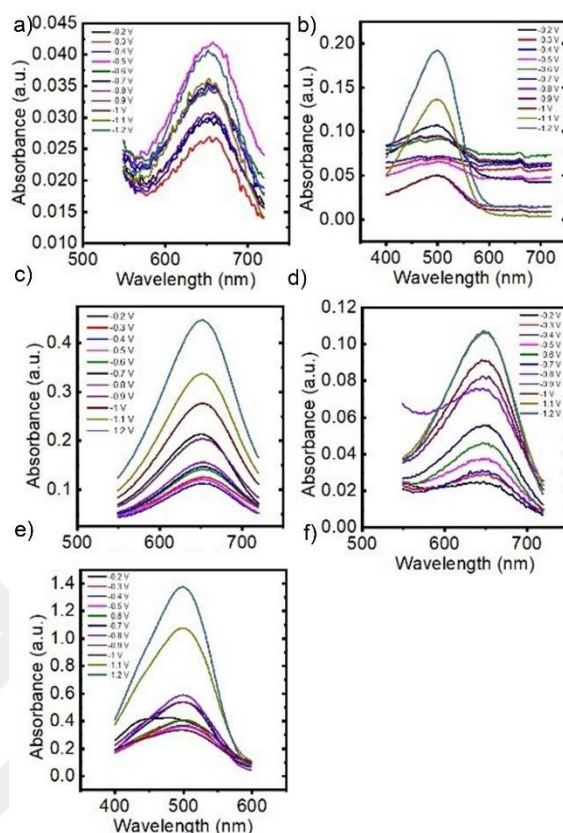


Figure 4-32: UV-vis spectroscopy results after pulsed NO₃RR and NO₂RR experiments at constant electrolysis potentials and pulsed potential of -0.9 V for 2 s. a) NH₃ quantification using the Indophenol Blue Method for NS-rGO NO₃RR after pre-reduction at -1.5 V. b) NO₂ quantification using the Griess Test for NS-rGO after pre-reduction at -1.5 V. c) NH₃ quantification using the Indophenol Blue Method for NS-rGO NO₃RR without pre-reduction at -1.5 V. d) NO₂ quantification using the Griess Test for NS-rGO without pre-reduction at -1.5 V. e) NH₃ quantification using the Indophenol Blue Method for NS-rGO NO₂RR.

Similar phenomena in NS-rGO of increasing H₂ yield rate and F.E. are also observed for Co/NS-rGO pulsed experiments at -0.9 V (Fig. 4-33). For Pre-red NO₃RR, even though there is a minor decrease in NH₃ F.E. compared to pulse at -0.4 V, the NH₃ yield rate is tripled compared to constant electrolysis and more than quadrupled compared to pulse at -0.4 V. This increase can be explained by the more favored reduction of desorbed NO₂ species to NH₃ at -0.9 V as seen by decreasing NO₂ yield and increased NH₃ yield rate at more negative potentials. The reason for decreasing NH₃ F.E. is explained by the increasing H₂ yield at -0.9 V pulse. However, this increase in H₂ yield is not counterintuitive for NH₃, possibly due to the superior protonation properties of the material. Also, more negative bias and less NO₃ concentration on the electrode are not a dominant parameter for Co/NS-rGO NO₃RR at the pre-reduced

electrode. The positive effect of -0.9 V pulse is not observed for NO₃RR at the bare electrode, possibly due to the inability of bare Co/NS-rGO to cause NO₂ desorption. Also, as of -0.4 V pulse, -0.9 V pulse does not favor NO₂RR but disturbs the facile reduction of NO₂ on Co/NS-rGO.

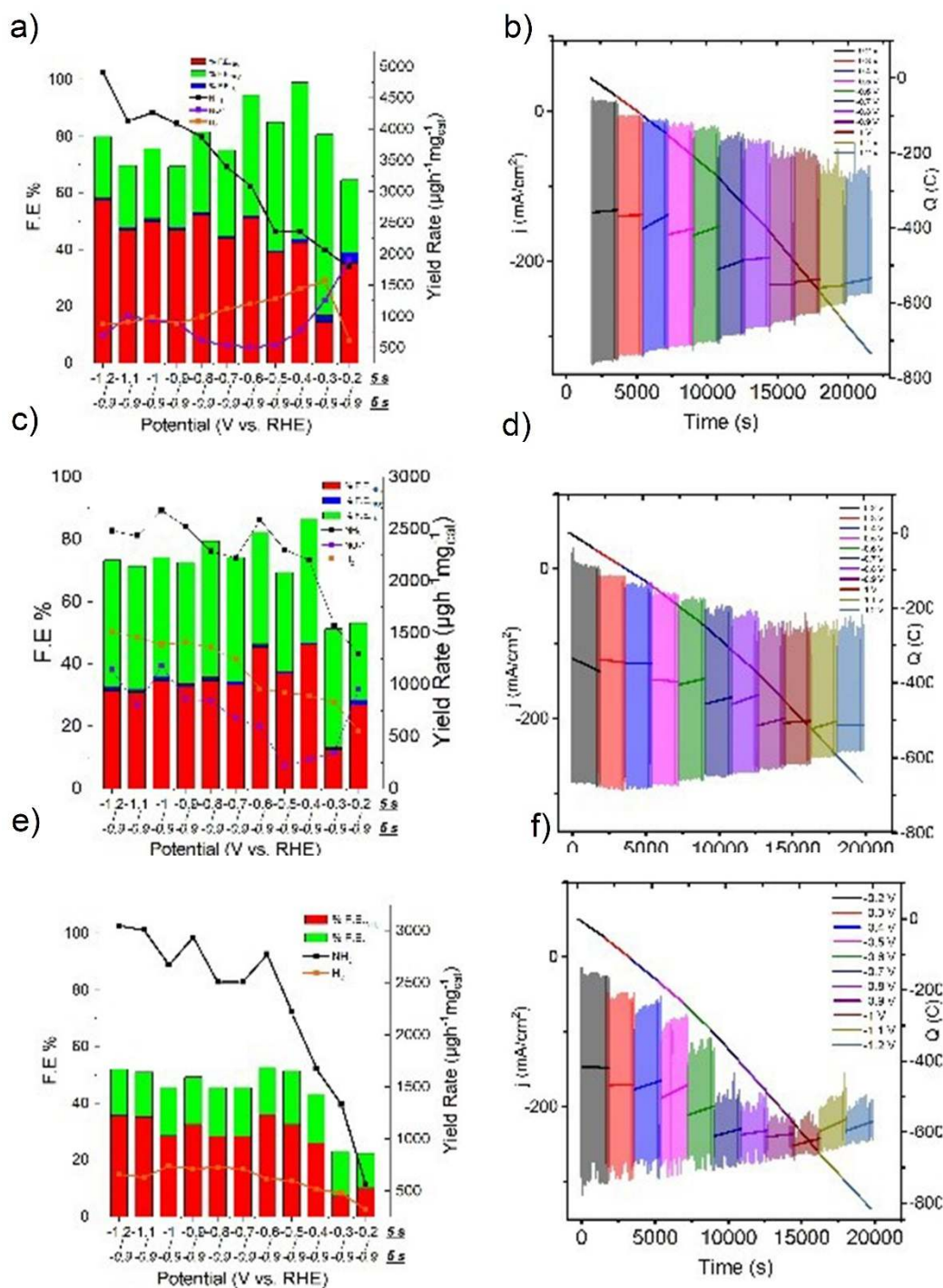


Figure 4-33: Pulsed Nitrate to ammonia electrosynthesis activity of Co/NS-rGO for 2 s at constant electrolysis potentials and pulsed potential of -0.9 V. a) NO₂⁻, NH₃, H₂ F.E., and Y.R. of Co/NS-rGO after pre-reduction at -1.5 V in 0.1 M KOH + 0.1 M KNO₃. b) j and q values for

Co/NS-rGO NO₃RR after pre-reduction at -1.5 V. c) NO₂⁻, NH₃, H₂ F.E. and Y.R. of Co/NS-rGO in 0.1 M KOH + 0.1 M KNO₃ d) j and q values for Co/NS-rGO NO₃RR e) NO₂⁻, NH₃, H₂ F.E. and Y.R. of Co/NS-rGO in 0.1 M KOH + 0.1 M KNO₂ f) j and q values for Co/NS-rGO NO₂RR

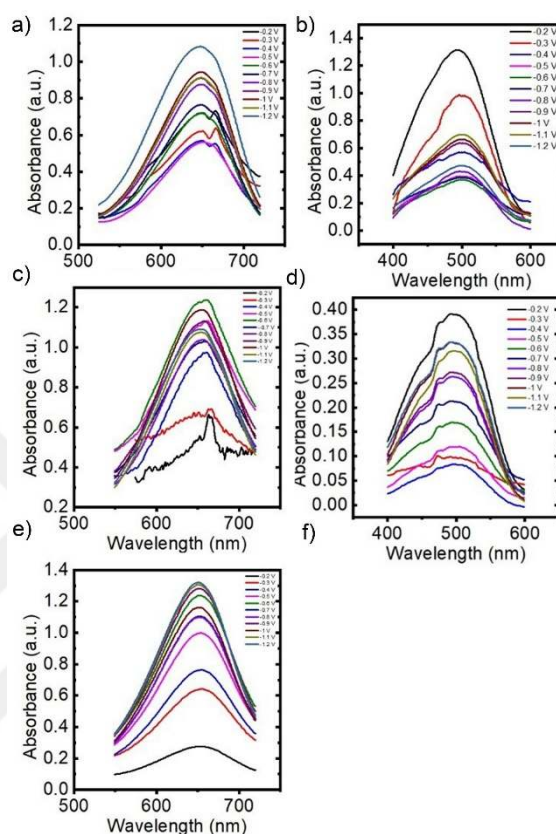


Figure 4-34: UV-vis spectroscopy results after pulsed NO₃RR and NO₂RR experiments at constant electrolysis potentials and pulsed potential of -0.9 V for 2 s. a) NH₃ quantification using the Indophenol Blue Method for Co/NS-rGO NO₃RR after pre-reduction at -1.5 V. b) NO₂ quantification using the Griess Test for Co/NS-rGO after pre-reduction at -1.5 V. c) NH₃ quantification using the Indophenol Blue Method for Co/NS-rGO NO₃RR without pre-reduction at -1.5 V. d) NO₂ quantification using the Griess Test for Co/NS-rGO without pre-reduction at -1.5 V. e) NH₃ quantification using the Indophenol Blue Method for Co/NS-rGO NO₂RR.

Pulsed electroreduction tests at -0.9 V for Co/NS-rGO 900 show that H₂ production is further increased for Co/NS-rGO 900 in these experiments owing to already superior HER properties of Co/NS-rGO is facilitated more than NH₃ production is suppressed (Fig. 4-35). Opposing Co/NS-rGO, Co/NS-rGO 900 has poor protonation properties, which is the reason for suppressed NO₃RR. Furthermore, Co/NS-rGO 900 has a lower NH₃ yield than Co/NS-rGO at pulse at -0.9 V. Its NH₃ yield is higher at constant CA measurements due to Co/NS-rGO 900 lacking NO₂ desorption properties. It is not mass-transfer limited at more positive potentials and poor protonation properties.

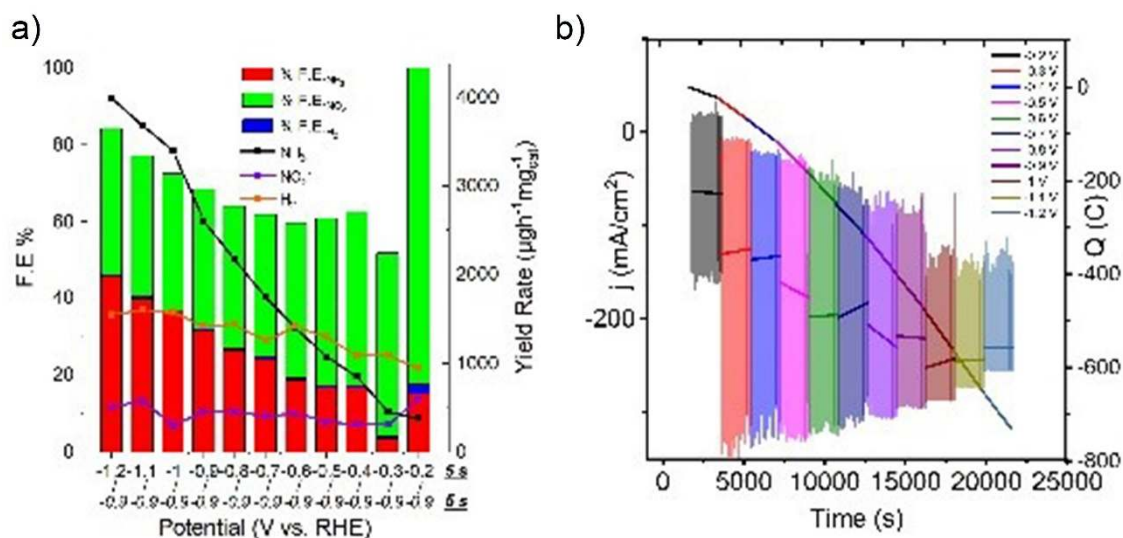


Figure 4-35: Pulsed Nitrate to ammonia electrosynthesis activity of Co/NS-rGO 900 for 2 s at constant electrolysis potentials and pulsed potential of -0.9 V. a) NO_3^- , NH_3 , H_2 F.E., and Y.R. of Co/NS-rGO after pre-reduction at -1.5 V in 0.1 M KOH + 0.1 M KNO_3 . b) j and q values for Co/NS-rGO NO_3RR after pre-reduction at -1.5 V.

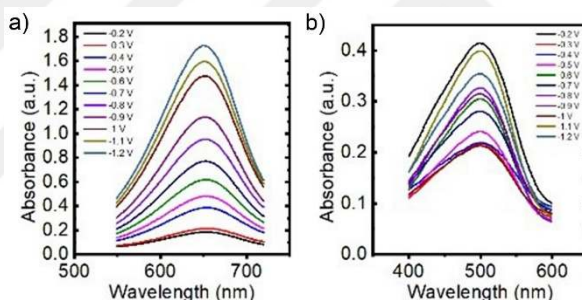


Figure 4-36: UV-vis spectroscopy results after pulsed NO_3RR and NO_2RR experiments at constant electrolysis potentials and pulsed potential of -0.9 V for 2 s. a) NH_3 quantification using the Indophenol Blue Method for Co/NS-rGO 900 NO_3RR after pre-reduction at -1.5 V. b) NO_2 quantification using the Griess Test for Co/NS-rGO 900 after pre-reduction at -1.5 V.

4.9 Conclusion

This work adopts Co/NS-rGO and Co/NS-rGO 900 catalysts as two model catalysts. The former consists of CoS moieties, while the latter has an amorphous structure with CoO particles. XAS study told us that two different mechanisms emerge for these catalysts: a mass-transfer limited mechanism for the former and a mechanism lacking a protonation step in NO_3RR for the latter. It was observed that the presence of CoS species facilitates NO_2RR and CoO species to promote HER. Pulsed electrolysis experiments results show that mass-transfer limited Co/NS-rGO favors more NH_3 production, possibly due to the reduction of desorbed NO_2 species or superior

protonation properties of the material, while pulsed CA measurements suppress NH_3 production for Co/NS-rGO 900 and facilitate HER .



Chapter 5:

Conclusion and Future Outlook

In this thesis, characterization and electrochemical screenings of Co decorated N, S doped Reduced Graphene Oxide (Co/NS-rGO) catalysts are carried out comprehensively. For the ORR study, we conclude that an optimum amount of Co (Co/NS-rGO) can induce CoS species along with Co SAs that facilitate ORR better than its more metal-loaded counterparts. We observed that CoS species favor the 4e⁻ pathway while CoO species lead to the 2e⁻ peroxide pathway. Also, the Co/NS-rGO catalyst is found to have better utilization of its active sites through ECSA-normalized RDE and RRDE analyses. Furthermore, we investigated the origin of ORR activity through the electronic properties of the material. The density of states, charge carrier concentration, flatband potential, and charge storage properties of Co/NS-rGO being the most superior, along with its high ORR activity, allows us to signify the importance of having an optimum composition of the metal and favorable surface species in the catalyst.

In the NO₃RR study, we investigated two different catalysts, one subjected to high heat treatment, i.e., Co/NS-rGO 900, while the other was not pyrolyzed. We claimed that these two catalysts follow two different reaction mechanisms by examining XAS behavior and Tafel Analysis. Co/NS-rGO is found to follow a mass-transfer limited mechanism where the Co/NS-rGO 900 is not mass-transfer limited but favors HER. Furthermore, we investigated the desorption of NO₂ species using EIS techniques and DRT models. Also, we studied pulsed electrolysis to understand whether the shift in negative bias on the electrode, mass-transfer limited potential regime, or reduction of desorbed species enables higher NH₃ yield. We discovered that pulsed electrolysis was detrimental to the already high NO₂RR activity of Co/NS-rGO and resulted in higher H₂ production for Co/NS-rGO 900 and suppressed NH₃ production. However, pulsed electrolysis was beneficial to NH₃ F.E. and yield of Co/NS-rGO, possibly due to the elimination of mass-transfer limitations, modulation of desorbed or adsorbed surface species, or altering the negative bias in the vicinity of the electrode.

We project that EIS techniques, kinetic analysis, and spectroscopic methods can be more accessible and informative in understanding reaction mechanisms and enhancing electrochemical activities. Also, very low metal loading, 1-5%, dispersion on high

surface area supports such as graphene with N and S elemental contents through doping can pave the way for electrochemically active surface species.

After improvements in the durability of the catalyst, one can operate transition metal dispersed high surface area catalysts in industrial applications to replace scarce and expensive PGM catalysts. The in-depth kinetic and mechanistic analysis of catalysts toward electrochemical reactions allows us to design more active catalysts that can perform in the desired range.



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